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DMIC Report 183

February 7, 1963

**BINARY AND TERNARY PHASE DIAGRAMS
OF
COLUMBIUM, MOLYBDENUM, TANTALUM,
AND TUNGSTEN
(Supplement to DMIC Report 152)**

**DEFENSE METALS INFORMATION CENTER
Battelle Memorial Institute
Columbus 1, Ohio**

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DMIC Report 183
February 7, 1963

BINARY AND TERNARY PHASE DIAGRAMS OF
COLUMBIUM, MOLYBDENUM, TANTALUM, AND TUNGSTEN

(Supplement to DMIC Report 152)

by

J. J. English

to

OFFICE OF THE DIRECTOR OF DEFENSE
RESEARCH AND ENGINEERING

DEFENSE METALS INFORMATION CENTER
Battelle Memorial Institute
Columbus 1, Ohio

ACKNOWLEDGMENT

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BINARY PHASE DIAGRAMS

<u>Identification Code</u>	
(5)-63	Columbium-Aluminum System
(5-1)-63	Columbium-Bismuth System
(7)-63	Columbium-Carbon System
(10)-63	Columbium-Cobalt System
(10-1)-63	Columbium-Copper System
(10-2)-63	Columbium-Gold System
(14-1)-63	Columbium-Manganese System
(16)-63	Columbium-Nickel System
(17)-63	Columbium-Nitrogen System
(18-1)-63	Columbium-Palladium System
(18-2)-63	Columbium-Platinum System
(22-1)-63	Columbium-Tellurium System
(24)-63	Columbium-Tin System
(28-1)-63	Columbium-Yttrium System
(37)-63	Molybdenum-Hafnium System
(43)-63	Molybdenum-Osmium System
(49)-63	Molybdenum-Rhodium System
(49-1)-63	Molybdenum-Ruthenium System
(55)-63	Molybdenum-Vanadium System
(55-1)-63	Molybdenum-Yttrium System
(57-1)-63	Tantalum-Aluminum System
(60)-63	Tantalum-Chromium System
(61)-63	Tantalum-Cobalt System
(61-1)-63	Tantalum-Gold System
(63-1)-63	Tantalum-Iridium System
(64-1)-63	Tantalum-Manganese System
(66)-63	Tantalum-Nitrogen System
(68-1)-63	Tantalum-Platinum System
(70-1)-63	Tantalum-Rhodium System
(73-1)-63	Tantalum-Thorium System
(77-1)-63	Tantalum-Yttrium System
(78)-63	Tantalum-Zirconium System
(80)-63	Tungsten-Boron System
(81)-63	Tungsten-Carbon System
(84-1)-63	Tungsten-Iridium System
(85-1)-63	Tungsten-Lead System
(89)-63	Tungsten-Oxygen System
(89-1)-63	Tungsten-Palladium System
(92-1)-63	Tungsten-Rhodium System
(98-1)-63	Tungsten-Yttrium System

TERNARY PHASE DIAGRAMS

(99-1)-63	Columbium-Aluminum-Nickel System ($\text{Ni}_3\text{Al}-\text{Ni}_3\text{Cb}$ System)
(99-2)-63	Columbium-Aluminum-Silicon System (1400 C)
(99-3)-63	Columbium-Aluminum-Titanium System (20 C)

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(Continued)

<u>Identification Code</u>	
(99-4)-63	Columbium-Aluminum-Titanium System (1200 C)
(99-5)-63	Columbium-Boron-Silicon System (1600 C)
(99-6)-63	Columbium Carbide-Hafnium Carbide-Uranium Carbide System (2050 C)
(99-7)-63	Columbium Carbide-Hafnium Carbide-Vanadium Carbide System (2050 C)
(99-8)-63	Columbium-Carbon-Uranium System (1700 C)
(99-9)-63	Columbium-Chromium-Nickel System (1100 C)
(99-10)-63	Columbium-Chromium-Silicon System (1000 C)
(99-11)-63	Columbium-Chromium-Silicon System (Melting Temperatures, C)
(99-12)-63	Columbium-Chromium-Vanadium System (1450 C)
(99-13)-63	Columbium-Iron-Phosphorus System (Room Temperature)
(102-1)-63	Columbium-Molybdenum-Carbon System (1900 C)
(102-2)-63	Columbium-Molybdenum-Chromium System (1000 C)
(102-3)-63	Columbium-Molybdenum-Chromium System (1200 C)
(102-4)-63	Columbium-Molybdenum-Chromium System (Liquidus Isotherms)
(105-1)-63	Columbium-Molybdenum-Uranium System (700 C, 800 C, 900 C)
(105-2)-63	Columbium-Molybdenum-Uranium System (1100 C)
(105-3)-63	Columbium-Molybdenum-Uranium System (1200 C)
(105-4)-63	Columbium-Molybdenum-Uranium System (1300 C)
(107-1)-63	Columbium-Nitrogen-Oxygen System (1500 C)
(107-2)-63	Columbium-Oxygen-Titanium System (1500 C)
(107-3)-63	Columbium-Oxygen-Zirconium System (1500 C)
(107-4)-63	Columbium-Silicon-Vanadium System (Schematic)
(116-1)-63	Columbium-Titanium-Zirconium System (Melting Temperatures, C)
(118-1)-63	Columbium-Uranium-Zirconium System (1000 C)
(122-1)-63	Columbium-Tungsten-Silicon System (Room Temperature)
(123-1)-63	Columbium-Tungsten-Zirconium System (1100 C)
(123-2)-63	Columbium-Tungsten-Zirconium System (1600 C)
(123-3)-63	Columbium-Tungsten-Zirconium System (1800 C)
(123-4)-63	Columbium-Tungsten-Zirconium System (2000 C)
(123-5)-63	Molybdenum-Aluminum-Nickel System (1175 C)
(123-6)-63	Molybdenum-Aluminum-Silicon System (1600 C)
(129-1)-63	Molybdenum-Boron-Nickel System (1000 C)
(130-1)-63	Molybdenum-Boron-Titanium System (1200 C)
(130-2)-63	Molybdenum-Boron-Titanium System (1700 C)
(148-1)-63	Molybdenum-Hafnium-Rhenium System (1600 C)
(148-2)-63	Molybdenum-Hafnium-Rhenium System (2000 C)
(148-3)-63	Molybdenum-Hafnium-Rhenium System (2400 C)
(149-1)-63	Molybdenum-Iron-Phosphorus System (Room Temperature)
(149-2)-63	Molybdenum-Iron-Silicon System (20 C)
(155-1)-63	Molybdenum-Nickel-Silicon System (1100 C)
(155-2)-63	Molybdenum-Nickel-Titanium System (1175 C)
(157-1)-63	Molybdenum-Oxygen-Titanium System (1500 C)
(157-2)-63	Molybdenum-Oxygen-Zirconium System (1500 C)
(157-3)-63	Molybdenum-Silicon-Vanadium System (800 C)
(162-1)-63	Molybdenum-Titanium-Vanadium System (Room Temperature)
(162-2)-63	Molybdenum-Titanium-Zirconium System (1500 C)
(164-1)-63	Molybdenum-Tungsten-Chromium System (1800 C)
(167-1)-63	Molybdenum-Tungsten-Silicon System (1900 C)
(167-2)-63	Molybdenum-Uranium-Vanadium System (500 C)
(167-3)-63	Molybdenum-Uranium-Vanadium System (600 C)
(167-4)-63	Molybdenum-Uranium-Vanadium System (1000 C)
(167-5)-63	Molybdenum-Uranium-Zirconium System (1000 C)
(167-6)-63	Tantalum-Aluminum-Silicon System (1400 C)
(167-7)-63	Tantalum-Boron-Nickel System (TaB ₂ -Ni)
(168-1)-63	Tantalum Carbide-Hafnium Carbide-Uranium Carbide System (2050 C)

TABLE OF CONTENTS (Continued)

<u>Identification Code</u>	
(168-2)-63	Tantalum Carbide-Hafnium Carbide-Vanadium Carbide System (2050 C)
(169-1)-63	Tantalum-Carbon-Uranium System (1700 C)
(169-2)-63	Tantalum-Chromium-Nickel System (Solidus Temperature)
(173-1)-63	Tantalum-Cobalt-Iron System (20 C)
(173-2)-63	Tantalum-Cobalt-Iron System (1390 C)
(178-1)-63	Tantalum-Silicon-Titanium System ($TaSi_2-TiSi_2$)
(178-2)-63	Tantalum-Silicon-Vanadium System (Schematic)
(186-1)-63	Tantalum-Tungsten-Silicon System ($TaSi_2-WSi_2$)
(187-1)-63	Tungsten-Aluminum-Nickel System (800 C)
(187-2)-63	Tungsten-Aluminum-Nickel System (1200 C)
(187-3)-63	Tungsten-Aluminum-Silicon System (1500 C)
(188-1)-63	Tungsten-Boron-Thorium System (1800 C)
(194-1)-63	Tungsten-Carbon-Thorium System (1500 C)
(195-1)-63	Tungsten-Carbon-Uranium System (1000 and 1500 C)
(195-2)-63	Tungsten-Carbon-Vanadium System (1500 and 1800 C)
(195-3)-63	Tungsten-Chromium-Cobalt System (Room Temperature)
(202-1)-63	Tungsten-Chromium-Silicon System (1500 C)
(203-1)-63	Tungsten-Cobalt-Iron System (20 C)
(203-2)-63	Tungsten-Cobalt-Iron System (1400 C)
(203-3)-63	Tungsten-Germanium-Silicon System (1000 C)
(206-1)-63	Tungsten-Iron-Sulfur System (Room Temperature)
(209-1)-63	Tungsten-Silicon-Titanium System ($TiSi_2-WSi_2$ Schematic)

BIBLIOGRAPHY

BINARY AND TERNARY PHASE DIAGRAMS OF COLUMBIUM,
MOLYBDENUM, TANTALUM, AND TUNGSTEN

SUMMARY

This report supplements DMIC Report 152 which is a compilation of binary and ternary phase diagrams of columbium, molybdenum, tantalum, and tungsten. Forty new binary and 80 new ternary diagrams are included, some of these being revised versions of the previously published diagrams. Included with each binary diagram and with some ternary diagrams is a short discussion listing terminal solubilities and crystal structures of intermediate phases. Many of the diagrams are tentative and are subject to revision as additional data become available.

INTRODUCTION

Increased interest in the refractory metals columbium, molybdenum, tantalum, and tungsten is reflected in the growing number of phase diagrams that are being determined for alloys of these metals. In DMIC Report 152, phase diagrams of alloy systems based on the four metals were assembled. Occasional supplements to DMIC Report 152 have been planned in order that up-to-date information on phase relationships in the refractory-metal-alloy systems will be available to those who need it. Data on approximately 70 additional systems have been collected by DMIC since the publication of Report 152. They are presented here along with new data on systems already reported on in DMIC 152.

It is hoped that the users of this report will supply the Defense Metals Information Center with additional phase-diagram information as it becomes available and also with any literature references that might have been inadvertently overlooked.

ORGANIZATION OF THE REPORT

The phase diagrams in this report are presented in two sections - one on binary and the other for ternary phase diagrams. Within each section, the diagrams are subdivided into four groups according to base-metal system - columbium, molybdenum, tantalum, and tungsten. The systems are then arranged in alphabetical order in these groups according to the spelling of the second element in the system. When two refractory metals occur in the same ternary system, they are listed first. For example, the columbium-molybdenum-carbon system is not listed as the columbium-carbon-molybdenum system.

Each diagram is printed on a separate page and has a code number at the lower left corner of the page to assist in relating this report with DMIC Report 152, which it supplements. As an example of the use of this code, the revised columbium-aluminum system has been given the designation (5)-63. The (5) refers to the number in the lower corner of the now out-dated columbium-aluminum system found in DMIC Report 152. The columbium-manganese system, coded (14-1)-63 is a completely new system and follows the columbium-lanthanum system, coded (14) in DMIC Report 152.

Reference numbers in this report are a continuation of the references in the bibliography of DMIC Report 152. Thus any reference of lower number than 234 (the first reference in this report) will be found in the bibliography section of DMIC Report 152.

NOTES ON DIAGRAMS IN DEFENSE METALS INFORMATION CENTER REPORT 152

The following notes refer to the phase diagrams which appeared in DMIC Report 152, April, 1961; there are no new diagrams to accompany them in this report.

Molybdenum-Aluminum System:

The compound Al_7Mo has been identified by Claire⁽³²²⁾. Al_7Mo has a monoclinic structure with $a = 5.12\text{\AA}$, $b = 13.0\text{\AA}$, $c = 13.5\text{\AA}$, and $\beta = 95$ degrees. A peritectic reaction occurs at $706 \pm 3^\circ\text{C}$ where Al_7Mo reacts with liquid to form Al_{12}Mo .

Tantalum-Chromium System:

The solubility of tantalum in chromium was reported to increase from 1.5 weight per cent at 1200°C to 8 weight per cent at 1600°C .⁽³²³⁾

Tantalum-Ruthenium System:

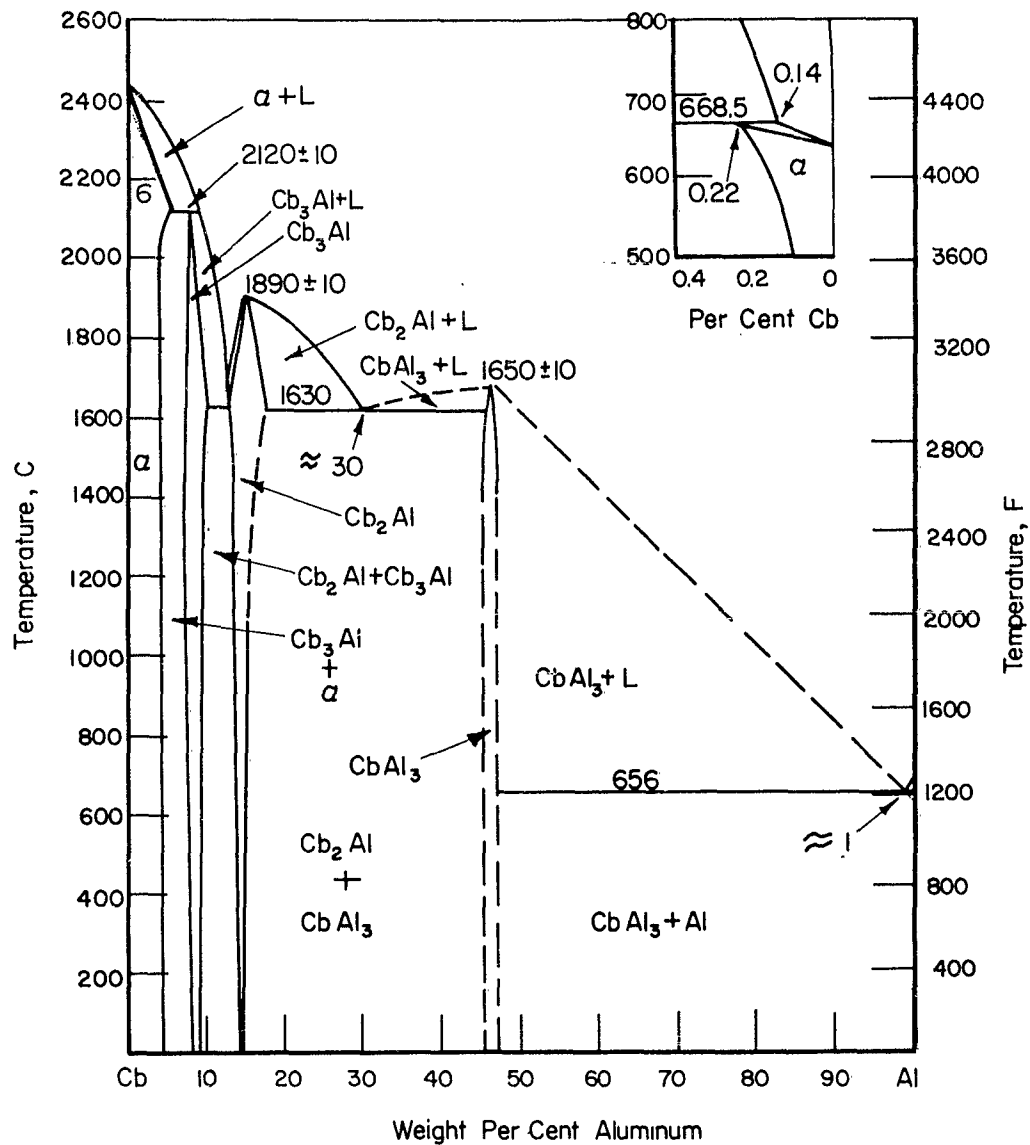
The μ phase is based on a CsCl B-2 type structure at 30 atomic per cent ruthenium, transforming to a distorted CsCl structure at 40 and 45 atomic per cent ruthenium.⁽³²⁴⁾

Tungsten-Hafnium System:

The previously reported value for the transformation of hafnium of $1875 \pm 20^\circ\text{C}$ has been observed to occur at $1750 \pm 20^\circ\text{C}$ by Deardorf⁽³²⁵⁾. Giessen⁽³²⁾ has verified the new value in more recent studies.

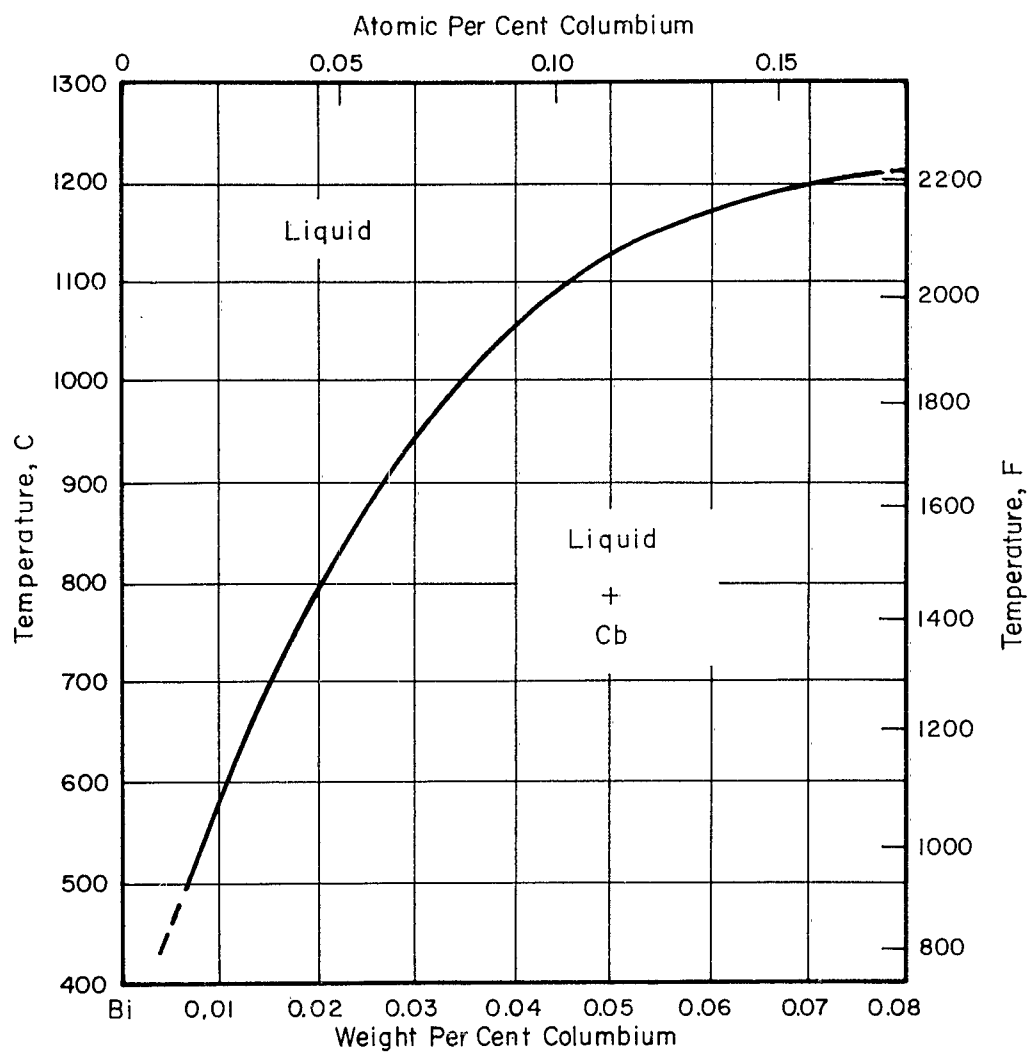
BINARY PHASE
DIAGRAMS

COLUMBIUM-ALUMINUM SYSTEM



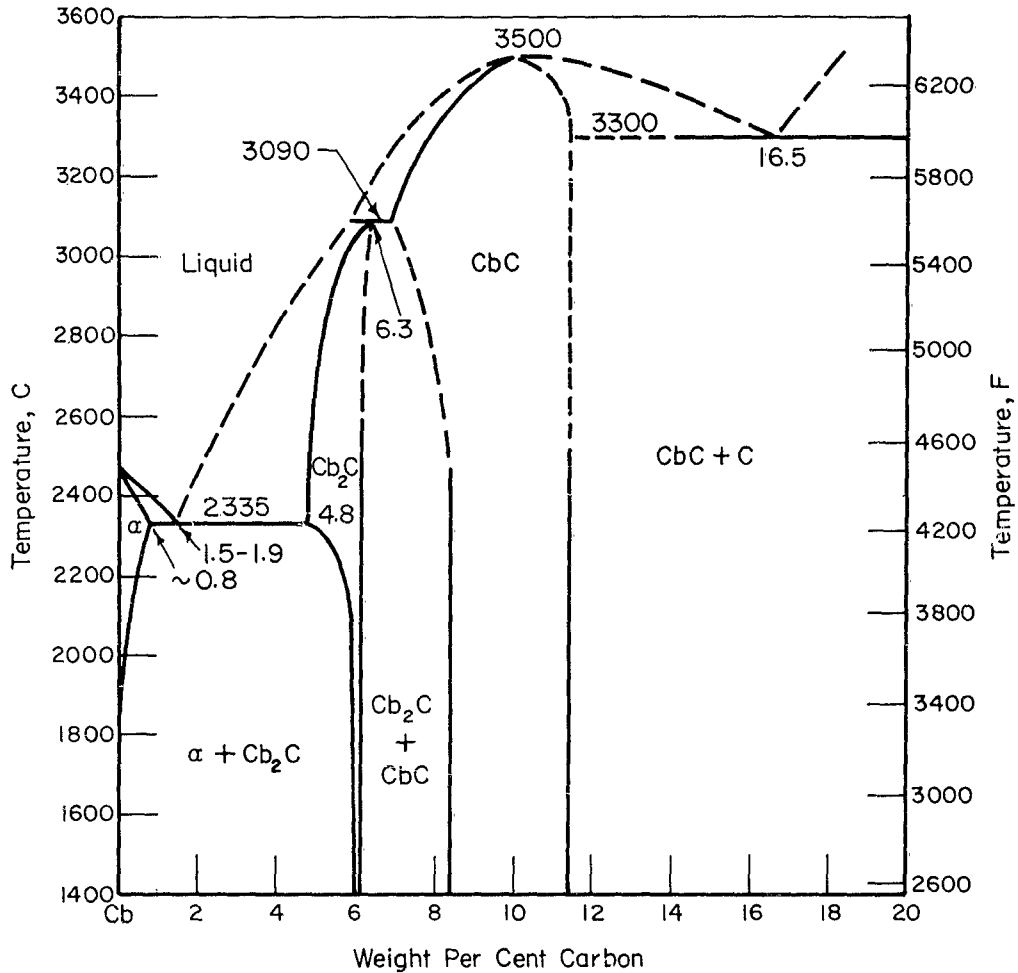
Cb₃Al has a cubic, β -tungsten type structure with $a = 5.187 \text{ \AA}$.⁽¹⁾ Cb₂Al is tetragonal (σ -phase) with $a = 5.438 \text{ \AA}$, $c = 8.601 \text{ \AA}$, and $c/a = 1.582$.⁽³⁾ Cb₂Al has been reported to form by a peritectic reaction at 1890°C⁽²³⁴⁾ instead of melting congruently, as shown above.⁽²³⁵⁾ The columbium-rich boundary of the Cb₂Al region was reported to be near 16 weight per cent aluminum (41 atomic per cent) at 1250°C.⁽²³⁶⁾ Aluminum is soluble in columbium up to 6 weight per cent at 2120°C, decreasing to 4.5 per cent at room temperature.⁽²³⁴⁾

COLUMBIUM-BISMUTH SYSTEM



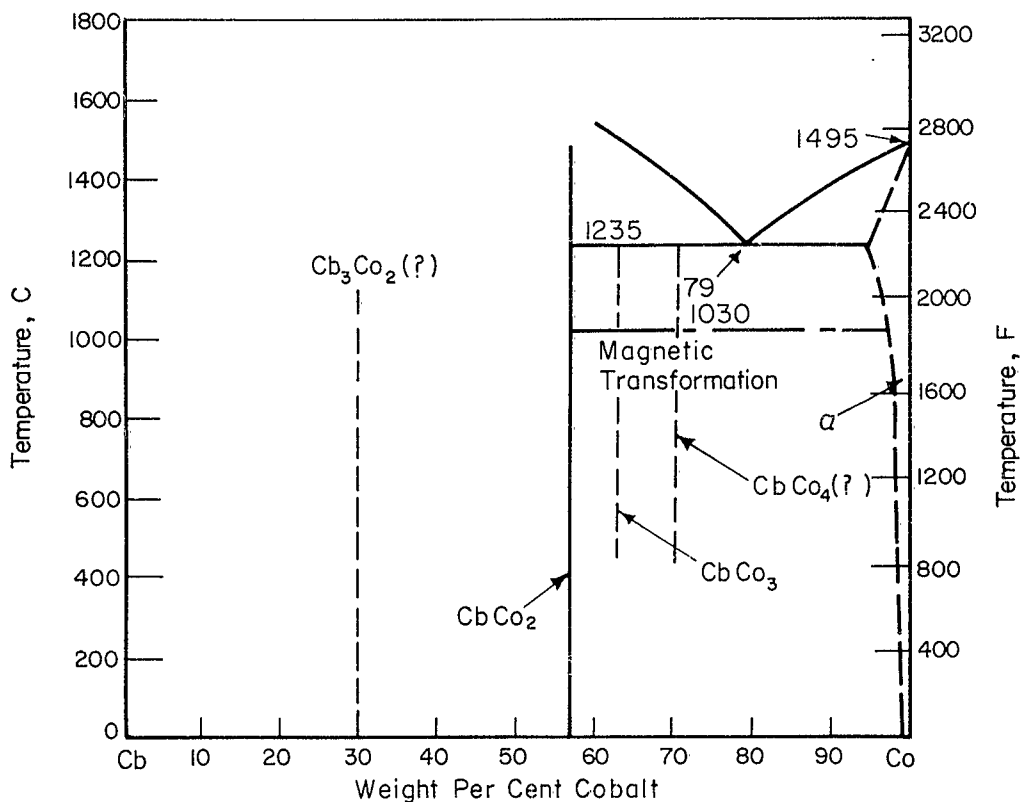
The solubility of columbiu in liquid bismuth (mp = 271 C) is shown above.⁽²³⁷⁾ There was no indication of the formation of intermetallic compounds in the system.

COLUMBIUM-CARBON SYSTEM



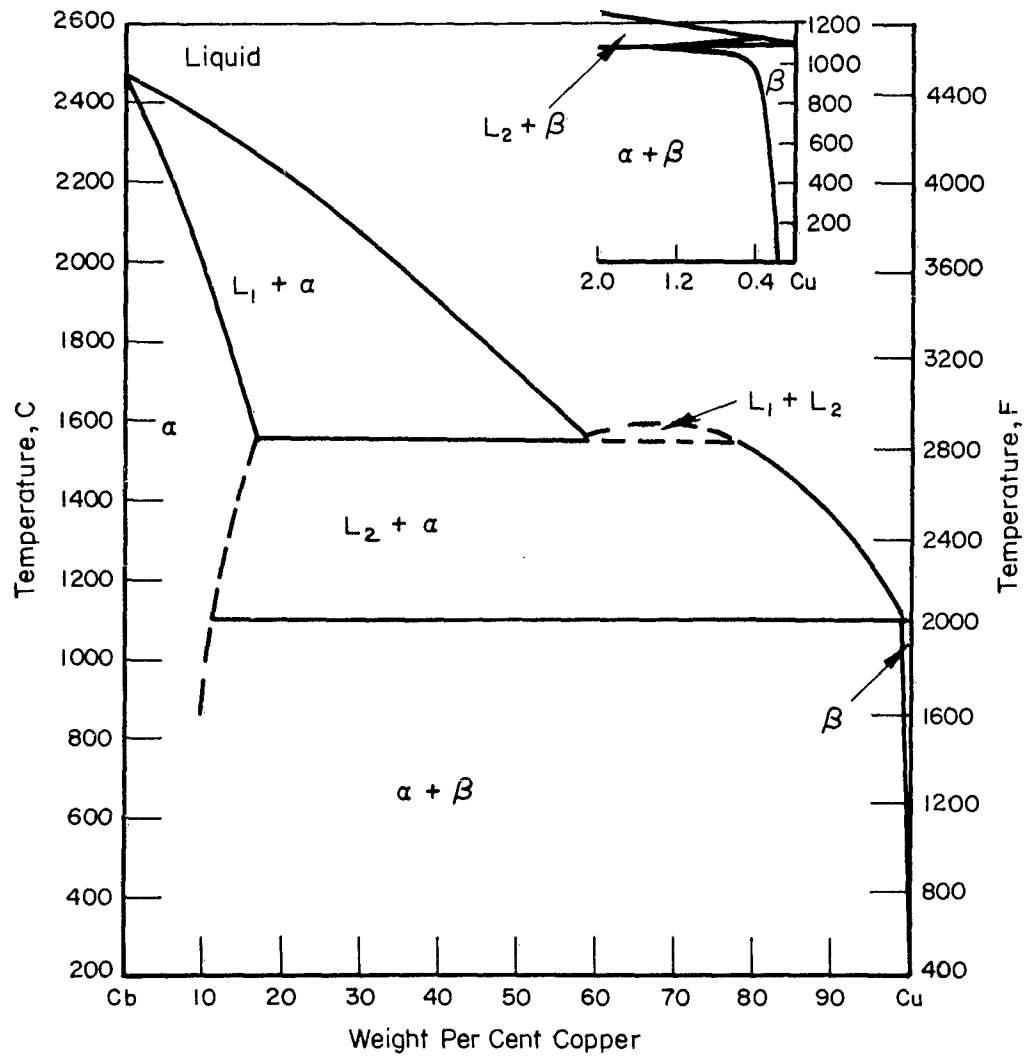
Cb₂C has a hexagonal structure with the single-phase region ranging from CbC_{0.39} at the eutectic temperature to a maximum of CbC_{0.52} at the peritectic temperature.⁽²³⁸⁾ CbC is face-centered cubic with $a = 4.470$ at 25°C.^(8,9,238) The maximum melting point of CbC occurs at a composition of CbC_{0.86}. Beyond the maximum melting point, the solidus drops to meet the CbC-C eutectic at 3300°C and 60.5 atomic per cent carbon.⁽²³⁹⁾ The maximum solubility of carbon in columbium is 0.7 to 0.8 weight per cent.^(10,239)

COLUMBIUM-COBALT SYSTEM



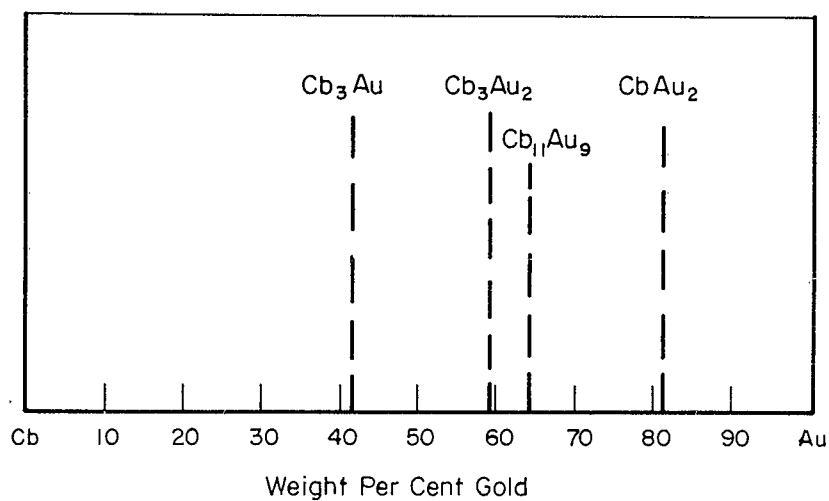
$CbCo_2$ is believed to exist in two modifications. One is at 33.3 atomic per cent columbium, and has a cubic $MgCu_2$ (C15) type of structure with $a = 6.758 \text{ \AA}$; the second structure exists around 27 atomic per cent columbium and has the $MgNi_2$ (C36) type of structure with $a = 4.738 \text{ \AA}$, $c = 15.46 \text{ \AA}$, and $c/2a = 1.631$.^(14, 15, 16) Two additional phases were reported to exist at 1100 C, Cb_3Co_2 and $CbCo_4$.⁽²⁴¹⁾ The solubility of columbium in cobalt is about 5 weight per cent at 1100 C.⁽²⁴⁰⁾

COLUMBIUM-COPPER SYSTEM



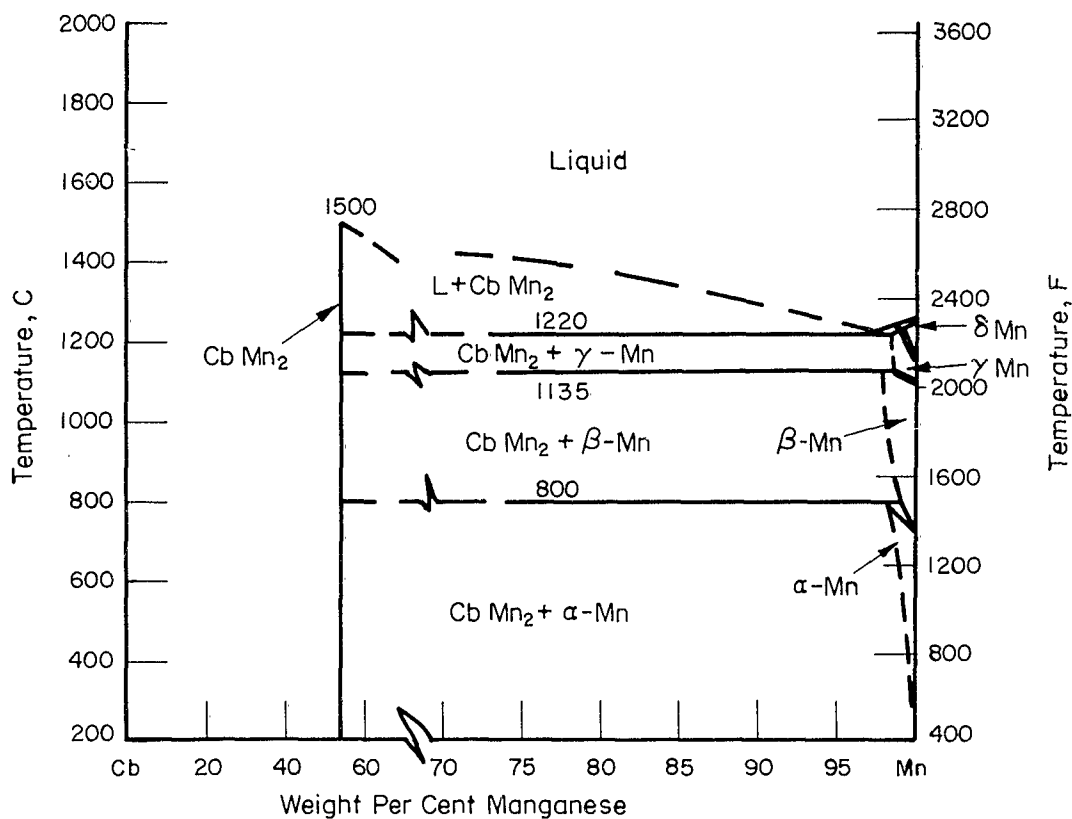
No intermetallic compounds are found in the columbium-copper system.⁽²⁴¹⁾

COLUMBIUM-GOLD SYSTEM



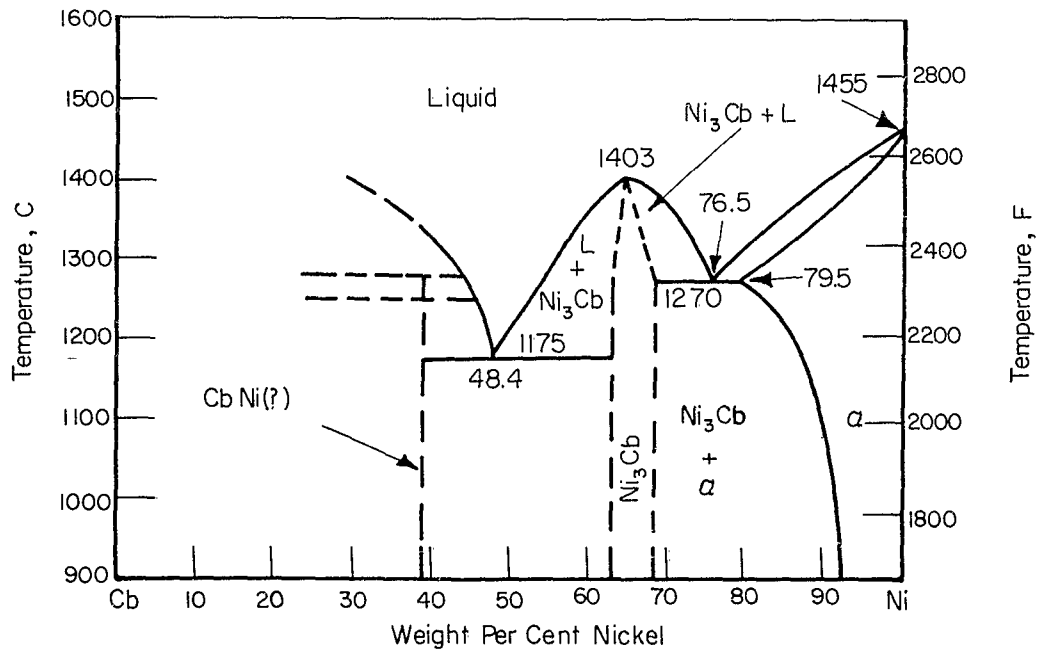
Cb_3Au is a cubic Cr_3O -type compound with $a = 5.20$ kX. Cb_3Au_2 has a $D_{4h}^{17}\text{-I4}/\text{mm}$ structure with $a = 3.37$ kX and $c = 5.3.03$ kX. $\text{Cb}_{11}\text{Au}_9$ possesses a β -manganese structure with $a = 7.04$ kX. The structure of CbAu_2 is of the B_2Al type with $a = 4.60$ kX and $c = 2.71$ kX.⁽²⁴²⁾

COLUMBIUM-MANGANESE SYSTEM



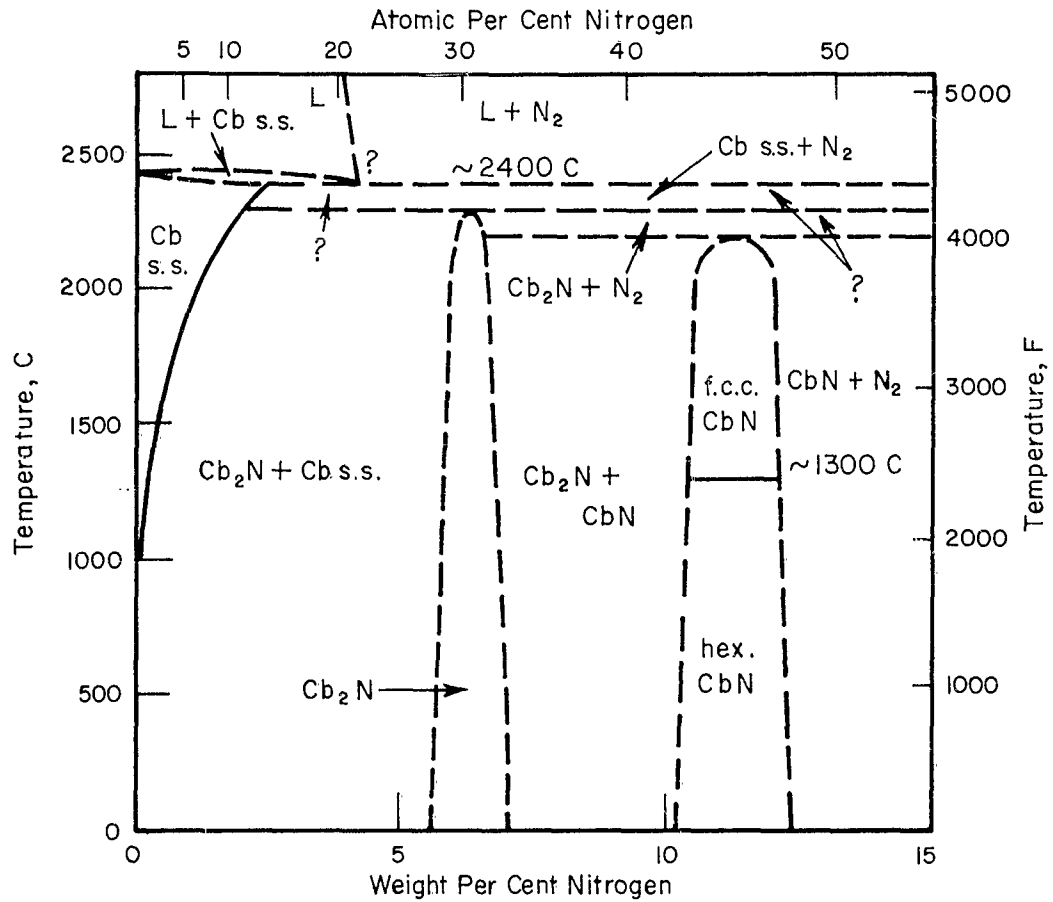
CbMn₂ has a hexagonal MgZn₂-type structure with $a = 4.881 \text{ kX}$, $c = 7.953 \text{ kX}$, and $c/a = 1.629$. Alloys containing less than 2 weight per cent columbium transform from γ -manganese to δ -manganese by a peritectic reaction near the melting point. (299)

COLUMBIUM-NICKEL SYSTEM



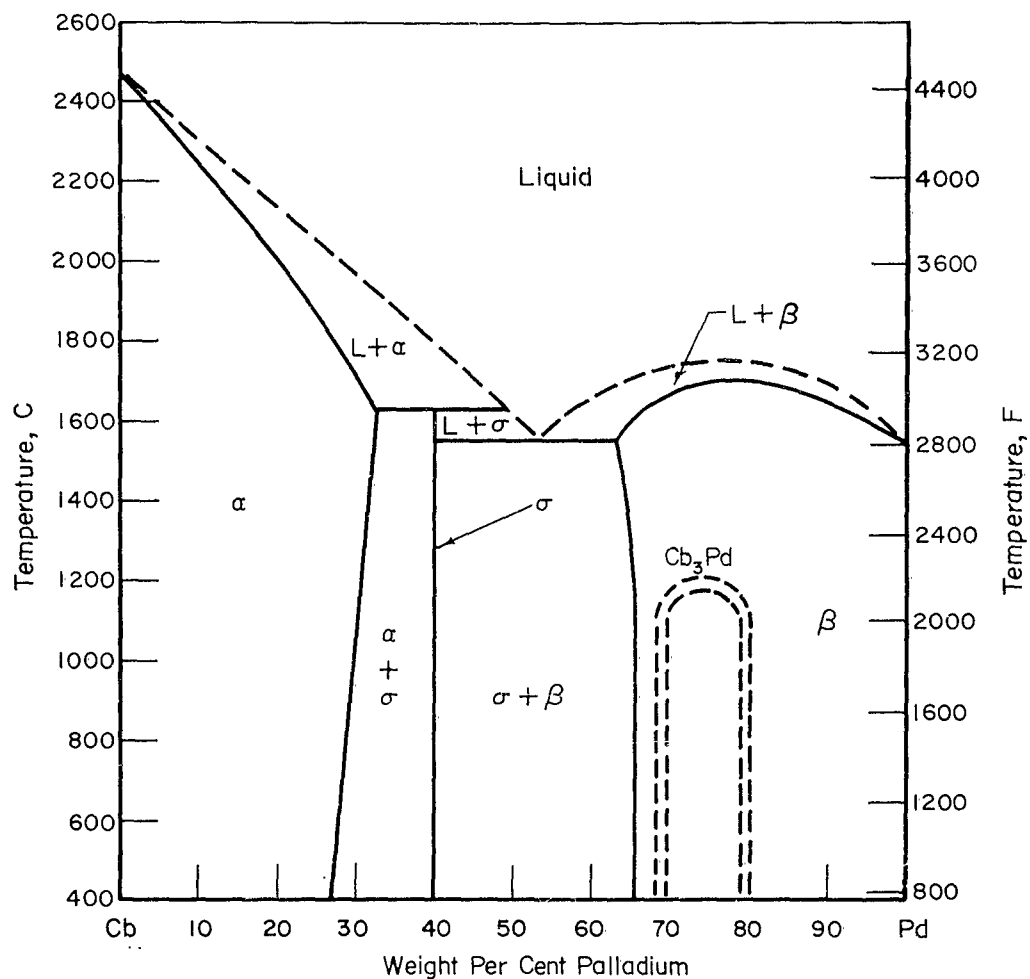
$CbNi_3$ has an orthorhombic $TiCu_3$ -type structure with lattice parameters: $a = 5.10$ kX, $b = 4.55$ kX, and $c = 4.25$ kX.⁽²³⁾ The solubility of columbium in nickel is approximately 15 weight per cent at 1250 C. The solubility of nickel in columbium is less than 5 weight per cent.⁽²⁴⁾ The diagram was constructed from the data prepared by Pogodin and Selekman.⁽²⁵⁾

COLUMBIUM-NITROGEN SYSTEM



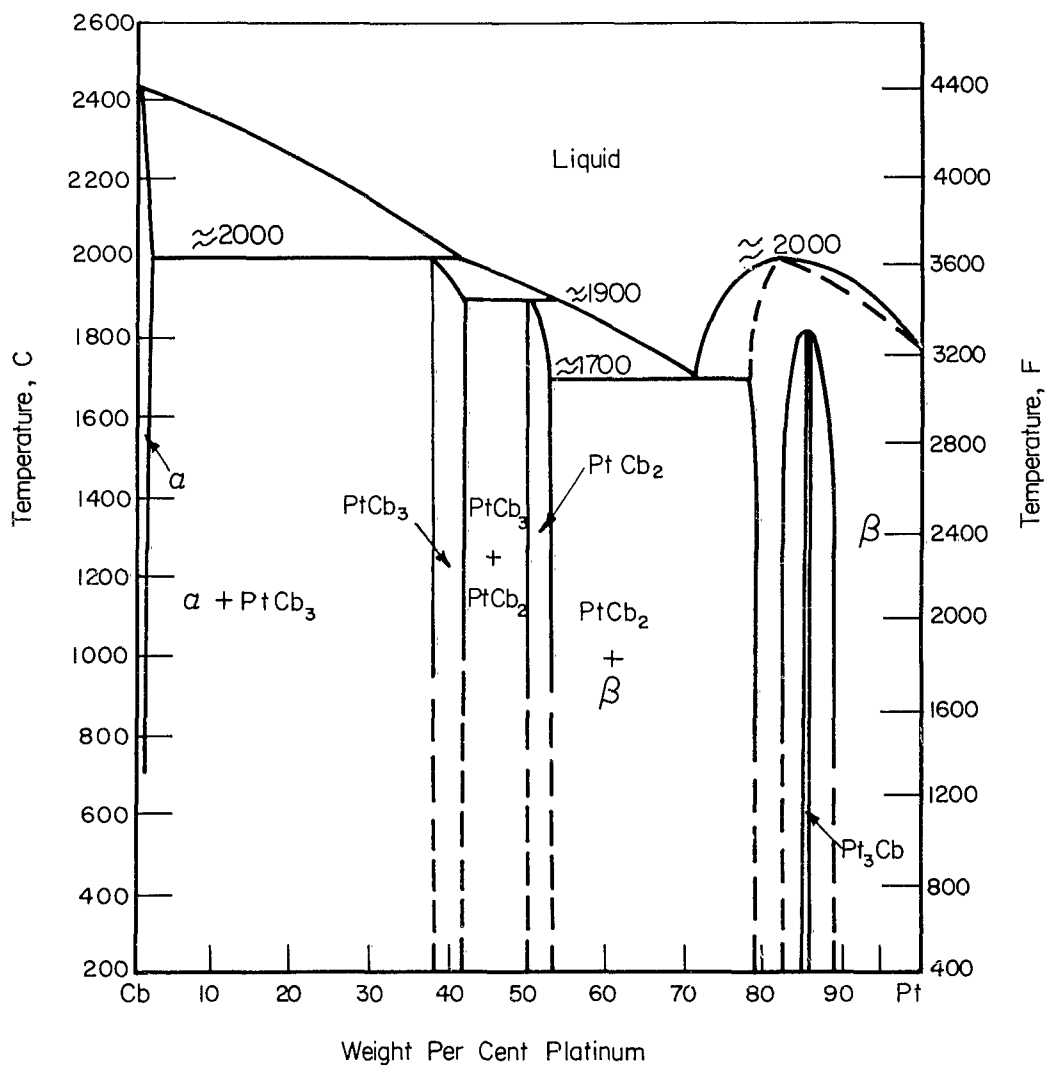
Cb_2N is a close-packed-hexagonal structure, with the lattice parameters varying between $a = 3.057 \text{ \AA}$, $c = 4.957 \text{ \AA}$ ($c/a = 1.622$) and $a = 3.050 \text{ \AA}$, $c = 5.005 \text{ \AA}$ ($c/a = 1.641$).^(21, 243) The high-temperature structure of CbN is face-centered cubic with $a = 4.386$ to 4.394 \AA .⁽²⁴³⁾ The low-temperature structure of CbN is hexagonal, with the lattice parameters varying between $a = 2.953 \text{ \AA}$, $c = 11.243 \text{ \AA}$ ($c/a = 3.804$) and $a = 2.953 \text{ \AA}$, $c = 11.257 \text{ \AA}$ ($c/a = 3.813$).^(22, 243) The solubility of nitrogen in columbium is 0.25 weight per cent at 1200 C and 2.5 weight per cent at 2400 C.

COLUMBIUM-PALLADIUM SYSTEM



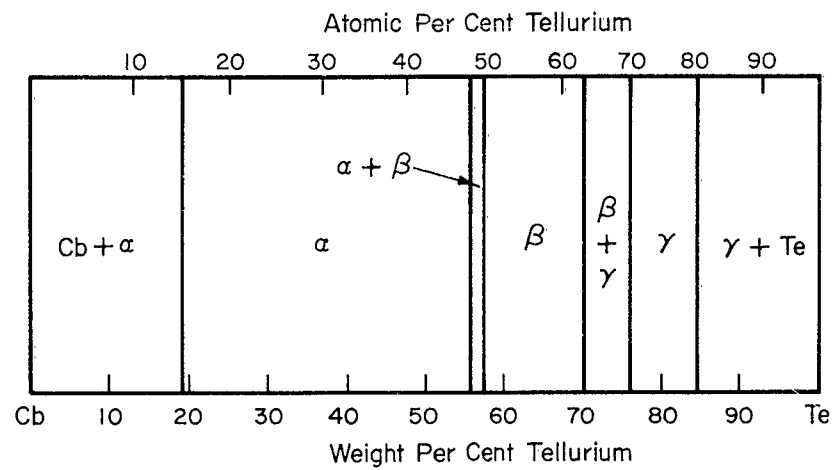
One intermediate phase having a tetragonal σ -phase structure, with $a = 9.89 \text{ \AA}$, $c = 5.11 \text{ \AA}$, $c/a = 0.52$, was reported.⁽³⁰⁾ An ordered structure is possibly present at the Cb_3Pd composition.⁽²⁴⁴⁾

COLUMBIUM-PLATINUM SYSTEM



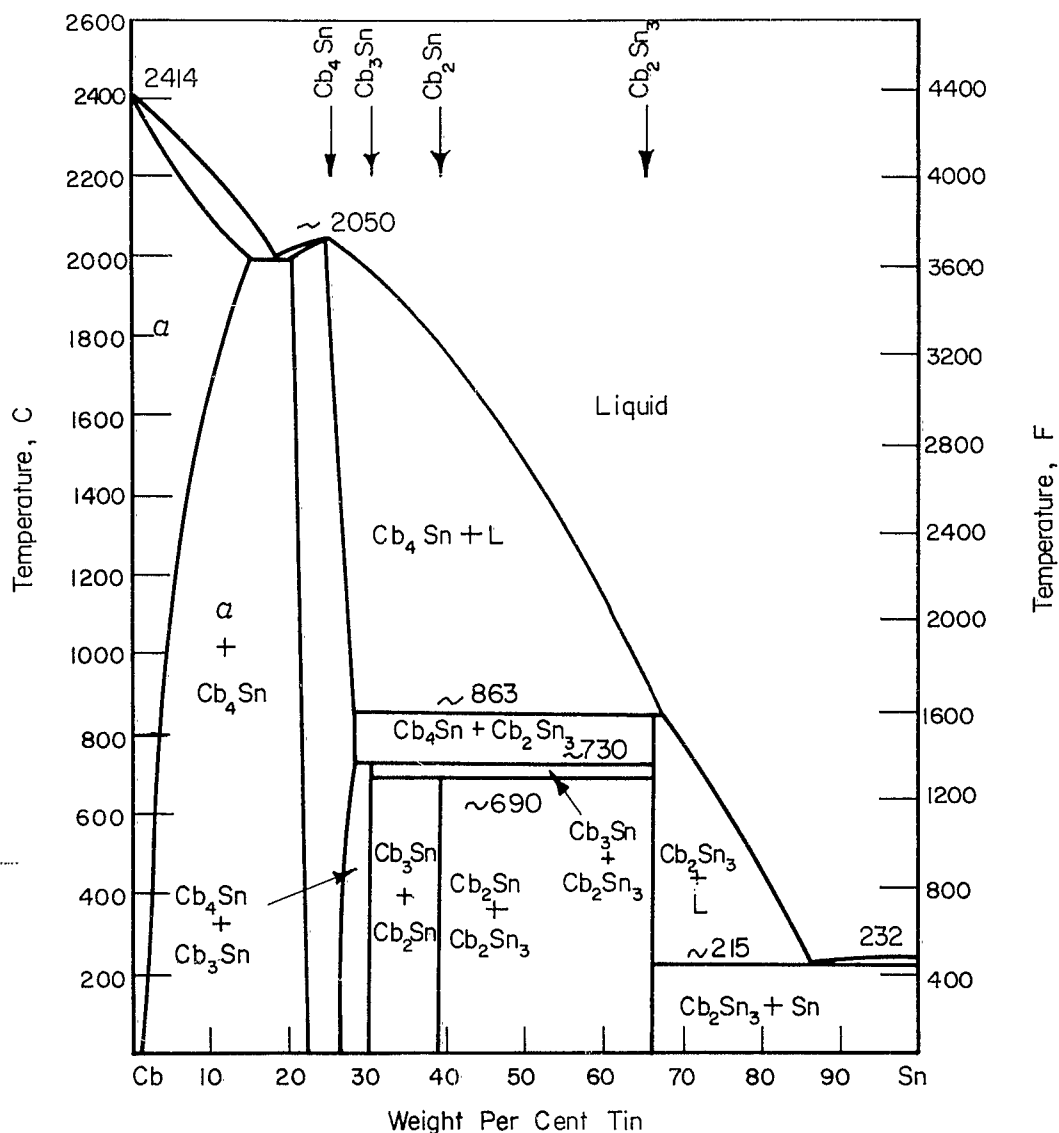
$PtCb_3$ has a β -tungsten type structure with its lattice parameters increasing from 5.137 to 5.156 kX with increasing columbium content. $PtCb_2$ has a tetragonal lattice and forms a eutectic with β at 55 atomic per cent platinum. Columbium is soluble in platinum up to 36 atomic per cent (21 weight per cent). The solubility limit of platinum in columbium is less than 1 atomic per cent. Pt_3Cb forms a superlattice near 1800 C.⁽²⁴⁵⁾

COLUMBIUM-TELLURIUM SYSTEM



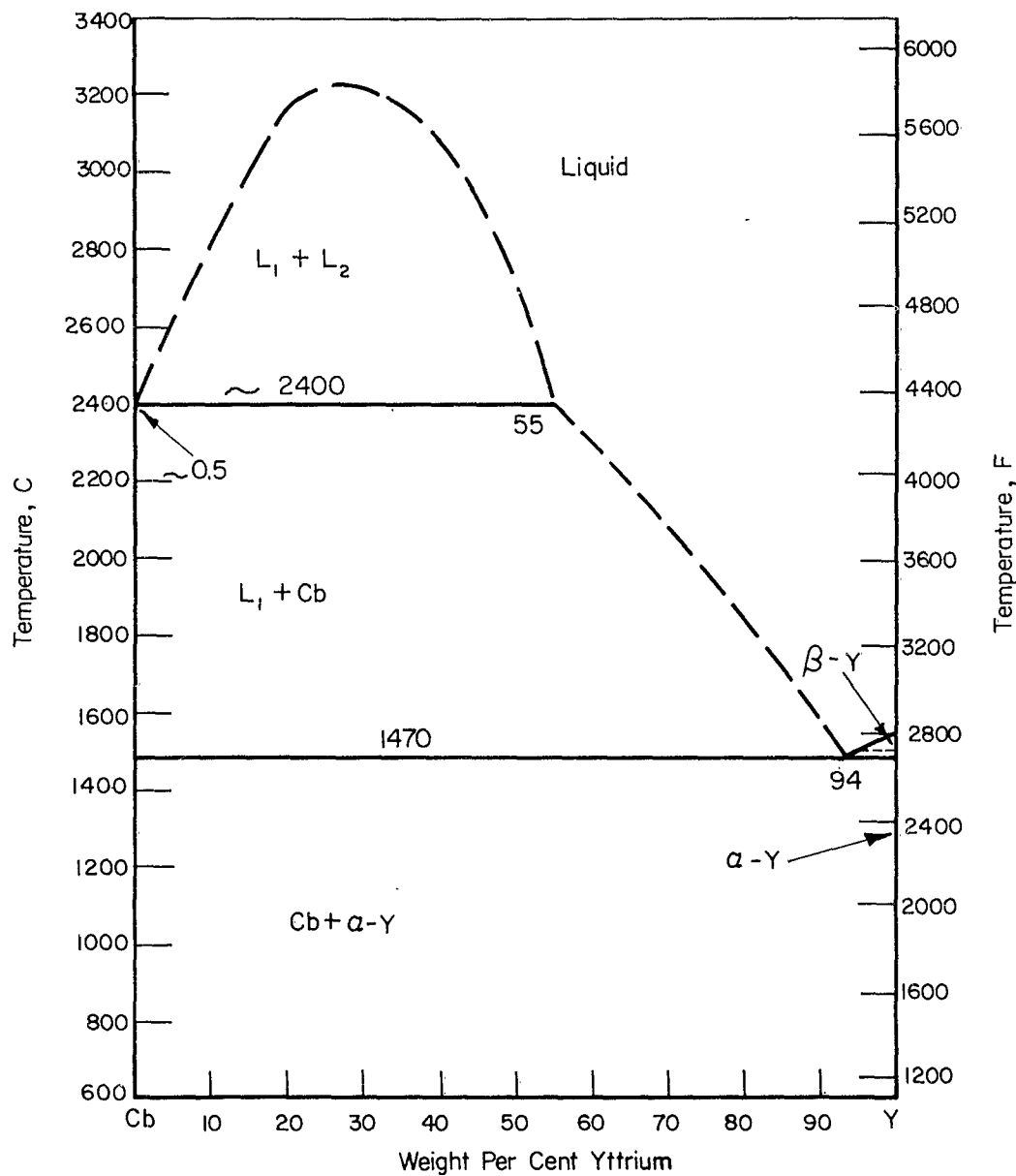
The above phase relationships are for 20 C.⁽²⁴⁶⁾

COLUMBIUM-TIN SYSTEM



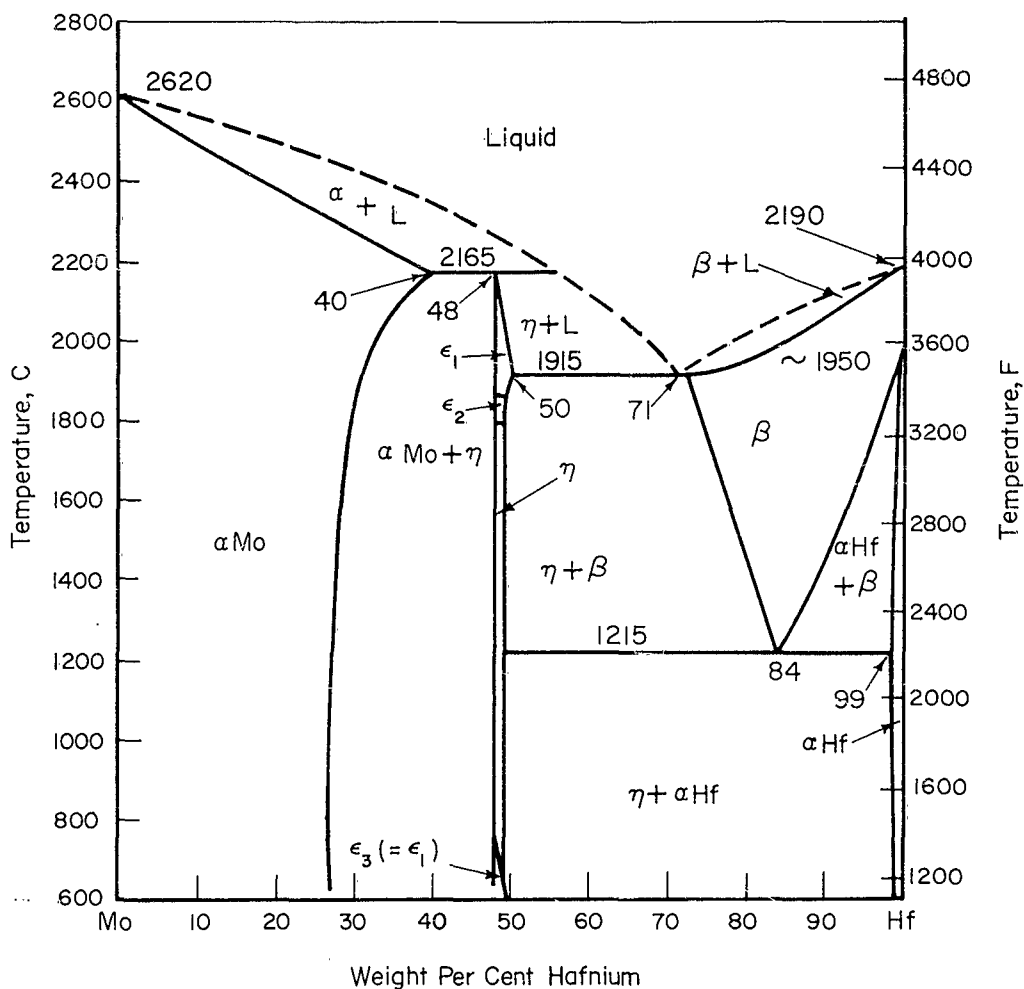
Cb₃Sn is a β -tungsten type structure with $a = 5.29 \text{ \AA}$.^(38,39) Agafonova, et al., determined the solid solubility of tin in columbium as 9.7 weight per cent at room temperature, increasing to 14 per cent at 2000 C. The solubility of columbium in tin is less than 0.1 weight per cent at the melting point of tin.⁽³⁸⁾ L. L. Wyman, et al., identified three additional phases in the system, Cb₄Sn, Cb₂Sn, and Cb₂Sn₃. A slightly higher solid solubility of tin in columbium was also indicated. The modified diagram was suggested by Wyman, et al.⁽³⁰⁴⁾

COLUMBIUM-YTTRIUM SYSTEM



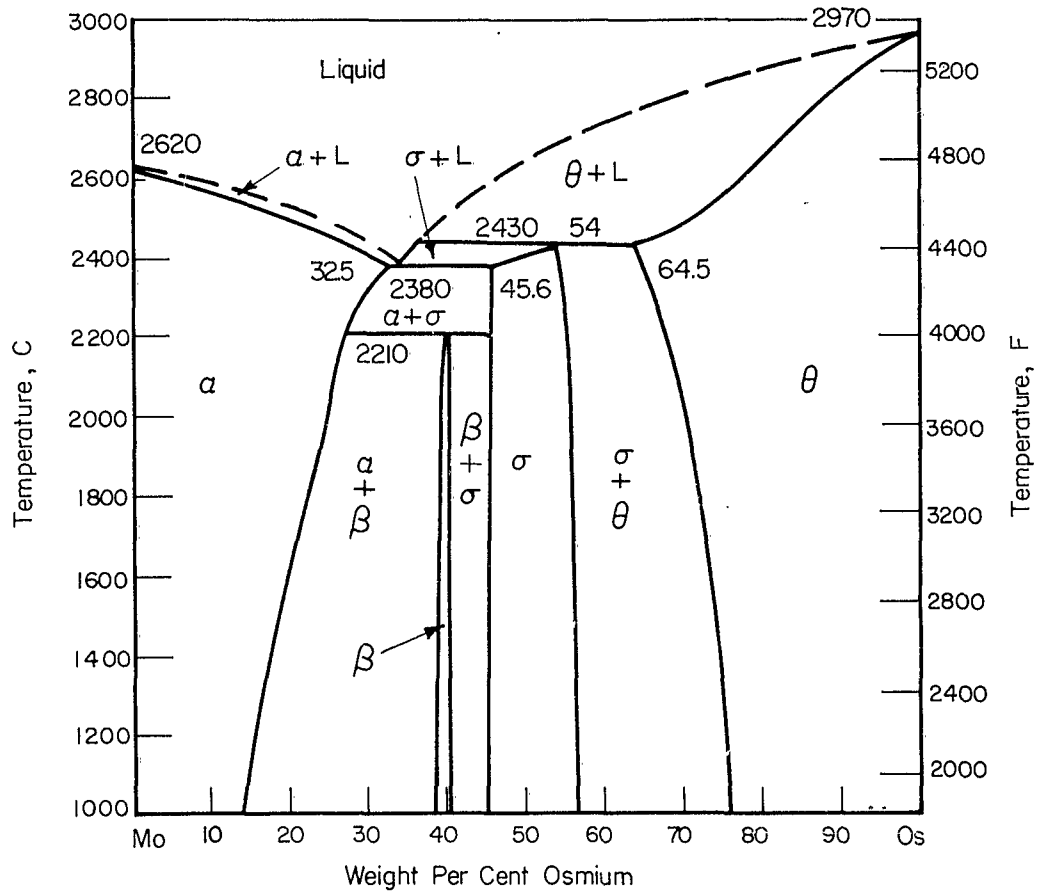
The maximum solubility of yttrium in columbium is less than 0.1 weight per cent, and the maximum solubility of columbium in yttrium is 0.2 ± 0.1 weight per cent.^(247, 248, 249) An inverse peritectic reaction in the yttrium-rich region is postulated for this system.⁽²⁴⁹⁾

MOLYBDENUM-HAFNIUM SYSTEM



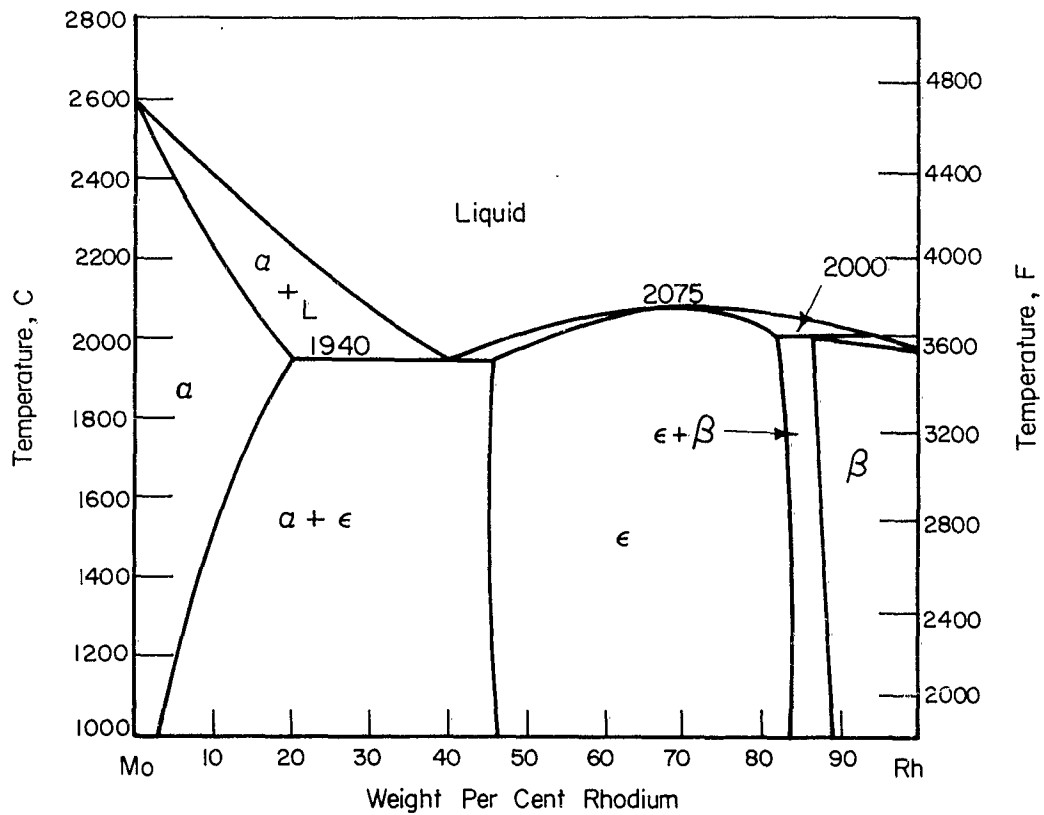
ϵ_1 (Mo_2Hf) is a hexagonal Laves phase of the C36-MgNi_2 type, with $a = 5.341 \text{ \AA}$, $c = 17.347 \text{ \AA}$, and $c/a = 3.248$. Between 1850 and 1816 C, ϵ_1 changes to ϵ_2 which is intermediate between the C36-MgNi_2 and C14-MgZn_2 structures; the lattice parameters are $a = 5.349 \text{ \AA}$, $c = 17.490 \text{ \AA}$, and $c/a = 3.270$. Upon annealing at 1752 C, ϵ_2 transforms to the cubic C15-MgCu_2 modification η with $a = 7.560 \text{ \AA}$. Annealing Mo_2Hf at 700 C for 2 weeks and quenching transforms the structure to the original C36-MgNi_2 structure of the ϵ_1 high-temperature phase. (78, 250)

MOLYBDENUM-OSMIUM SYSTEM



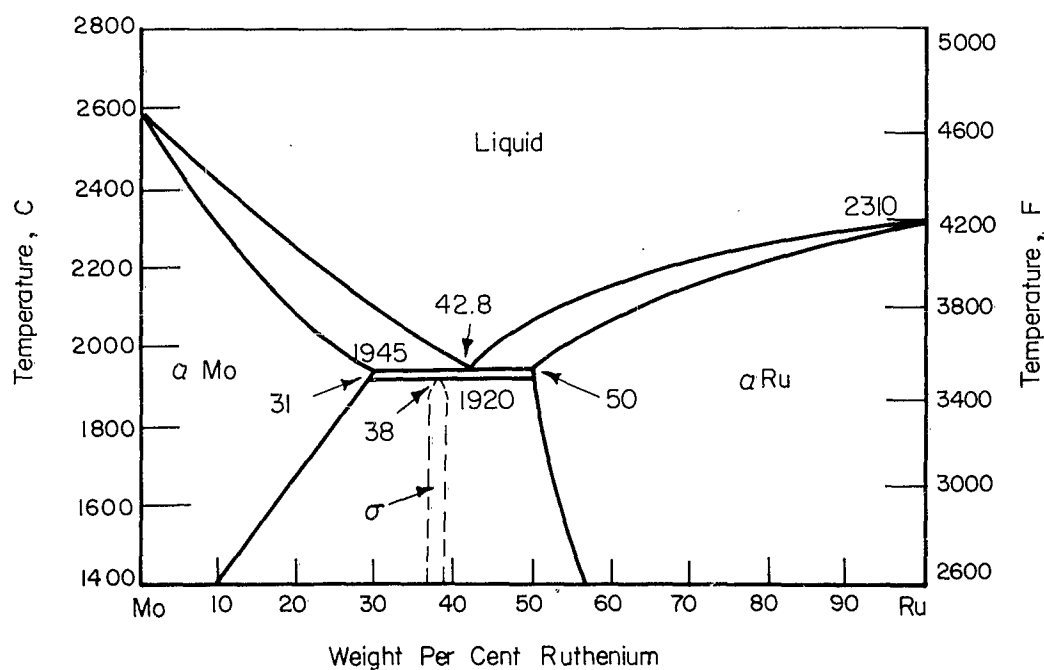
The β -Mo₃Os phase has a β -tungsten A15 type structure with the lattice parameters ranging from 4.971 Å at 24.5 atomic per cent osmium to 4.969 Å at 25 atomic per cent. The σ -phase is typical of σ -FeCe which has the atomic arrangement of tetragonal β -uranium; the lattice parameters decrease with the addition of osmium from $a = 9.632$ Å and $c = 4.950$ Å at 30 atomic per cent osmium (46 weight per cent) to $a = 9.613$ and $c = 4.934$ Å at 37.5 atomic per cent osmium (55 weight per cent). Osmium is soluble in molybdenum up to 19.5 atomic per cent (32.5 weight per cent) at 2380 C, decreasing to 7.0 atomic per cent (14 weight per cent) at 1000 C. Molybdenum dissolves in osmium up to 52 atomic per cent (35.5 weight per cent) at 2430 C, decreasing to 40 atomic per cent (25 weight per cent) at 1100 C.⁽²⁵⁰⁾

MOLYBDENUM-RHODIUM SYSTEM



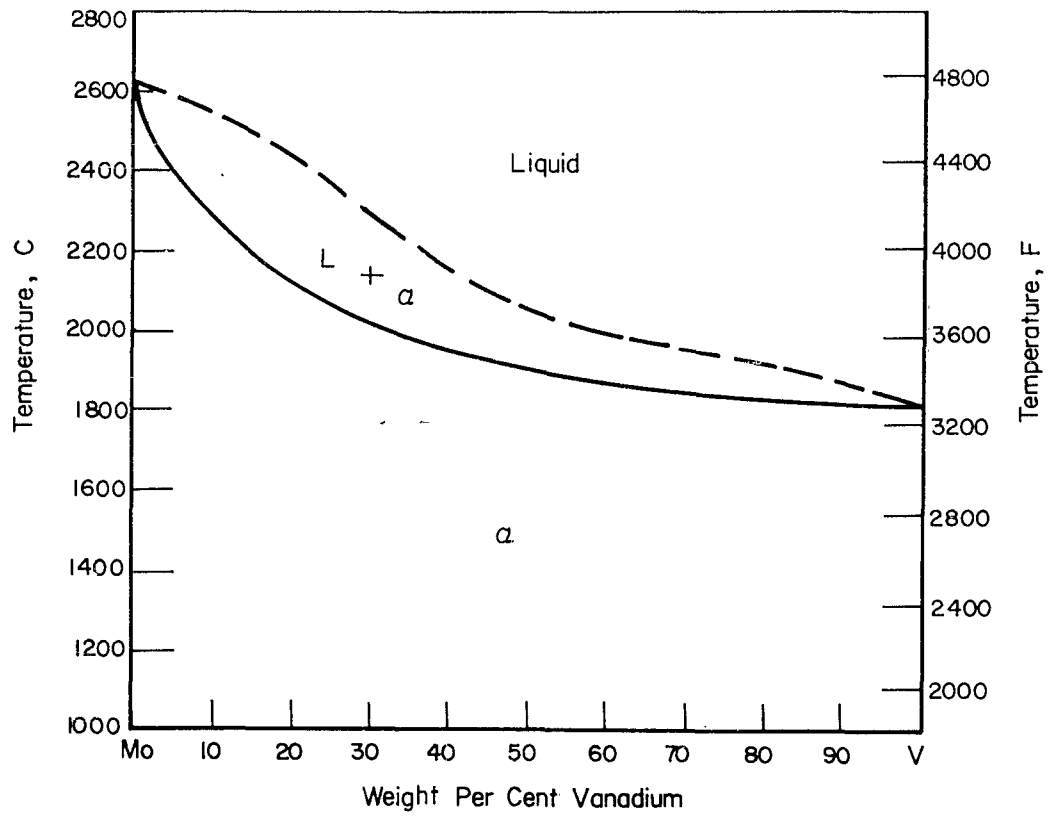
The ϵ -phase is close-packed hexagonal with $a = 2.740$ kX, $c = 4.380$ kX, and $c/a = 1.5999$ at 60 weight per cent rhodium.^(90,92) The maximum melting point for ϵ corresponds to the approximate composition MoRh_2 . Rhodium is soluble in molybdenum up to 21 weight per cent at 1940 C, decreasing to less than 3 weight per cent at 1100 C.⁽⁹⁰⁾ Rhodium can dissolve about 14 weight per cent (15 atomic per cent) molybdenum near 2000 C.⁽³⁰⁷⁾

MOLYBDENUM-RUTHENIUM SYSTEM



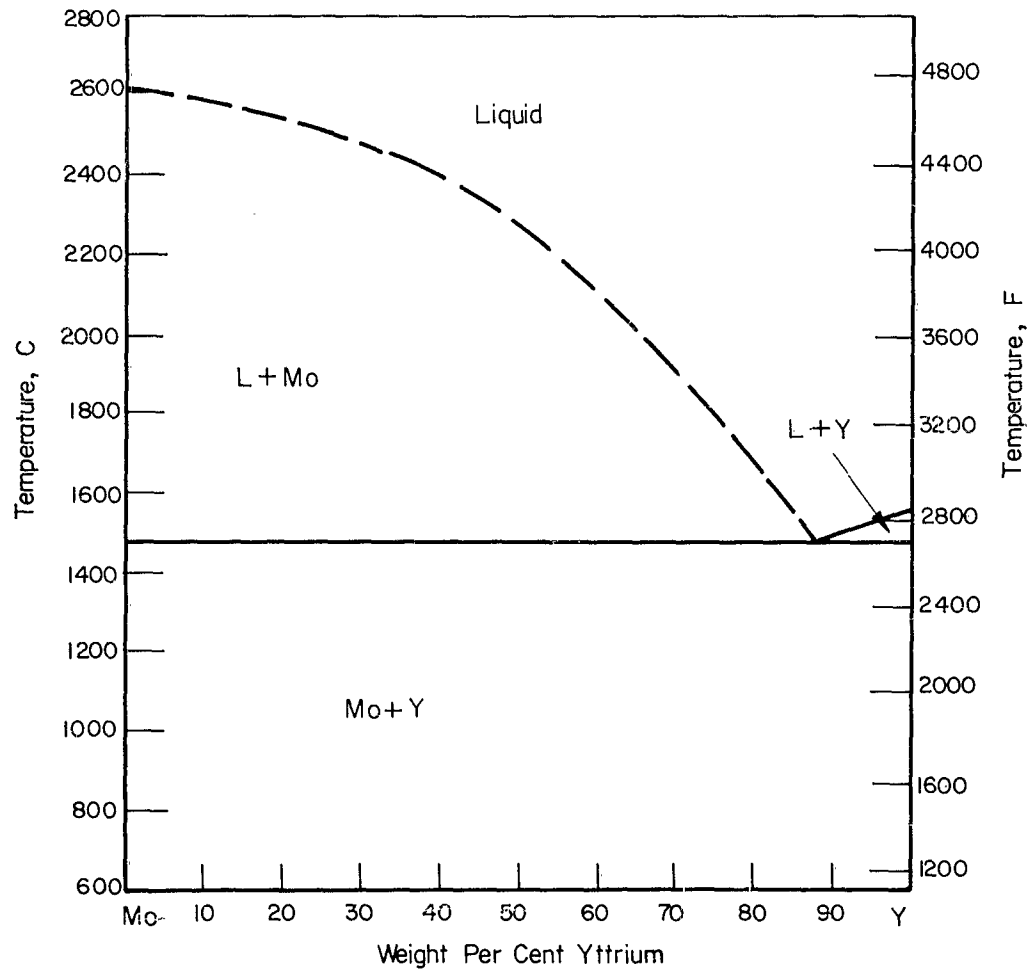
The σ -phase forms by a peritectoid reaction at 1920 C; the lattice parameters are $a = 9.538 \text{ kX}$, $c = 4.925 \text{ kX}$, and $c/a = 0.516$. The composition Mo_5Ru_3 is included in the composition range of the σ -phase. The solubility of ruthenium in molybdenum decreases rapidly from 31 weight per cent at 1945 C to 14 weight per cent at 1500 C. (251)

MOLYBDENUM-VANADIUM SYSTEM



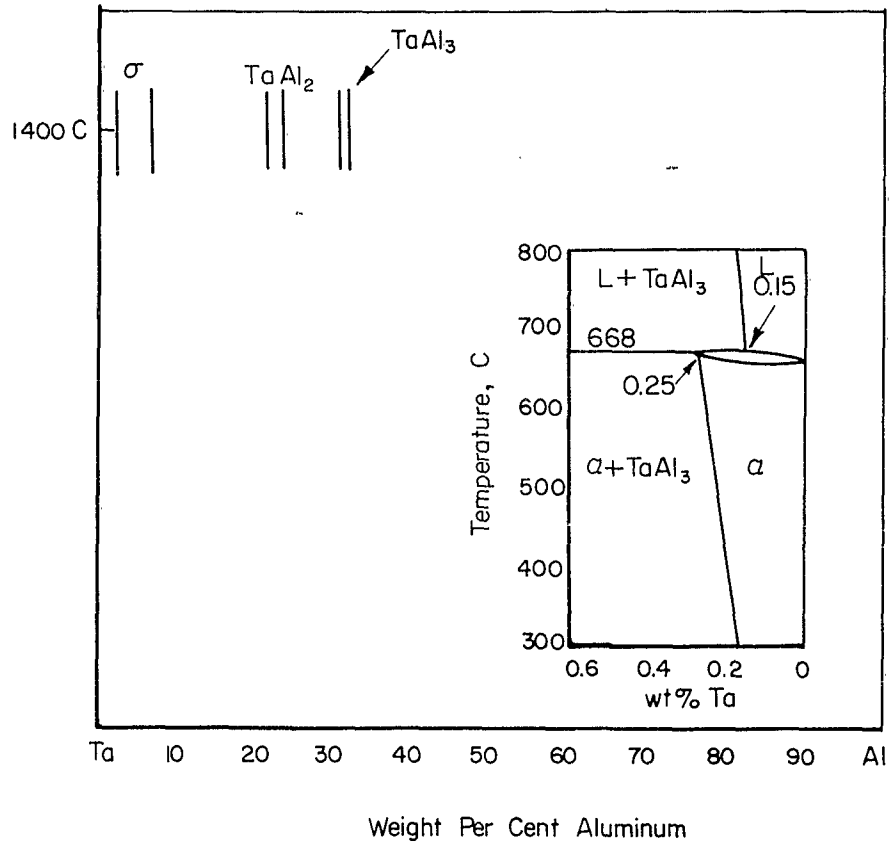
The molybdenum-vanadium system forms a continuous series of solid solutions.⁽³¹¹⁾

MOLYBDENUM-YTTRIUM SYSTEM



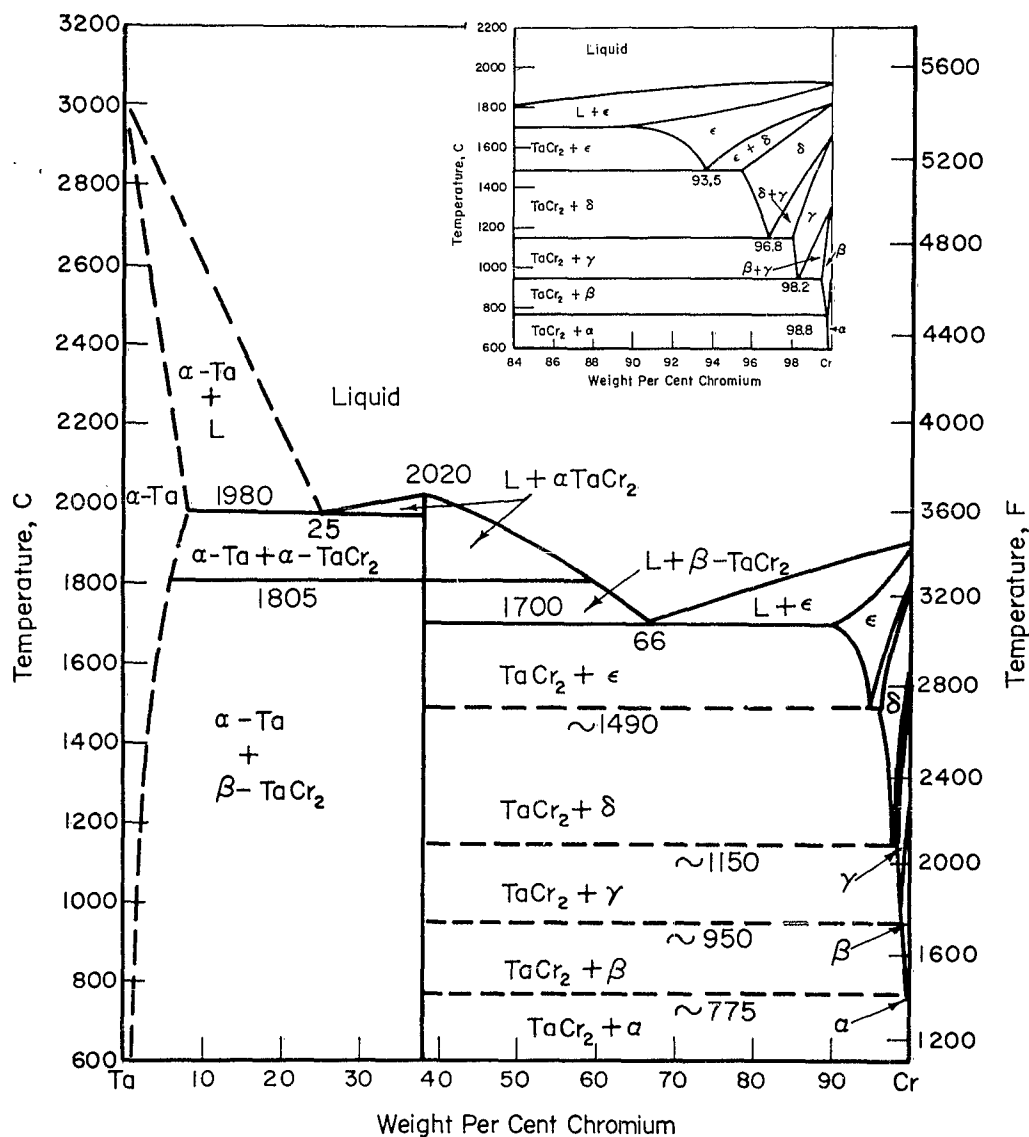
Terminal solubilities are probably less than 1 atomic per cent.^(247, 248)

TANTALUM-ALUMINUM SYSTEM



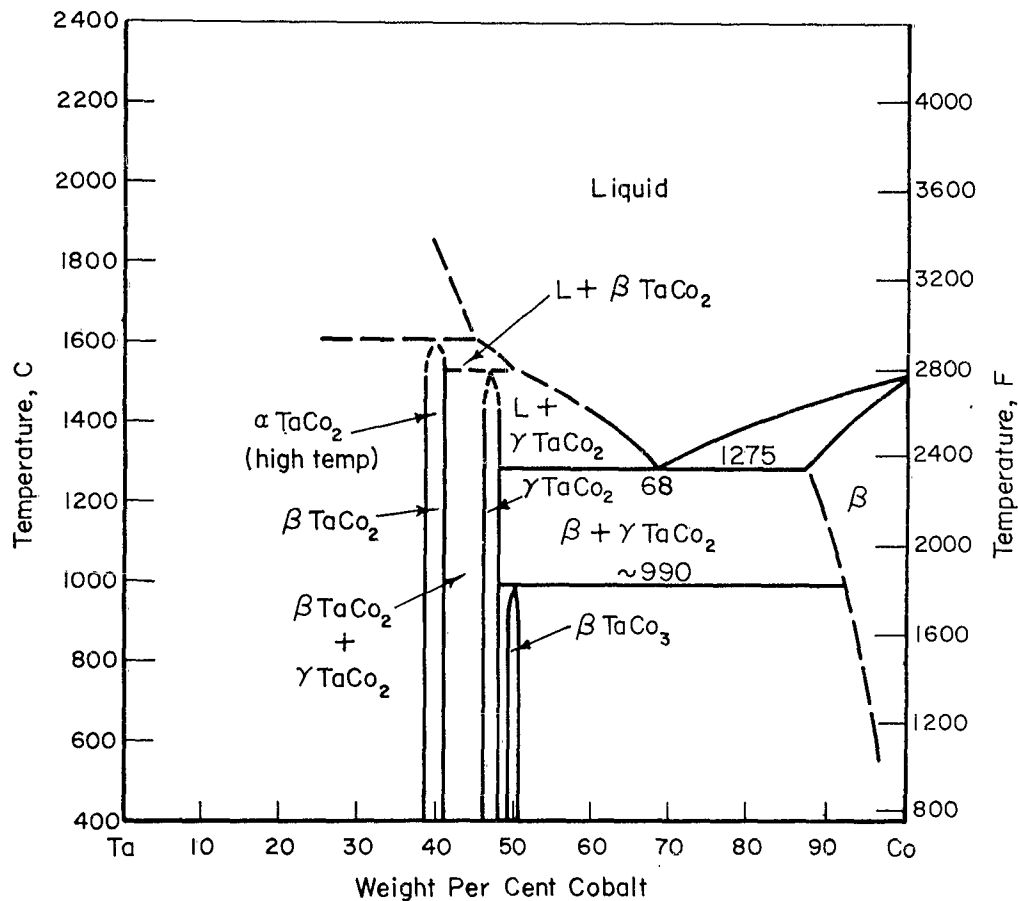
Three intermetallic compounds are present in the tantalum-aluminum system. $TaAl_3$ is isomorphous to $CbAl_3$, having a tetragonal structure with $a = 5.42 \text{ kX}$ and $c = 8.52 \text{ kX}$. A compound in the vicinity of 65 atomic per cent aluminum, possibly $TaAl_2$, exhibits a structure similar to the $ZrAl_3$ or $ZrSi_2$.⁽²⁶⁶⁾ Ta_2Al is a tetragonal σ -type phase with $a = 9.828 \text{ \AA}$, $c = 5.232 \text{ \AA}$, and $c/a = 0.532$.⁽²⁶⁷⁾ The above intermediate phase boundaries are for 1400 C.⁽²⁶⁶⁾ The solubility of tantalum in aluminum was determined by Glazov.⁽³¹⁰⁾

TANTALUM-CHROMIUM SYSTEM



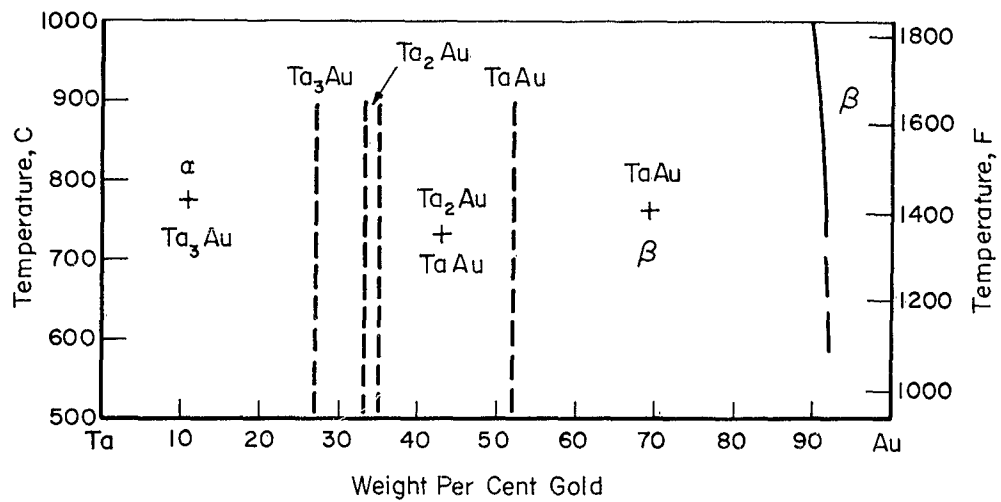
Duwez and Martens⁽¹²³⁾ reported that TaCr₂ has a polymorphic transformation between 1375 and 1590 C from the low-temperature cubic structure (MgCu₂-type with $a = 6.961$ Å) to the high-temperature hexagonal structure ($a = 4.925$ Å, $c = 8.062$ Å, $c/a = 1.637$). Elliott considers the compound isomorphous with MgZn₂ at all temperatures from 600 to 1200 C.⁽¹²⁴⁾ Grigor'ev et al.⁽²³³⁾ states that TaCr₂ undergoes a polymorphic transformation at 1805 C. They report that the solubility of chromium in tantalum is about 5 weight per cent at 1800 C, and the solubility of tantalum in chromium is about 10 weight per cent at 1700 C. Grigor'ev,⁽³⁰⁰⁾ et al., reported five structural modifications in the chromium-rich region. The ϵ solid solution has a body-centered cubic lattice whose parameters can be extrapolated to α -chromium. Alloys in the region of the γ solid solution have a similar lattice. Alloys with 5 weight per cent tantalum quenched from 1550 C contained body-centered cubic ϵ and a second phase δ which has a hexagonal lattice with $a = 2.841$ kX, $c = 4.786$ kX, and $c/a = 1.685$.

TANTALUM-COBALT SYSTEM



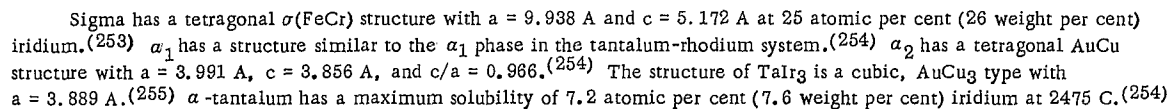
Two Laves phases were identified in a 39.4 weight per cent cobalt alloy.⁽³⁰⁵⁾ α -TaCo₂, a hexagonal MgZn₂-type structure with $a = 4.797$ Å, $c = 7.827$ Å, and $c/a = 1.632$ was present in the as-cast microstructure. Aging at 1100 and 1200 C transformed α -TaCo₂ to a cubic MgCu₂-type phase β -TaCo₂ with $a = 6.778$. Elliot also observed both structures.⁽¹²⁴⁾ γ -TaCo₂ is also a Laves phase of the hexagonal MgNi₂ type with $a = 4.700$ Å, $c = 15.42$ Å, and $c/a = 3.281$.^(120, 305) β -TaCo₃ is hexagonal with $a = 9.411$ Å, $c = 15.50$ Å, and $c/a = 1.647$.⁽³⁰⁵⁾ A metastable ordered cubic phase ($a = 3.647$ Å) of the AuCu₃ type was also reported.⁽³⁰⁵⁾ Contributions to the above diagram were made by Köster and Mulfinger⁽¹²¹⁾ and by Hoschimoto⁽¹²²⁾.

TANTALUM-GOLD SYSTEM

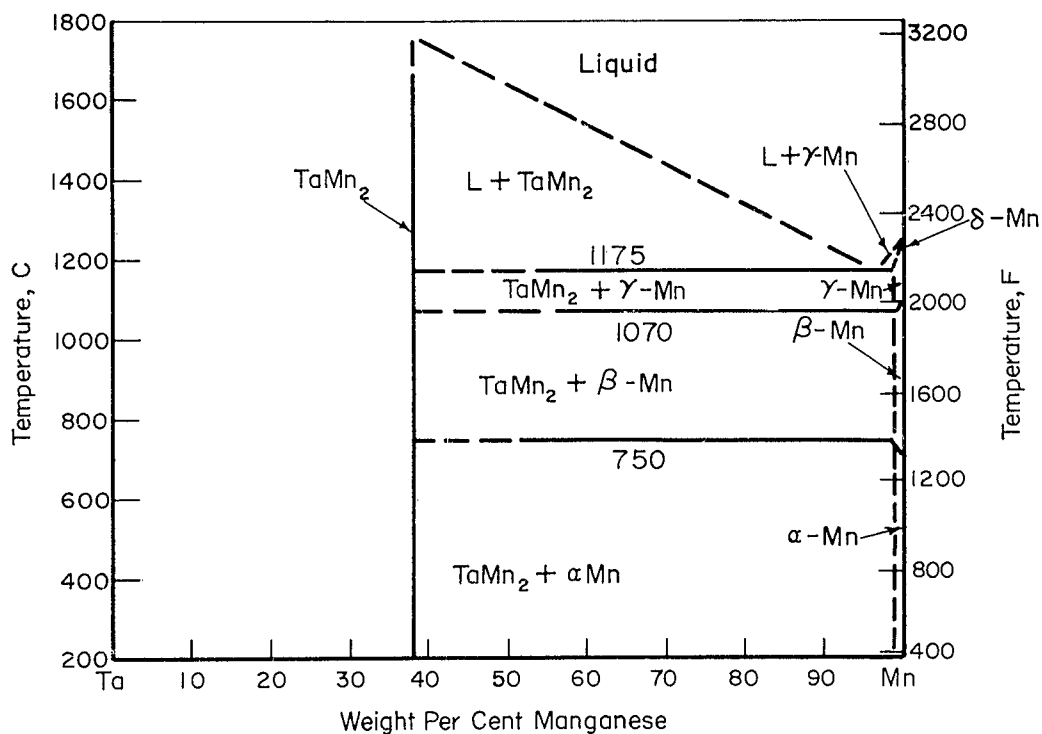


Ta_3Au has a cubic Cr_3O -type structure with $a = 5.21$ kX. Ta_2Au is a tetragonal σ -type phase with $a = 10.04$ kX and $c/a = 0.520$. $TaAu$ has a tetragonal structure with $a = 3.37$ kX and $c/a = 0.902$. The solubility of tantalum in gold is 10 weight per cent at 1000 C, decreasing to 8.7 weight per cent at 800 C. (252)

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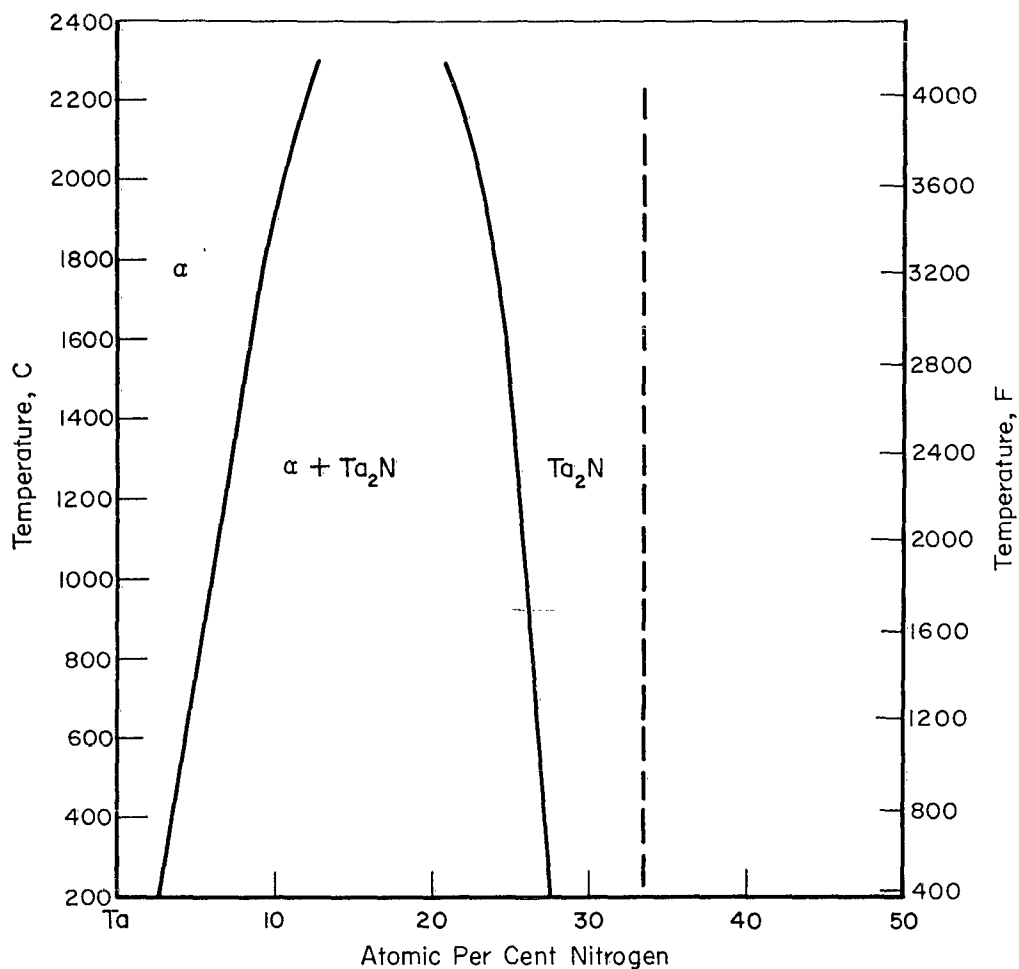


TANTALUM-MANGANESE SYSTEM



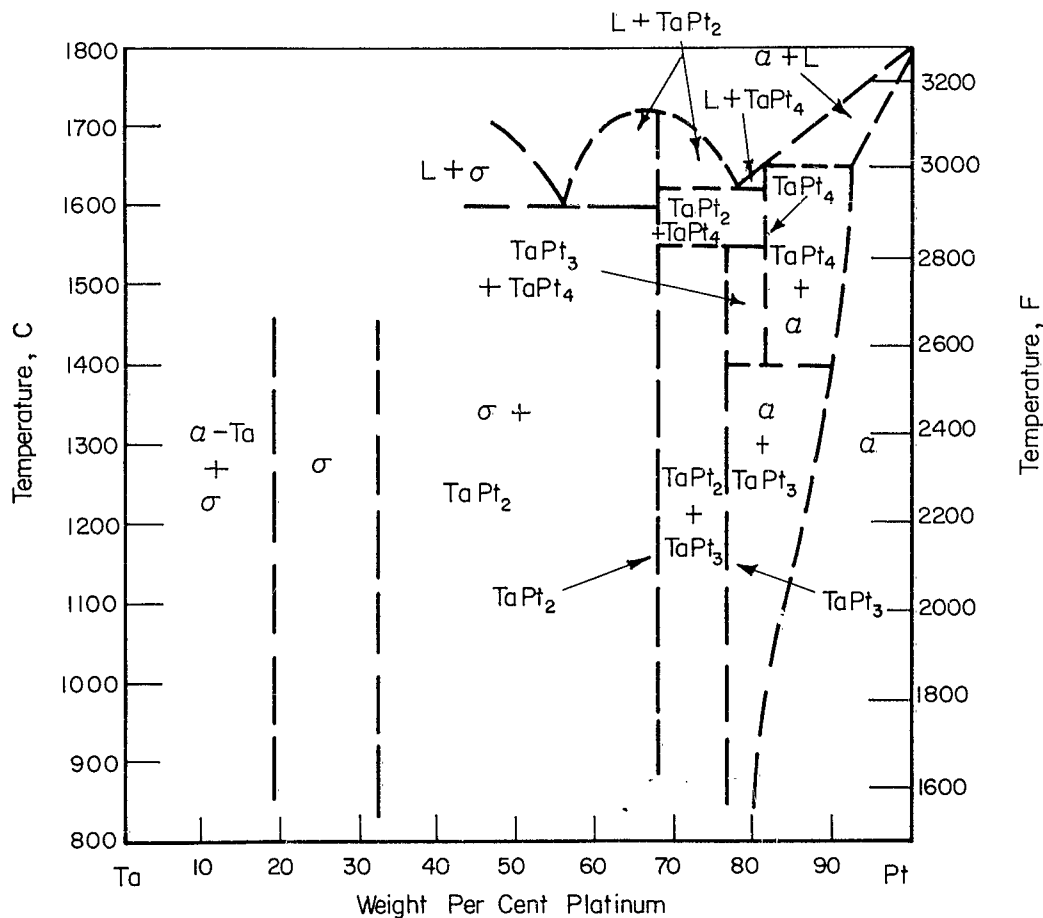
TaMn₂ has a hexagonal MgZn₂ Laves-phase-type structure with $a = 4.864 \text{ \AA}$, $c = 7.947 \text{ \AA}$, and $c/a = 1.634$.⁽¹²⁴⁾ This compound has a melting point above 1670 C.⁽²⁹⁸⁾ At 1070 C, the $\beta \rightleftharpoons \gamma$ transition occurs by a eutectoid reaction. The δ -manganese $\rightleftharpoons \gamma$ -manganese transformation is a peritectic reaction and occurs near the melting point.⁽²⁹⁸⁾

TANTALUM-NITROGEN SYSTEM



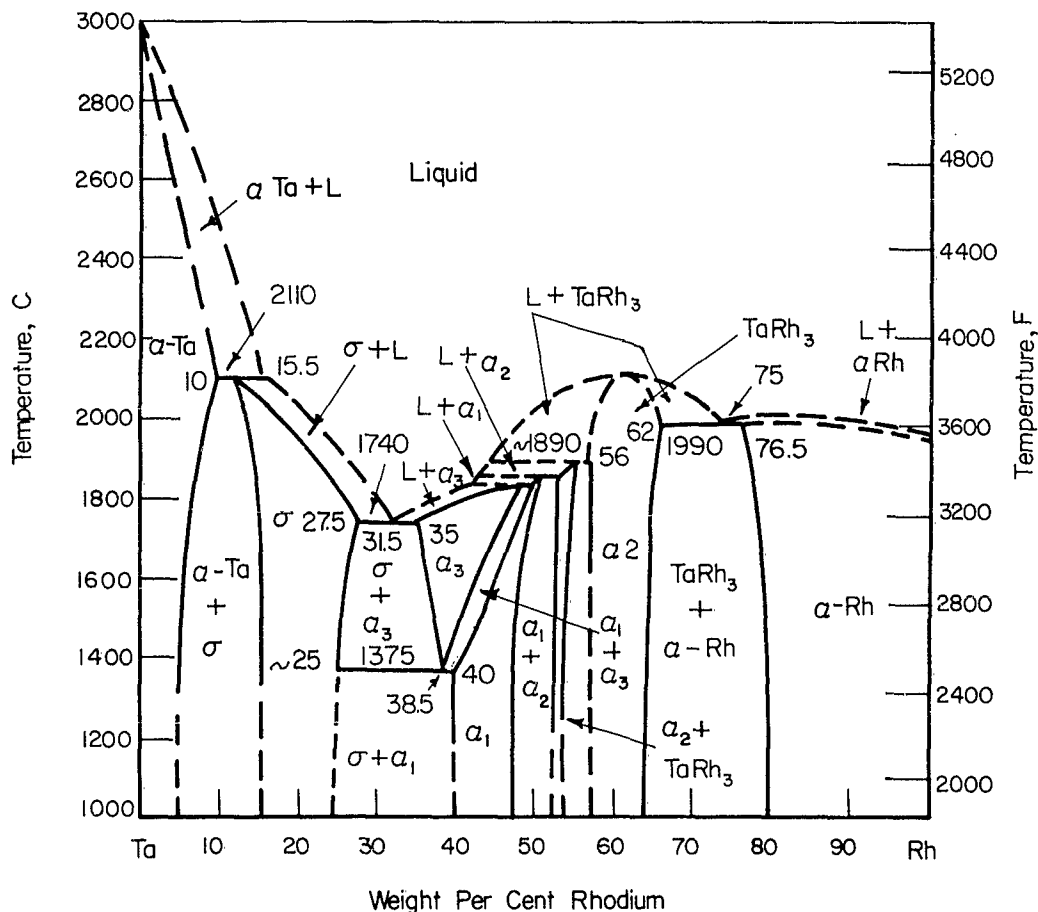
Two intermetallic compounds are definitely established for the tantalum-nitrogen system. TaN is close-packed hexagonal with $a = 5.181 \text{ kX}$, $c = 2.905 \text{ kX}$, and $c/a = 0.561$.⁽¹³²⁾ Ta_2N is also close-packed hexagonal with $a = 3.042 \text{ kX}$, $c = 4.909 \text{ kX}$, and $c/a = 1.614$.^(132, 133) The melting point of TaN has been given as 2890°C ⁽¹³⁴⁾ and 3090°C ⁽¹³⁵⁾. Chiotti has shown that TaN dissociates at high temperatures, forming the lower nitride, Ta_2N , and nitrogen.⁽¹³⁶⁾ The phase diagram was obtained from Reference (256).

TANTALUM-PLATINUM SYSTEM



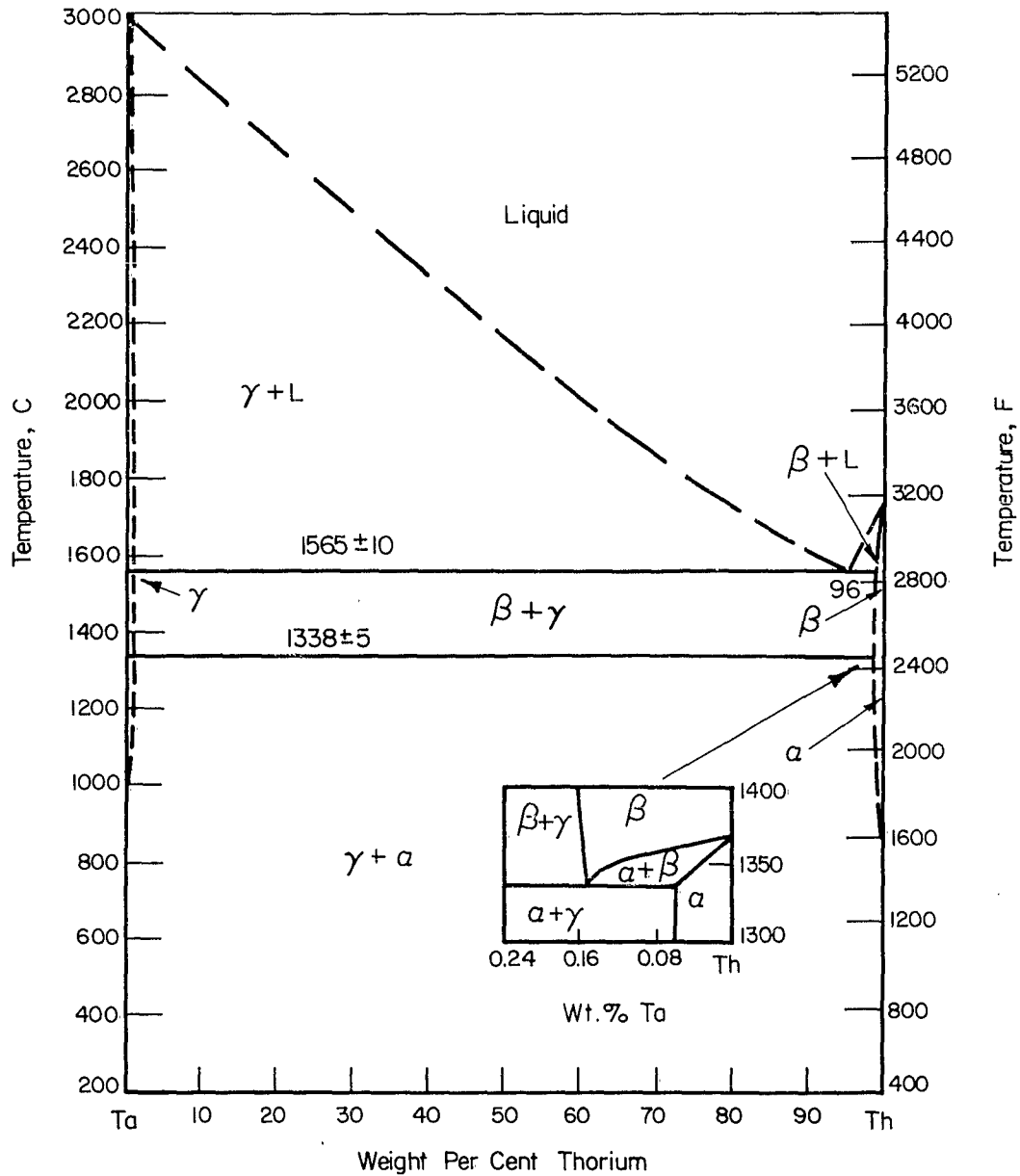
Four intermediate phases were observed between 50 and 100 per cent platinum. The σ -phase has a complex tetragonal structure with 30 atoms per unit cell;⁽²⁵⁷⁾ cell dimensions are $a = 9.95 \text{ \AA}$, $c = 5.16 \text{ \AA}$, and $c/a = 0.52$.⁽³⁰⁾ TaPt_4 is stable above 1000 C and has a tetragonal structure with $a = 8.58 \text{ \AA}$, $c = 10.60 \text{ \AA}$, and $c/a = 1.24$. TaPt_3 is tetragonal with $a = 6.45 \text{ \AA}$, $c = 6.98 \text{ \AA}$, and $c/a = 1.08$. The TaPt_2 phase was not clearly established by the X-ray results. The solubility of tantalum in platinum is about 10 atomic per cent (9 weight per cent) at 1500 C and 20 atomic per cent (19 weight per cent) at 1000 C. The exact phase boundaries and reaction isotherms were not determined in this investigation by Browning.⁽²⁵⁷⁾

TANTALUM-RHODIUM SYSTEM



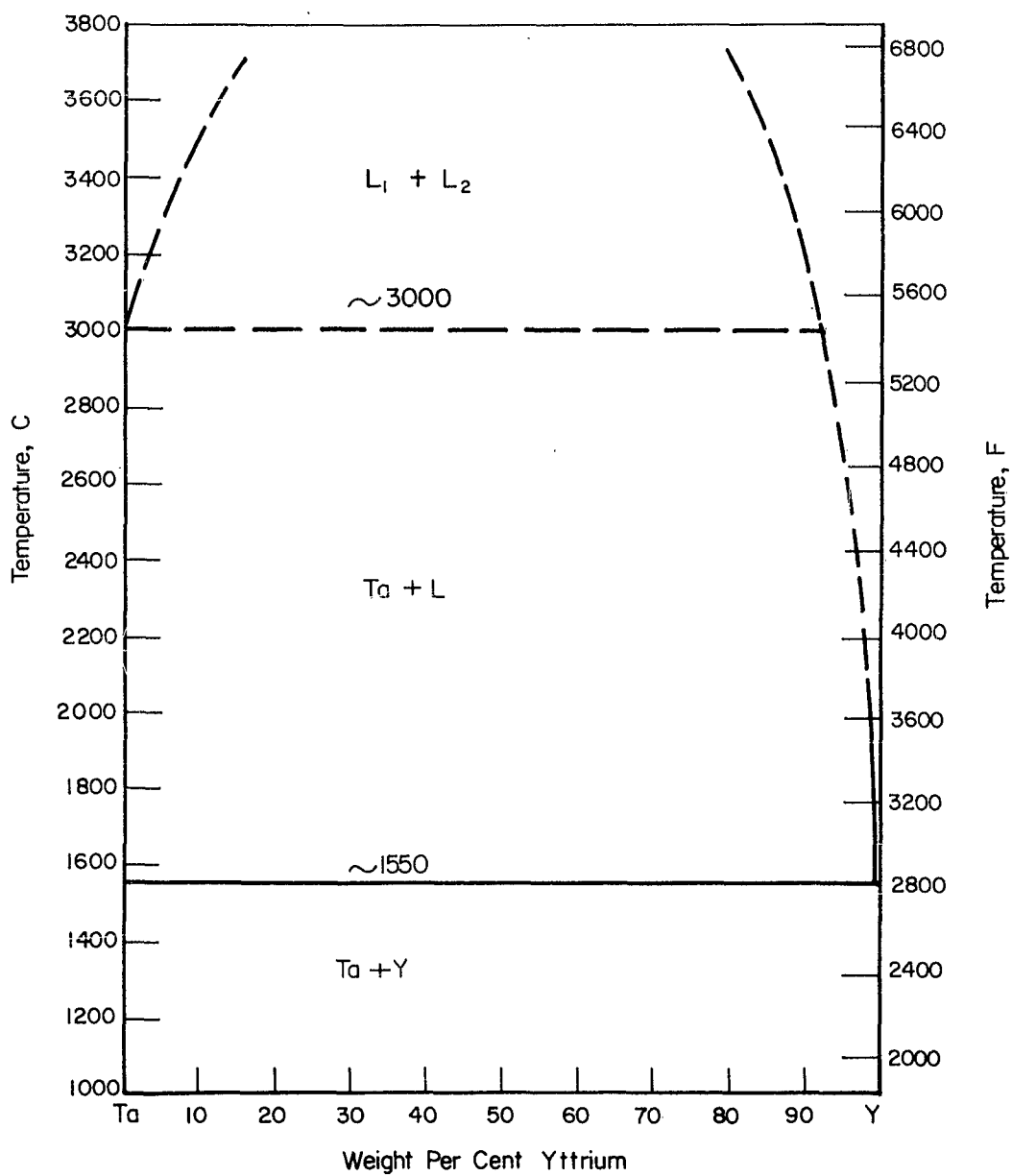
Five intermediate phases occur in the system. Sigma has a tetragonal $\sigma(\text{FeCr})$ type structure with $a = 9.754 \text{ \AA}$, $c = 5.058 \text{ \AA}$, and $c/a = 0.518$ at the rhodium-rich side.⁽⁹¹⁾ α_1 is orthorhombic, similar to VCo_3 , with $a = 5.62 \text{ \AA}$, $b = 9.48 \text{ \AA}$, and $c = 13.61 \text{ \AA}$.⁽²⁵⁴⁾ α_2 is orthorhombic and probably isomorphous with Co_2Si with $a = 5.45 \text{ \AA}$, $b = 8.15 \text{ \AA}$, and $c = 4.01 \text{ \AA}$.⁽²⁵⁴⁾ The structure of α_3 is unknown.⁽²⁵⁴⁾ TaRh_3 has a cubic AuCu_3 -type structure with $a = 3.86 \text{ \AA}$.⁽²⁵⁵⁾ The maximum solubility of rhodium in tantalum is 16 ± 1 atomic per cent (10 weight per cent) at 2110 C; tantalum is soluble in rhodium up to 15 ± 0.5 atomic per cent (23.5 weight per cent) at 1990 C.⁽²⁵⁴⁾

TANTALUM-THORIUM SYSTEM



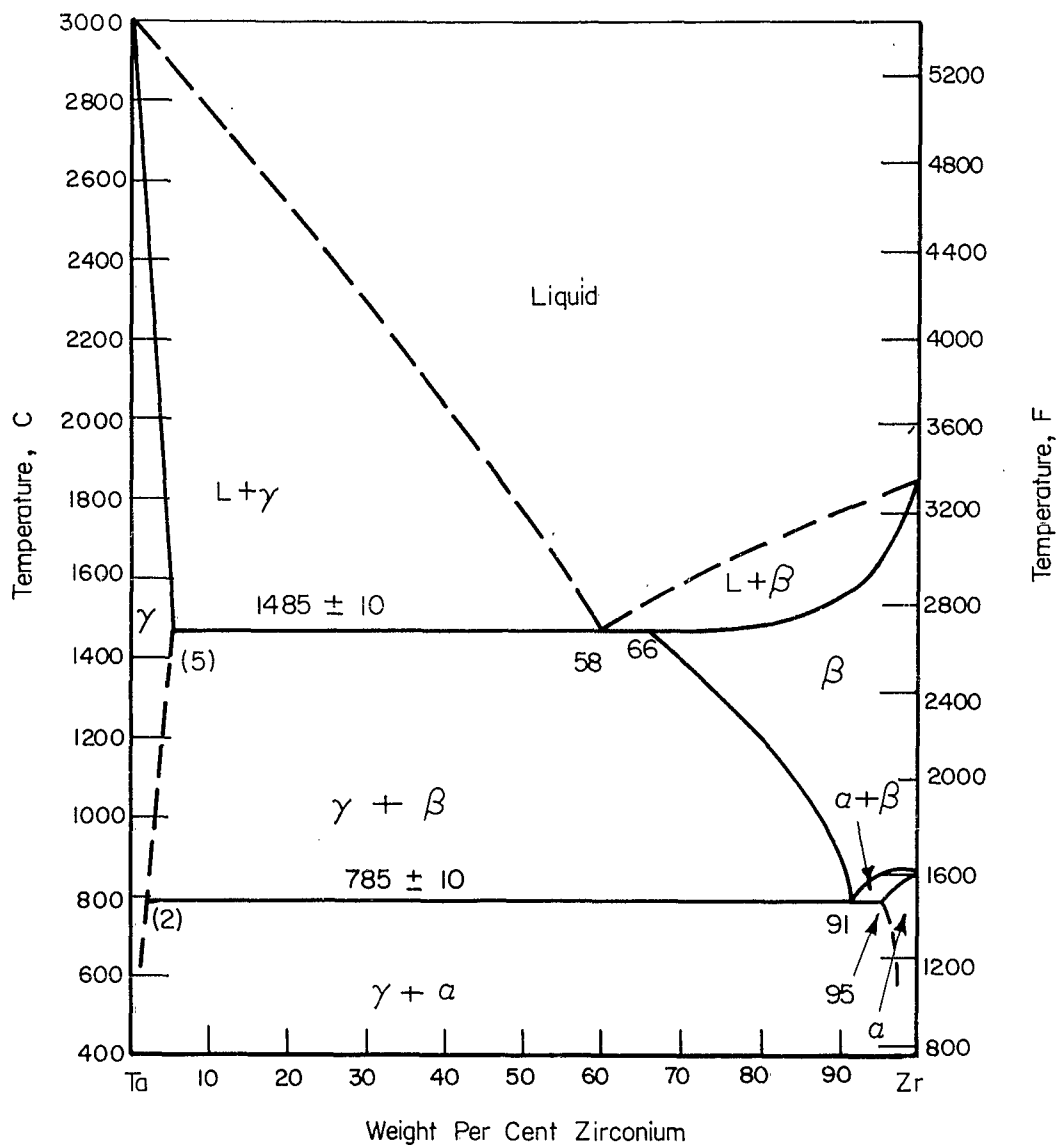
The solubility of thorium in tantalum is less than 0.2 weight per cent at the eutectic temperature. The solubility of tantalum in thorium at the eutectic temperature is about 0.4 weight per cent, and below 1340 C, is less than 0.2 weight per cent. No evidence of intermediate phases was found.⁽³⁰⁶⁾

TANTALUM-YTTRIUM SYSTEM



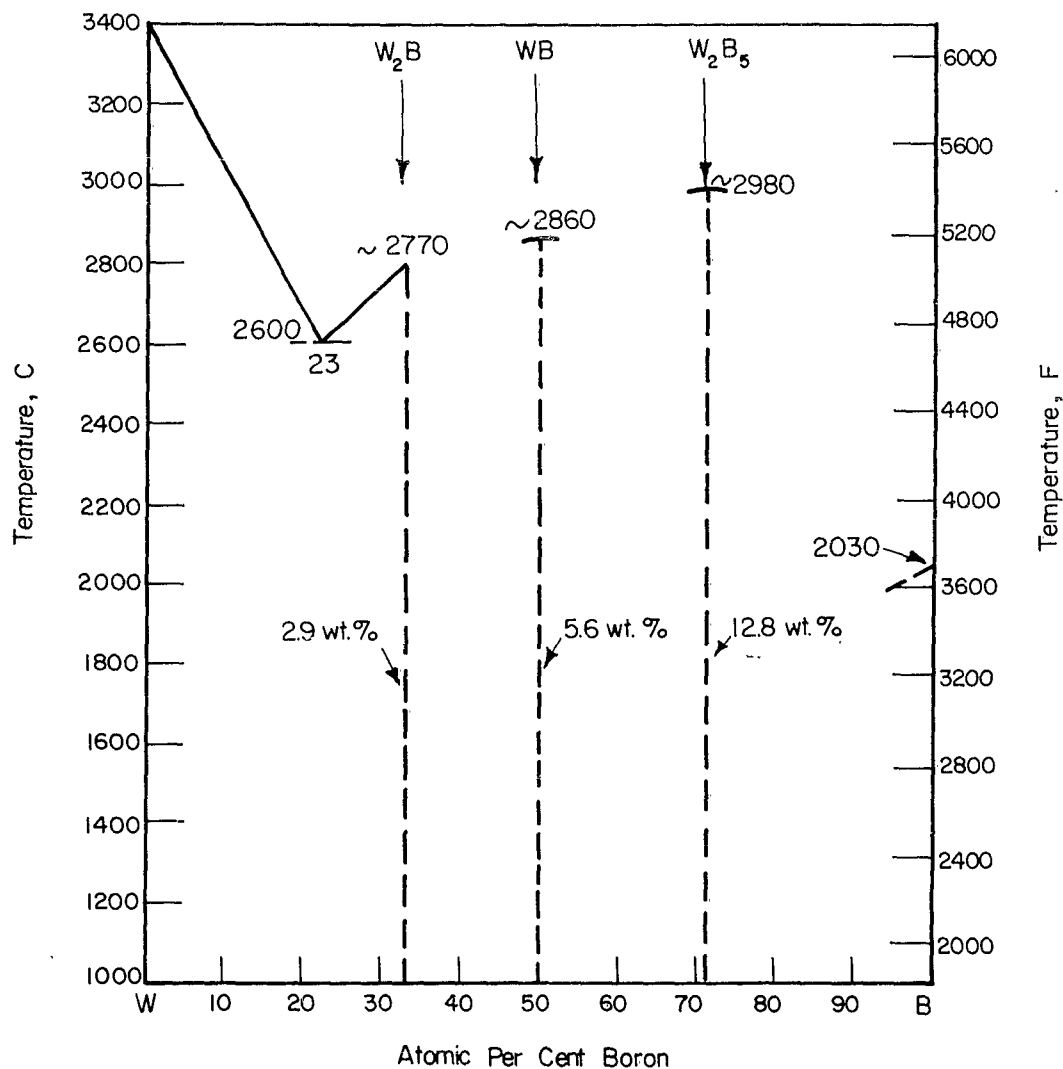
Terminal solubilities are about 0.1 weight per cent. (248, 249)

TANTALUM-ZIRCONIUM SYSTEM



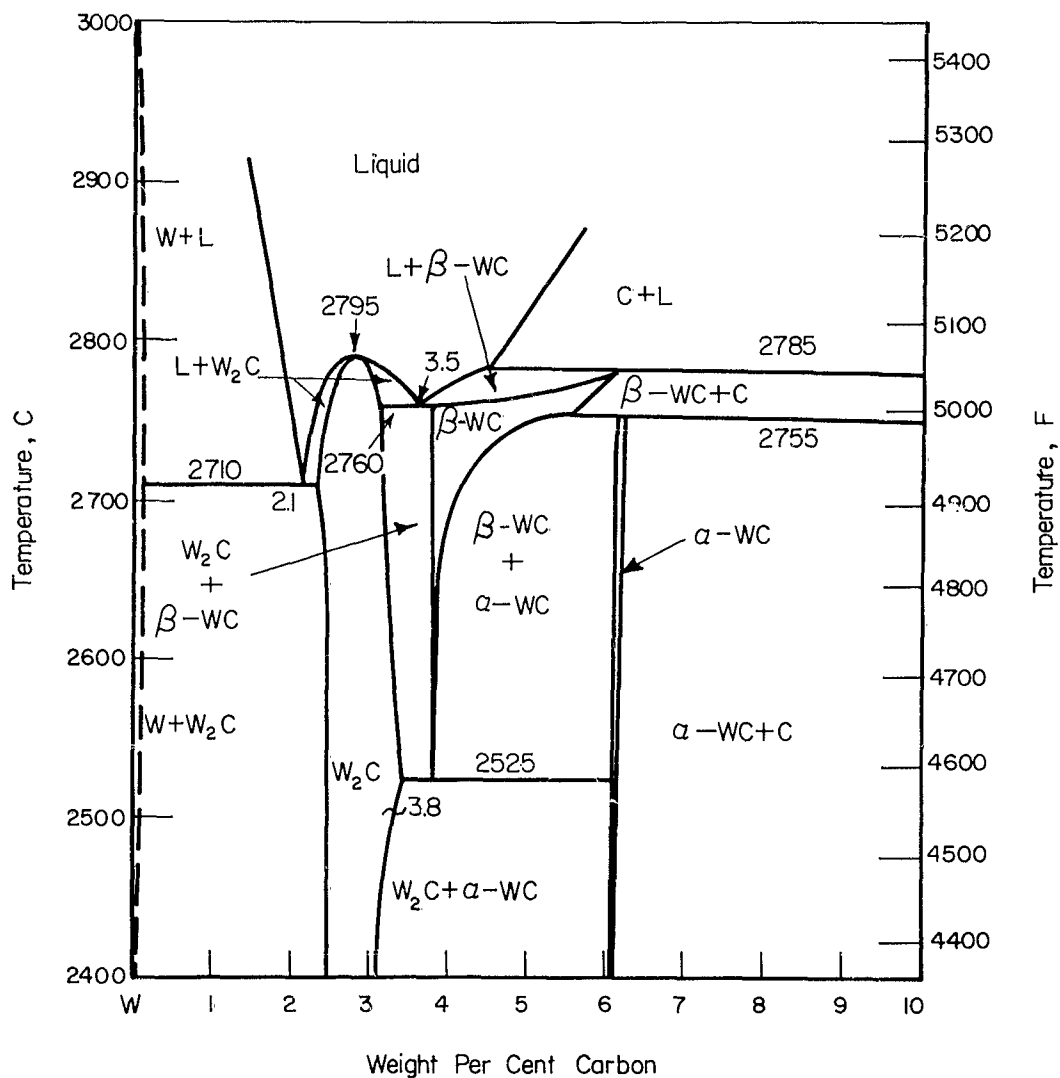
Pease, et al.,⁽²⁵⁸⁾ locate the eutectic temperature at 1485 C with the eutectic composition at 58 weight per cent (73 atomic per cent) zirconium. The eutectoid composition was given as 91 weight per cent (95.5 \pm 1 atomic per cent) zirconium at 785 \pm 10 C, the temperature agreeing with Emilyanov, et al.⁽¹⁶⁰⁾ Possible oxygen contamination could have affected Pease's results.

TUNGSTEN-BORON SYSTEM



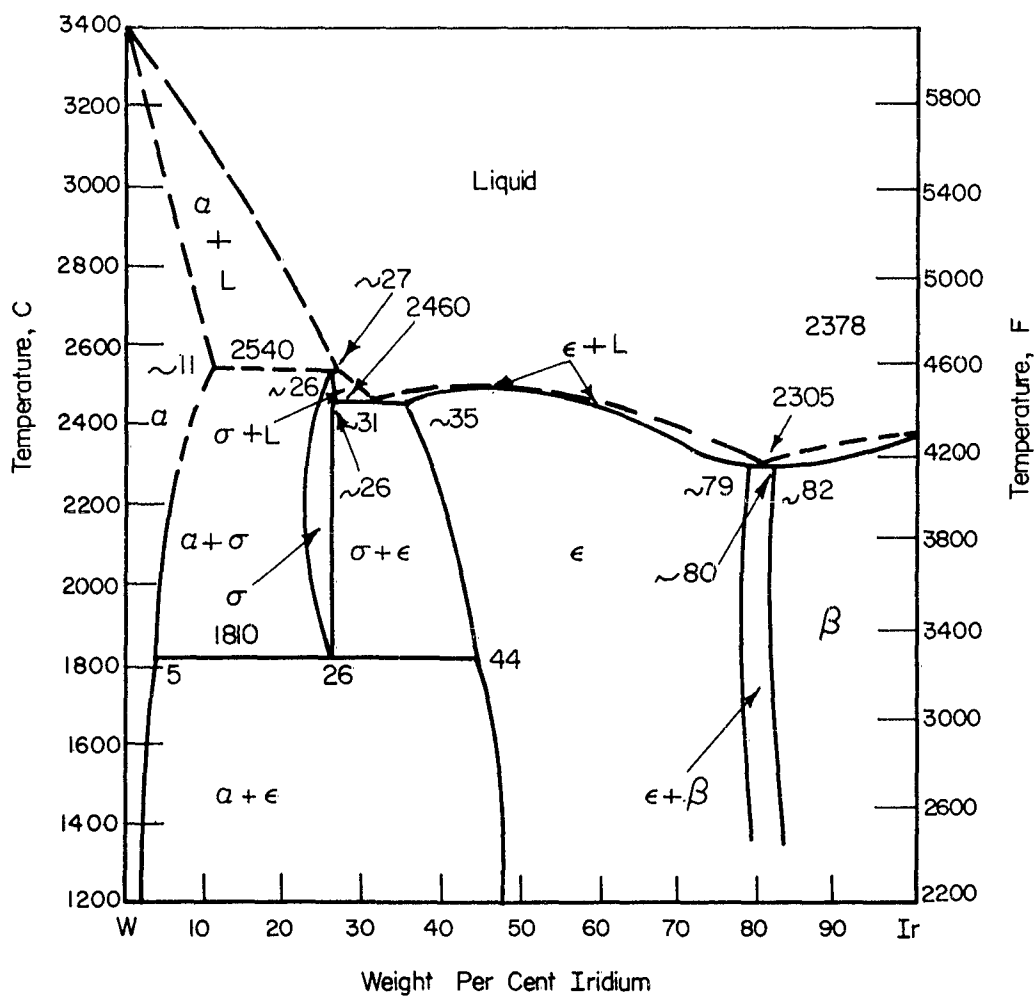
W₂B is tetragonal of the CuAl₂ (C16) type with $a = 5.564 \text{ \AA}$, $c = 4.740 \text{ \AA}$, and $c/a = 0.852$.⁽⁵⁸⁾ A low-temperature form of WB, stable below 1850 C, is tetragonal (MoB type) with $a = 3.115 \text{ \AA}$, $c = 16.93 \text{ \AA}$, and $c/a = 5.44$.⁽⁵⁸⁾ The high-temperature modification, corresponding to β -MoB, is orthorhombic (CrB type) with $a = 3.19 \text{ \AA}$, $b = 8.40 \text{ \AA}$, $c = 3.07 \text{ \AA}$.⁽¹⁶⁴⁾ W₂B₅ has a hexagonal defect structure with $a = 2.982 \text{ \AA}$, $c = 13.87 \text{ \AA}$, and $c/a = 4.65$.⁽¹⁶⁴⁾ The W-W₂B eutectic temperature is 2600 C at 23 atomic per cent boron.⁽²⁵⁹⁾

TUNGSTEN-CARBON SYSTEM



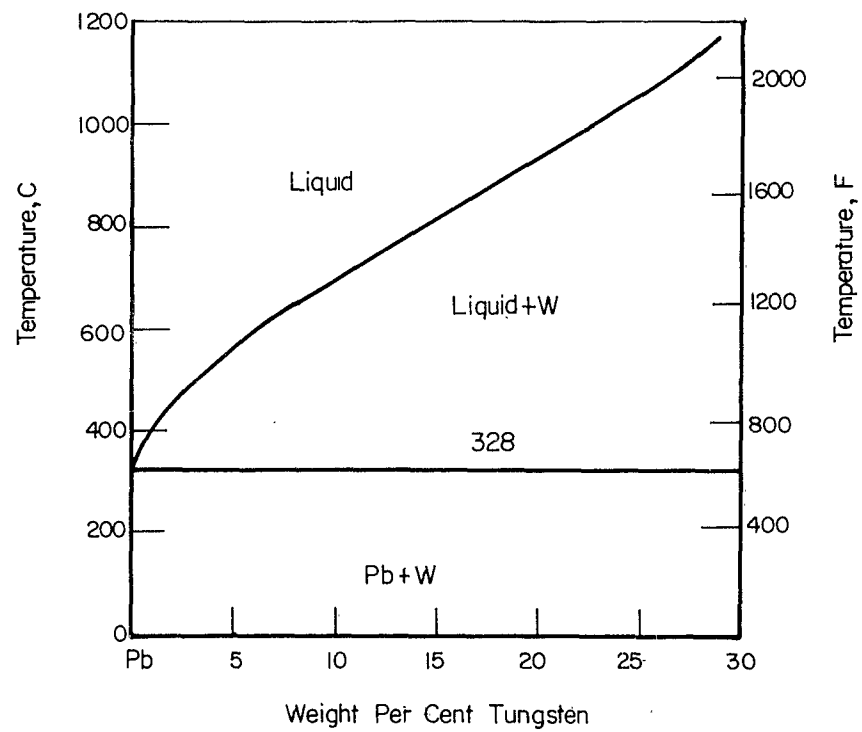
Three intermediate phases were identified by Dolloff, W_2C , α -WC, and β -WC.⁽²⁶⁰⁾ Only one crystal modification of W_2C was found by Dolloff in contrast to Becker's⁽¹⁶⁵⁾ and Goldschmidt's⁽²⁵⁹⁾ results of two structures -- the low-temperature modification being hexagonal with $a = 2.994$ Å, $c = 4.724$ Å, and $c/a = 1.578$, and the high-temperature W_2C modification having a face-centered cubic structure with $a = 4.16$ Å. A newly reported phase, β -WC, has a face-centered cubic structure with $a = 4.125$ Å for the composition $WC_{0.82}$.⁽²⁶⁰⁾ Hexagonal α -WC is essentially a line compound with $a = 2.906$ Å, $c = 2.837$ Å, and $c/a = 0.976$.⁽¹⁶⁵⁾ Goldsmith reported the solubility of carbon in tungsten as 0.30 atomic per cent near 2400 C, decreasing to 0.05 atomic per cent near 2000 C. and to insignificant amounts at lower temperatures.⁽²⁵⁹⁾

TUNGSTEN-IRIDIUM SYSTEM



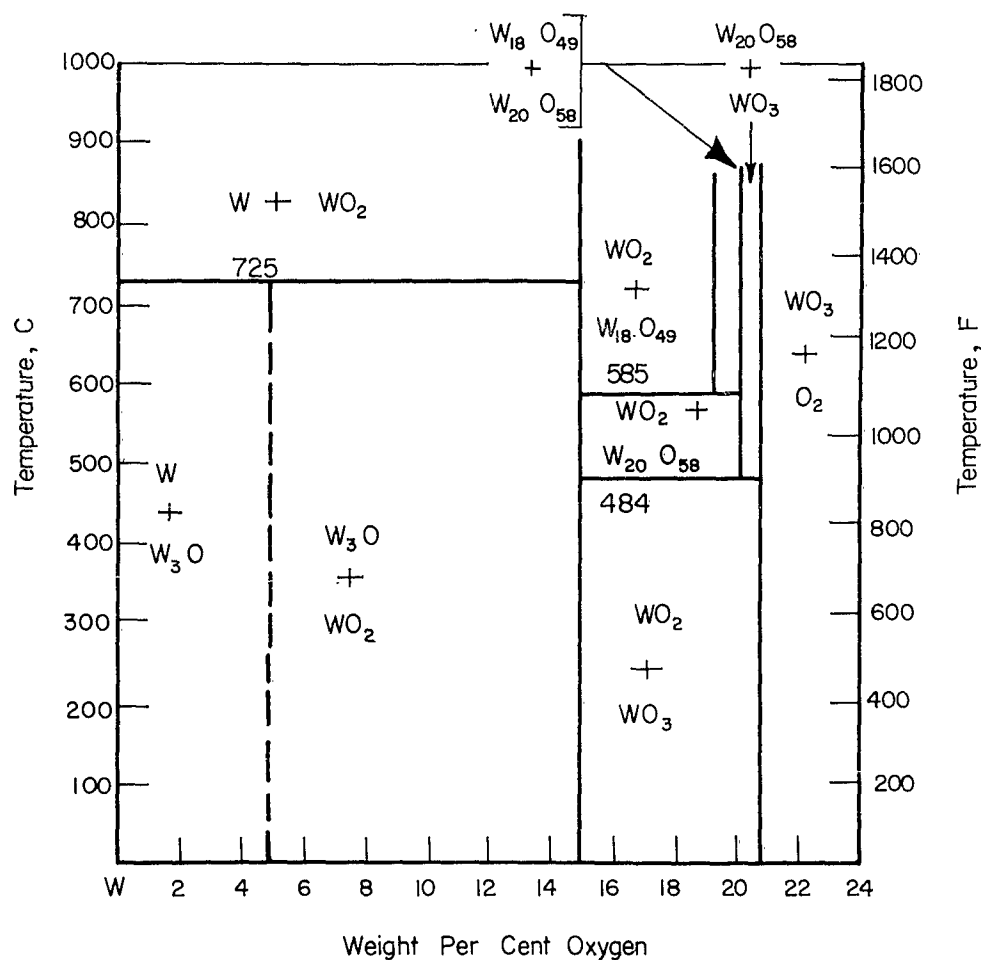
ϵ has been identified as a close-packed-hexagonal structure with $a = 2.736 \text{ \AA}$ and $c/a = 1.602$ at the iridium side and $a = 2.764 \text{ \AA}$ and $c/a = 1.611$ at the tungsten side of the homogeneity range.⁽²⁶¹⁾ The solubility of iridium in tungsten is about 11 weight per cent (~10 atomic per cent) at 2540 C, decreasing to about 5 weight per cent (~4 atomic per cent) at 1810 C. Tungsten is soluble in iridium up to 18 weight per cent (~19 atomic per cent) at 2305 C.⁽²⁶²⁾

TUNGSTEN-LEAD SYSTEM



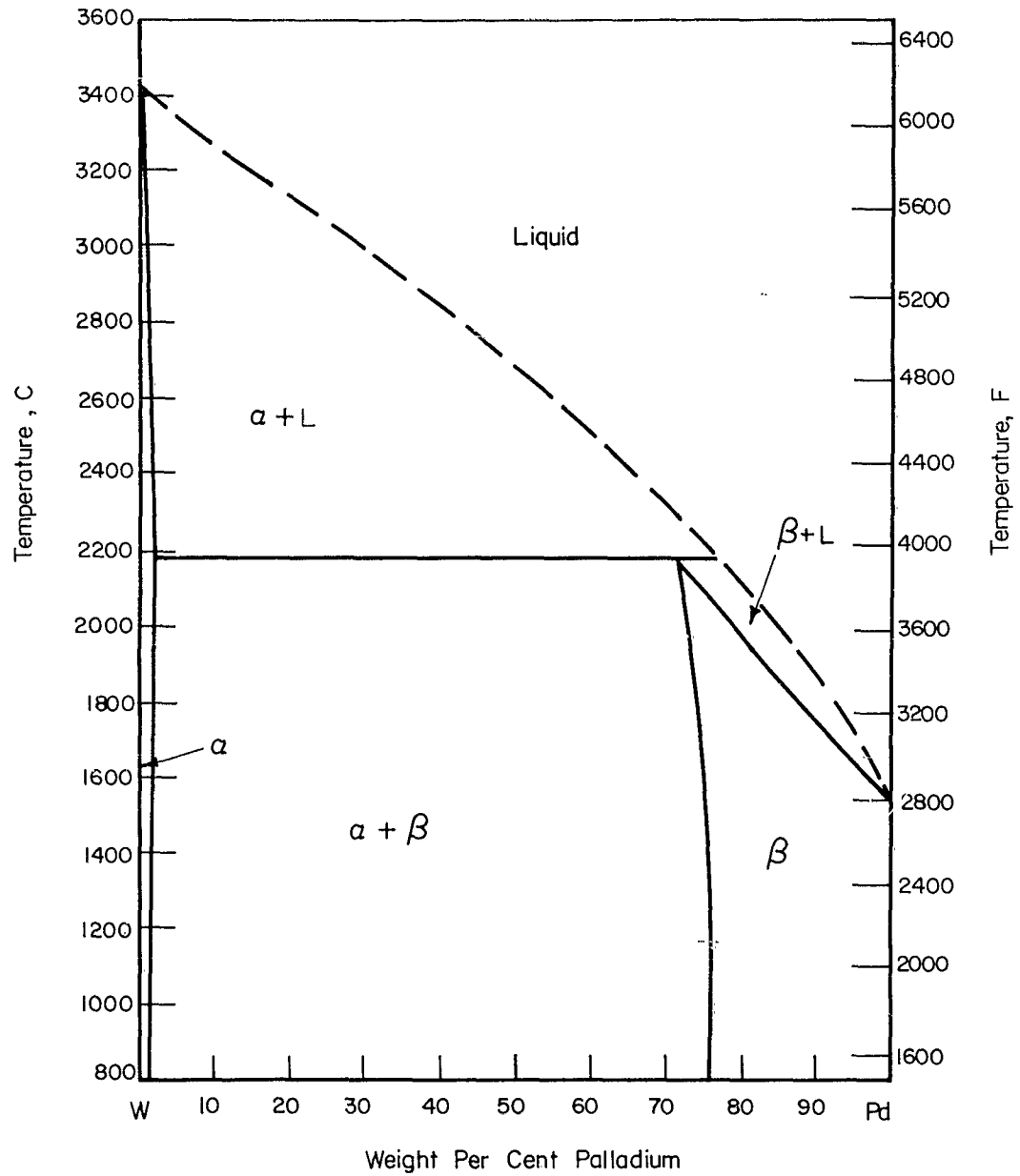
No intermetallic compounds exist in the system. (263)

TUNGSTEN-OXYGEN SYSTEM



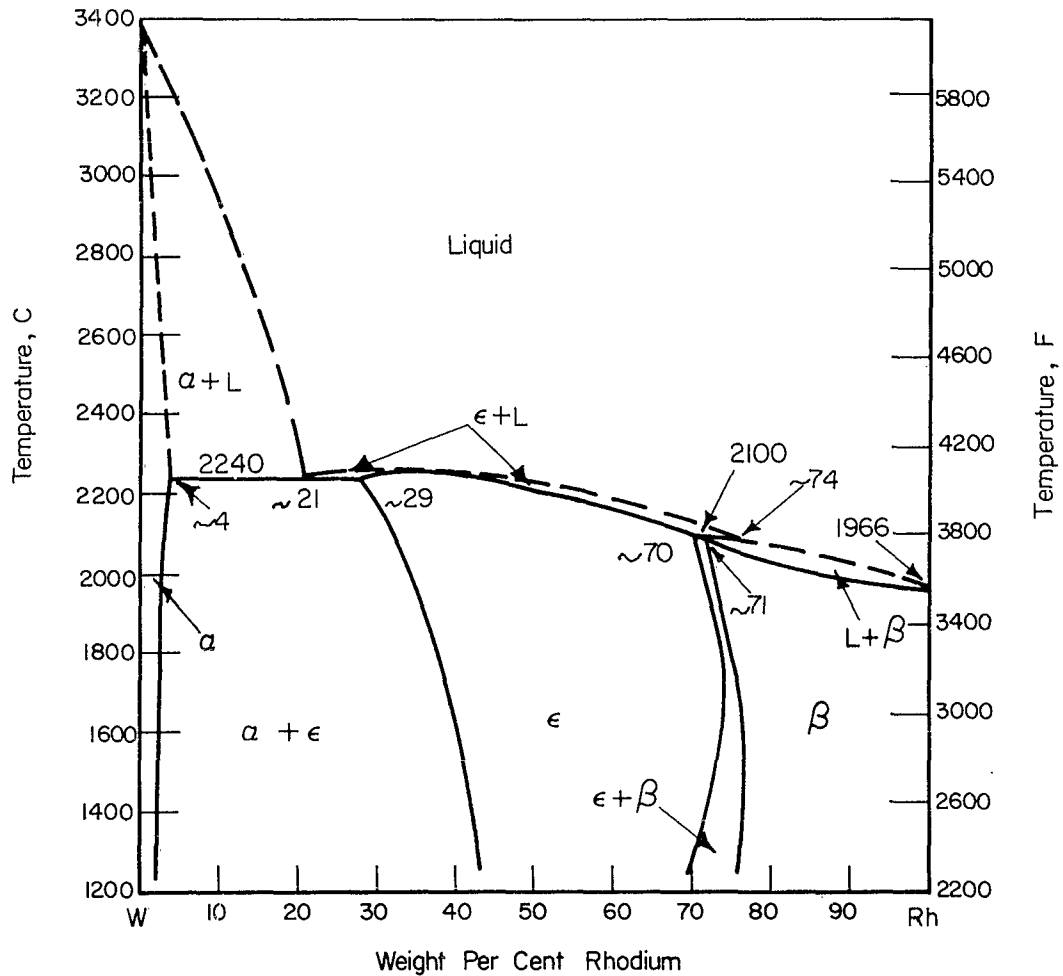
WO₂ has a monoclinic structure isomorphous with MoO₂, with $a = 5.560 \text{ \AA}$, $b = 4.884 \text{ \AA}$, $c = 5.546 \text{ \AA}$, $\beta = 118.93^\circ$, and 12 atoms per unit cell.⁽¹⁹⁹⁾ W₁₈O₄₉ has a monoclinic structure with $a = 18.32 \text{ \AA}$, $b = 3.79 \text{ \AA}$, $c = 11.04 \text{ \AA}$, $\beta = 115^\circ 2'$, and 67 atoms per unit cell.⁽²⁰⁰⁾ The structure of W₂₀O₅₈ is closely related to monoclinic ReO₃ (DO₉ type) with $a = 12.1 \text{ \AA}$, $b = 3.78 \text{ \AA}$, $c = 23.4 \text{ \AA}$, and $\beta = 95^\circ$.⁽²⁰⁰⁾ WO₃ is reported to have three structural modifications. The room temperature form is monoclinic with $a = 7.285 \text{ \AA}$, $b = 7.517 \text{ \AA}$, $c = 3.835 \text{ \AA}$, and $\beta = 90.90^\circ$.⁽²⁰¹⁾ At -50°C, a polymorphic transformation occurs, resulting in a structure of higher symmetry than the room-temperature modification.⁽²⁰²⁾ Between 700 and 750°C, a polymorphic transformation occurs, resulting in a tetragonal structure with $a = 5.25 \text{ \AA}$, $c = 3.92 \text{ \AA}$, $c/a = 0.746$, and 8 atoms per unit cell.⁽²⁰³⁾ The homogeneity ranges (oxygen/tungsten ratios) of the oxides at 1258°C are: WO₂ (1.99 to 2.02), W₁₈O₄₉ (2.66 to 2.77), and W₂₀O₅₈ (2.90 to 2.94).⁽²⁰⁰⁾ The diagram is from Reference (264).

TUNGSTEN-PALLADIUM SYSTEM



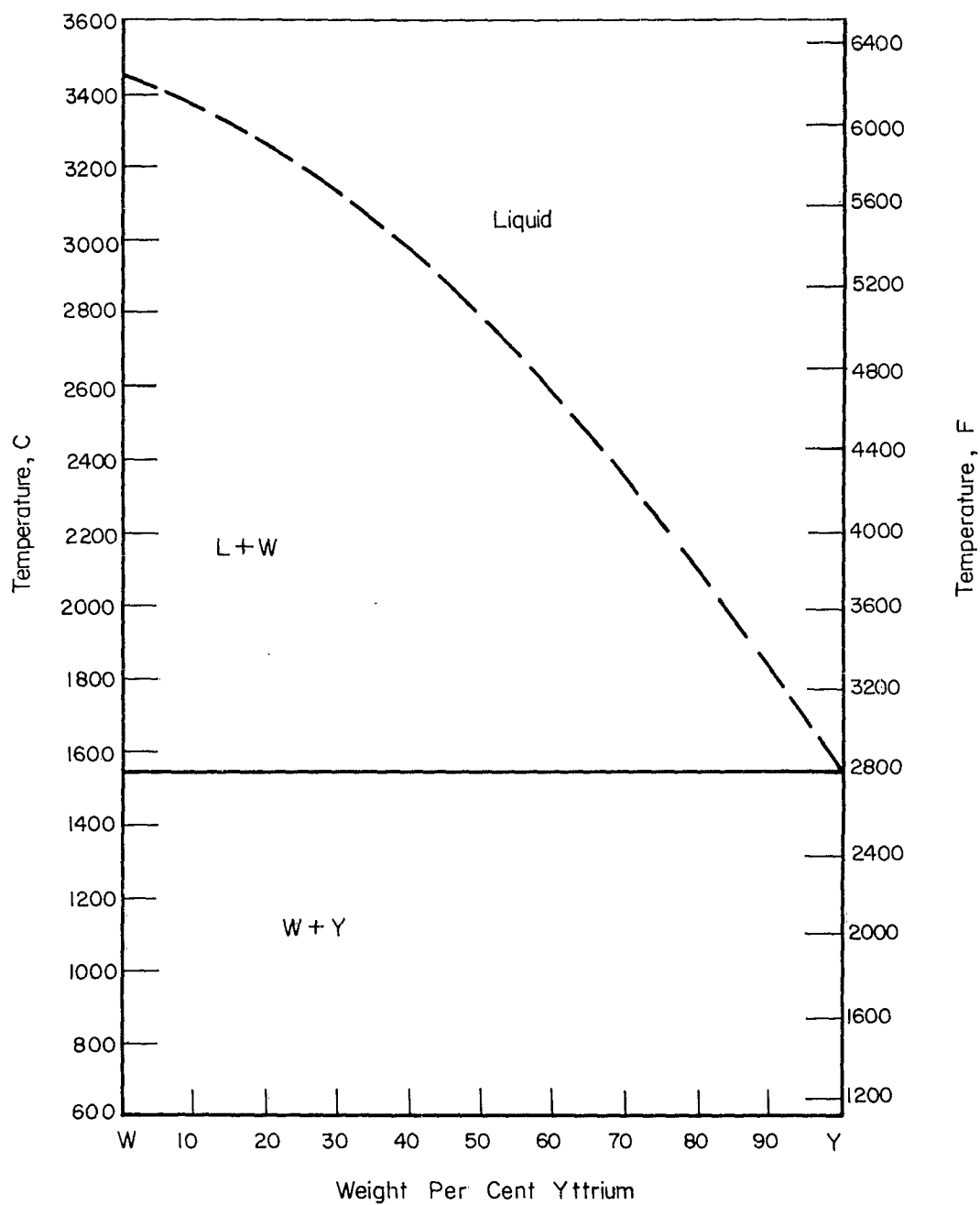
No intermediate phases are found in this system. The solubility of palladium in tungsten is 1.6 weight per cent at 1500 C. Tungsten dissolves in palladium up to 24 weight per cent at 1000 C. (265)

TUNGSTEN-RHODIUM SYSTEM



ε has a close-packed-hexagonal structure with $a = 2.708 \text{ \AA}$, $c = 4.328 \text{ \AA}$, and $c/a = 1.598$ at 70 weight per cent (80.8 atomic per cent) rhodium.⁽³⁰⁾ The solubility of rhodium in tungsten is about 4 weight per cent (6 atomic per cent) at 2240 C, decreasing to about 3 weight per cent (4 atomic per cent) at 1300 C. Tungsten is soluble in rhodium up to 29 weight per cent (19 atomic per cent) at 2100 C, decreasing to about 23 weight per cent (14 atomic per cent) at 1400 C.⁽²⁶²⁾

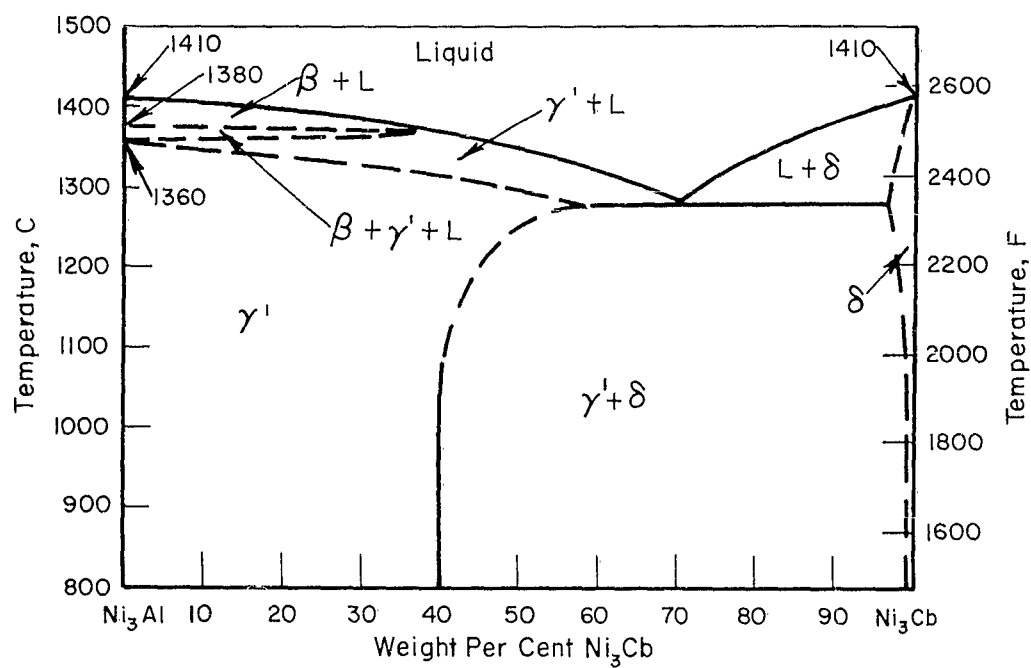
TUNGSTEN-YTTRIUM SYSTEM



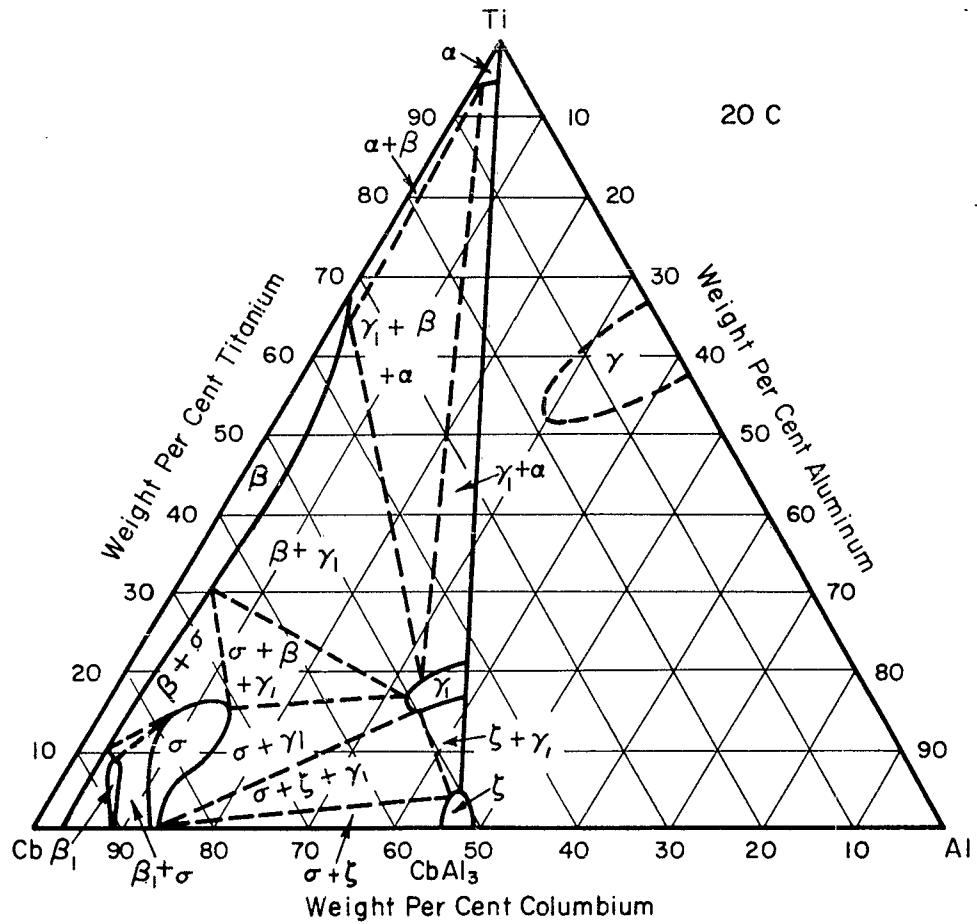
Terminal solubilities are probably less than 1 atomic per cent.⁽²⁴⁷⁾

TERNARY PHASE
DIAGRAMS

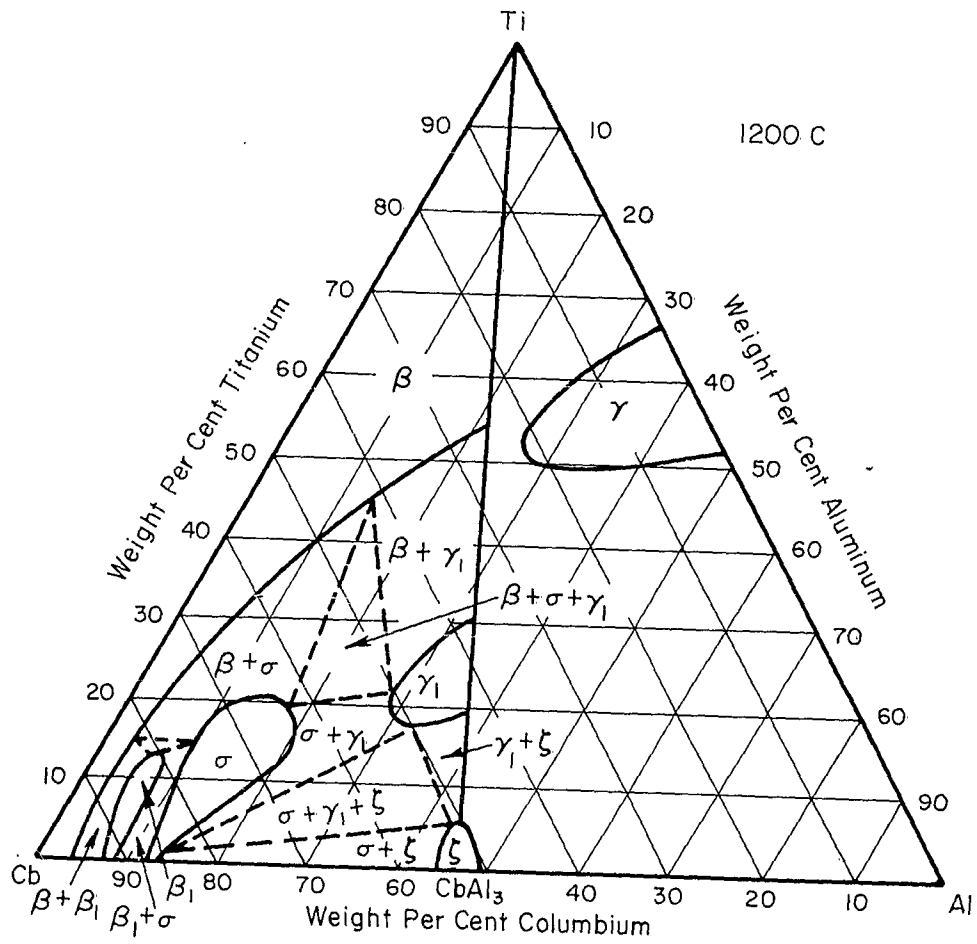
COLUMBIUM-ALUMINUM-NICKEL SYSTEM (Ni_3Al - Ni_3Cb SYSTEM) (317)



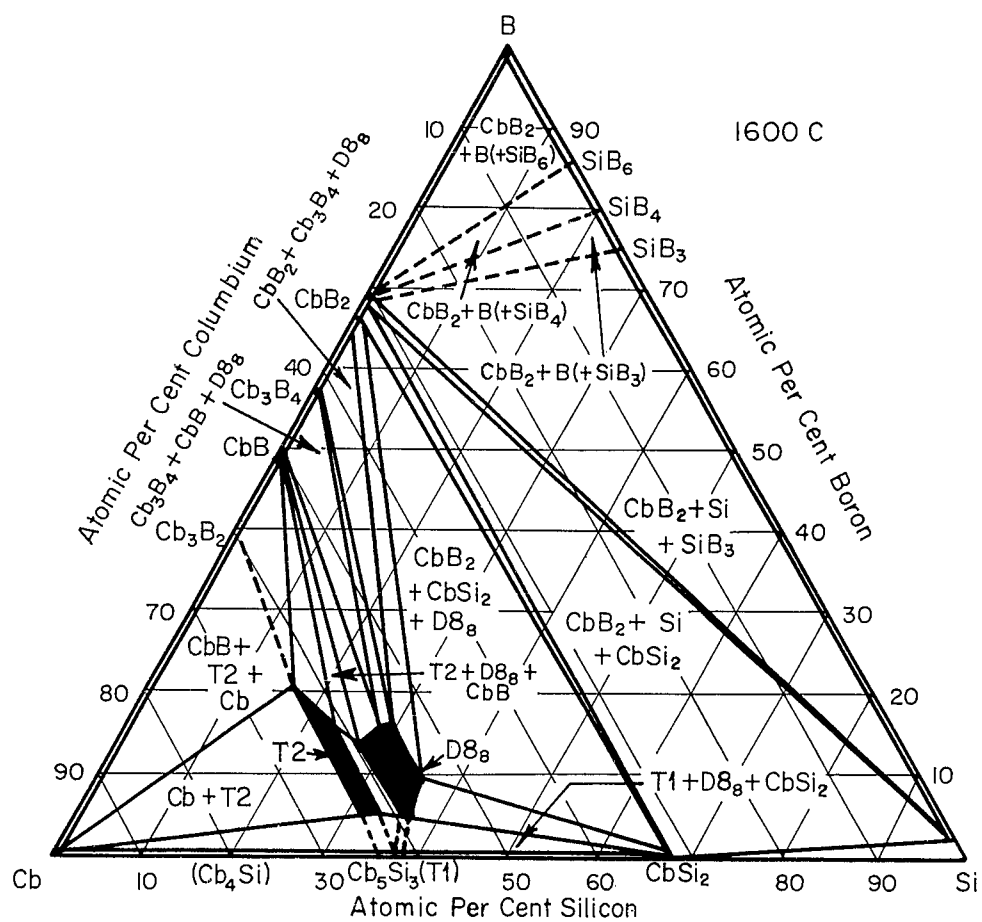
COLUMBIUM-ALUMINUM-TITANIUM SYSTEM (20 C)(269)



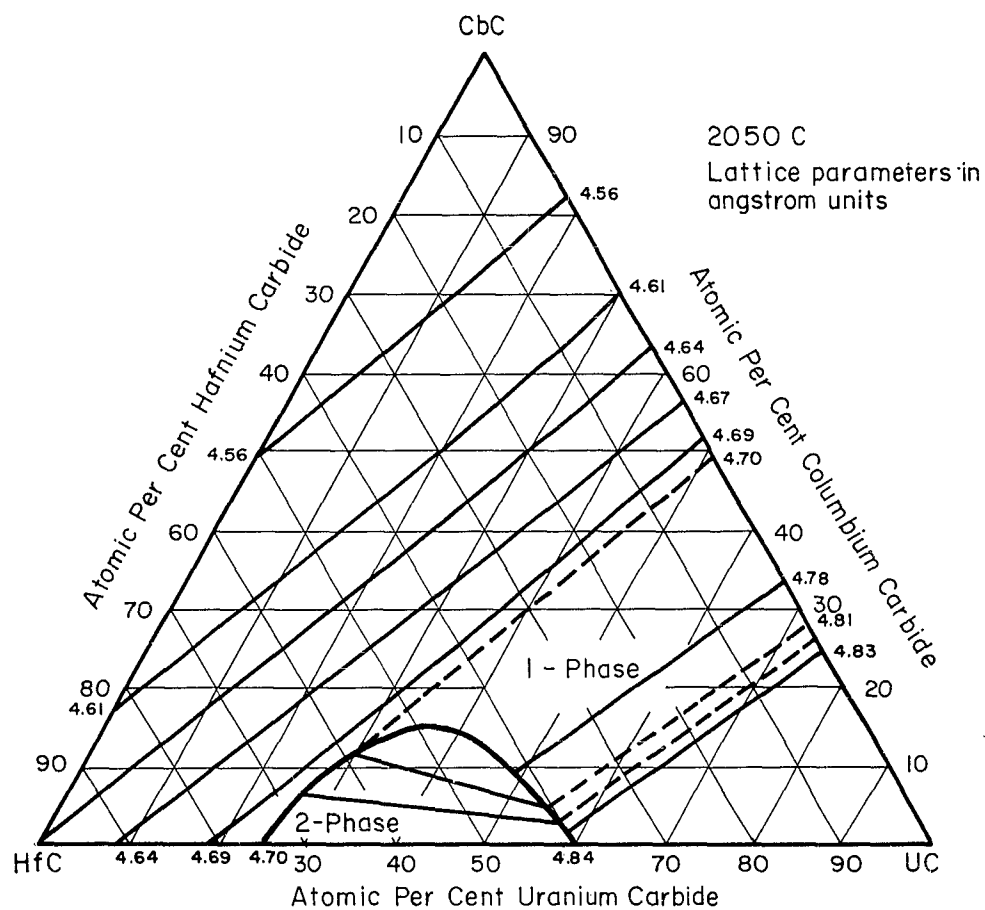
COLUMBIUM-ALUMINUM-TITANIUM SYSTEM (1200 C)(269)



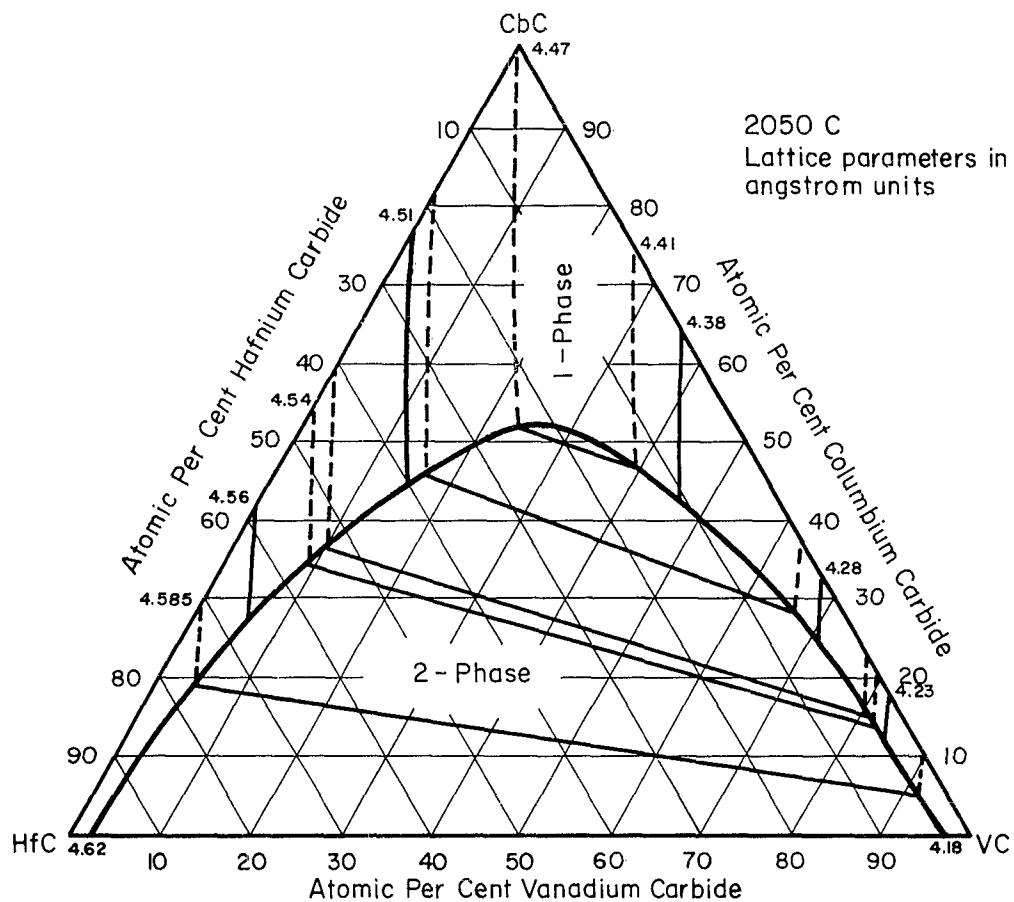
COLUMBIUM-BORON-SILICON SYSTEM (1600 C)(270)



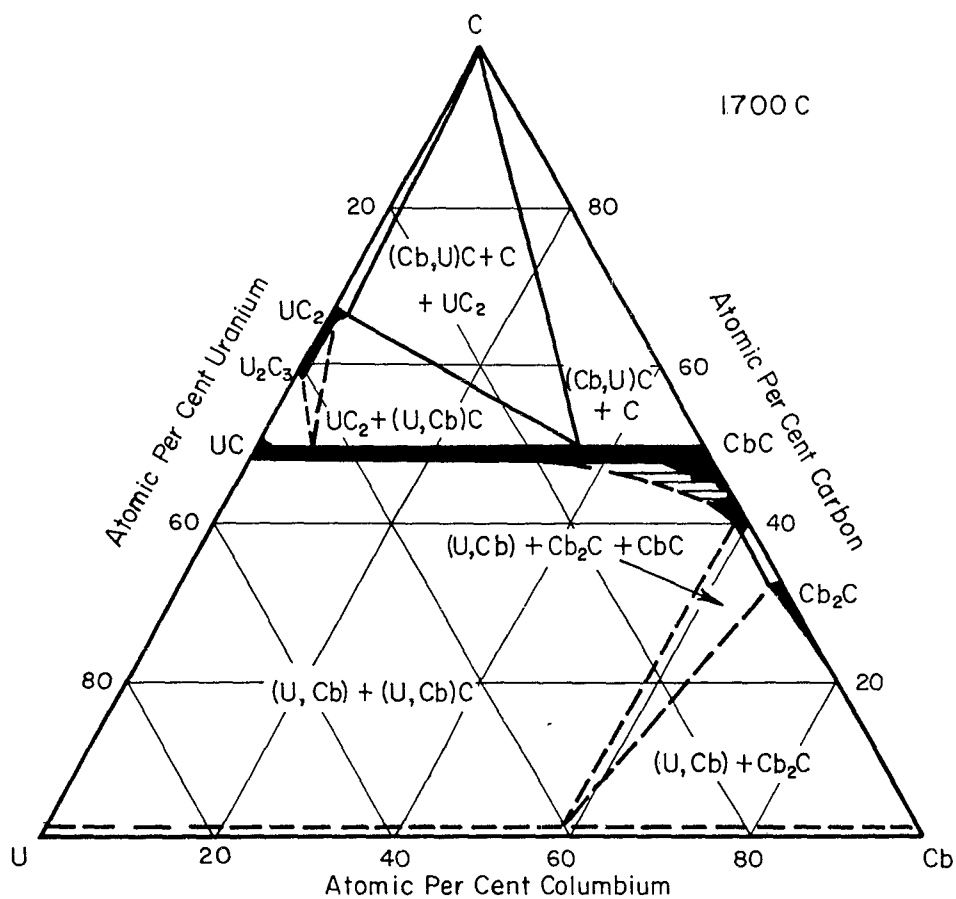
COLUMBIUM CARBIDE-HAFNIUM CARBIDE-URANIUM CARBIDE SYSTEM (2050 C)(271)



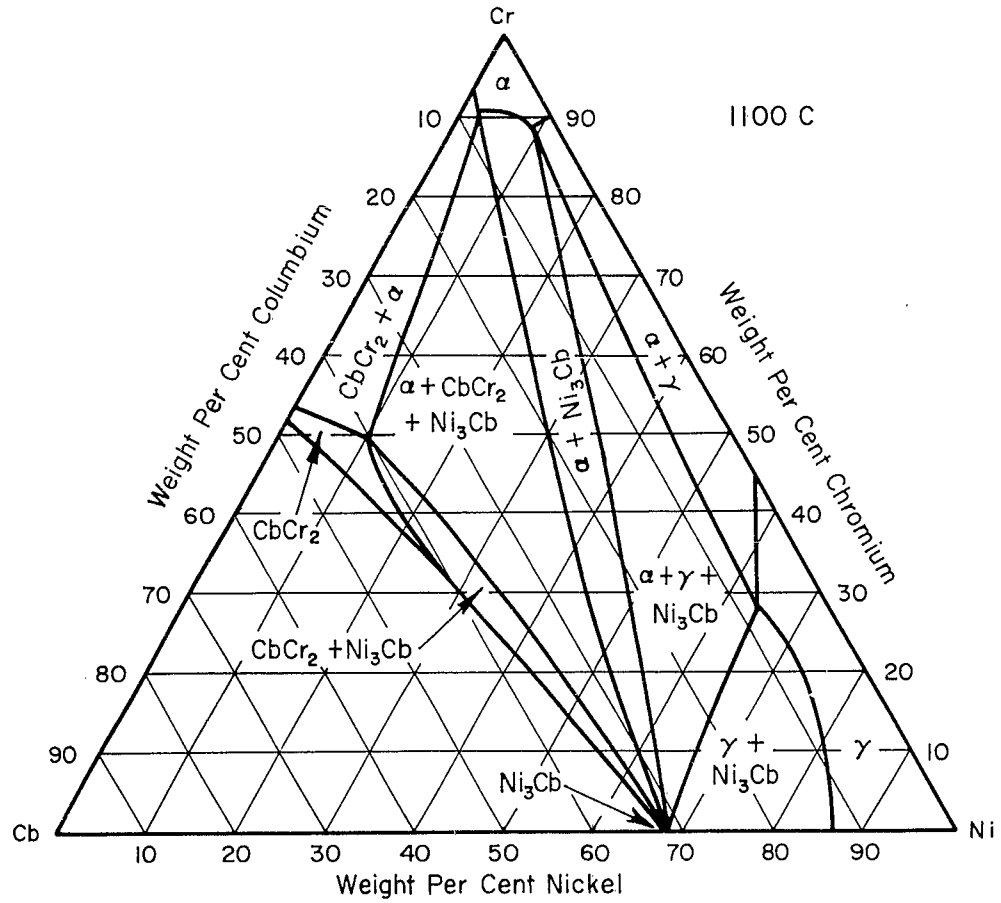
COLUMBIUM CARBIDE-HAFNIUM CARBIDE-VANADIUM CARBIDE SYSTEM (2050 C)⁽²⁷¹⁾



COLUMBIUM-CARBON-URANIUM SYSTEM (1700 C)(272)



COLUMBIUM-CHROMIUM-NICKEL SYSTEM (1100 C)⁽²⁷³⁾

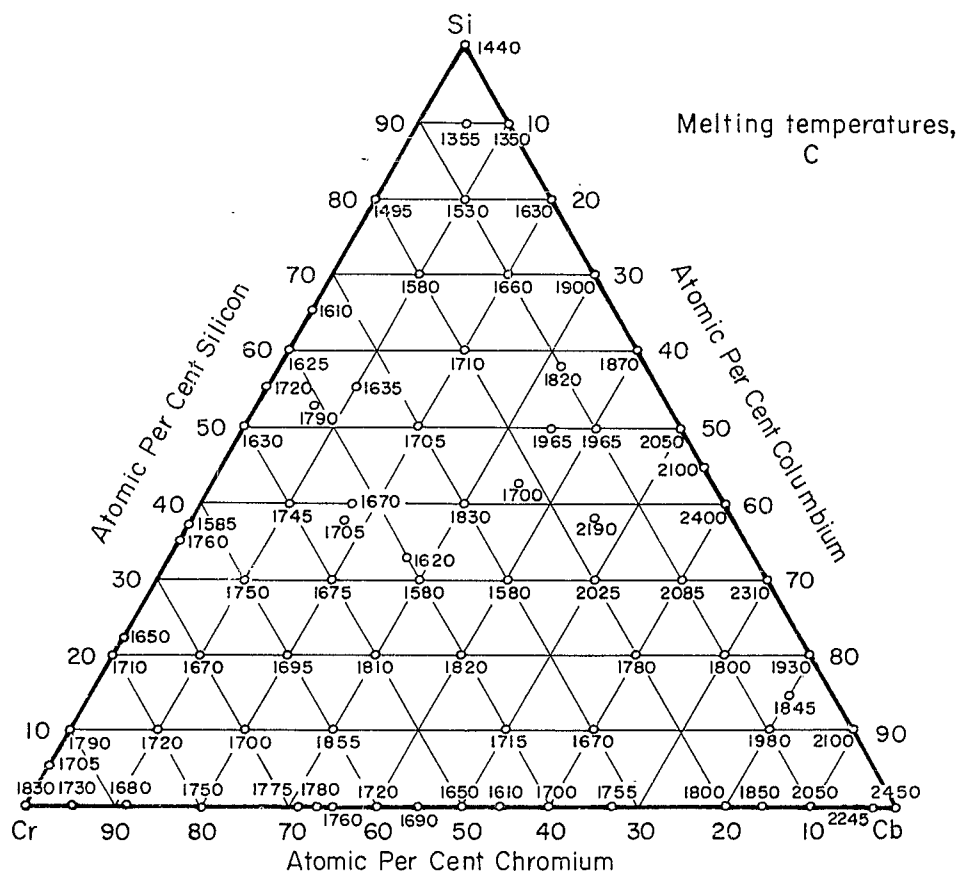


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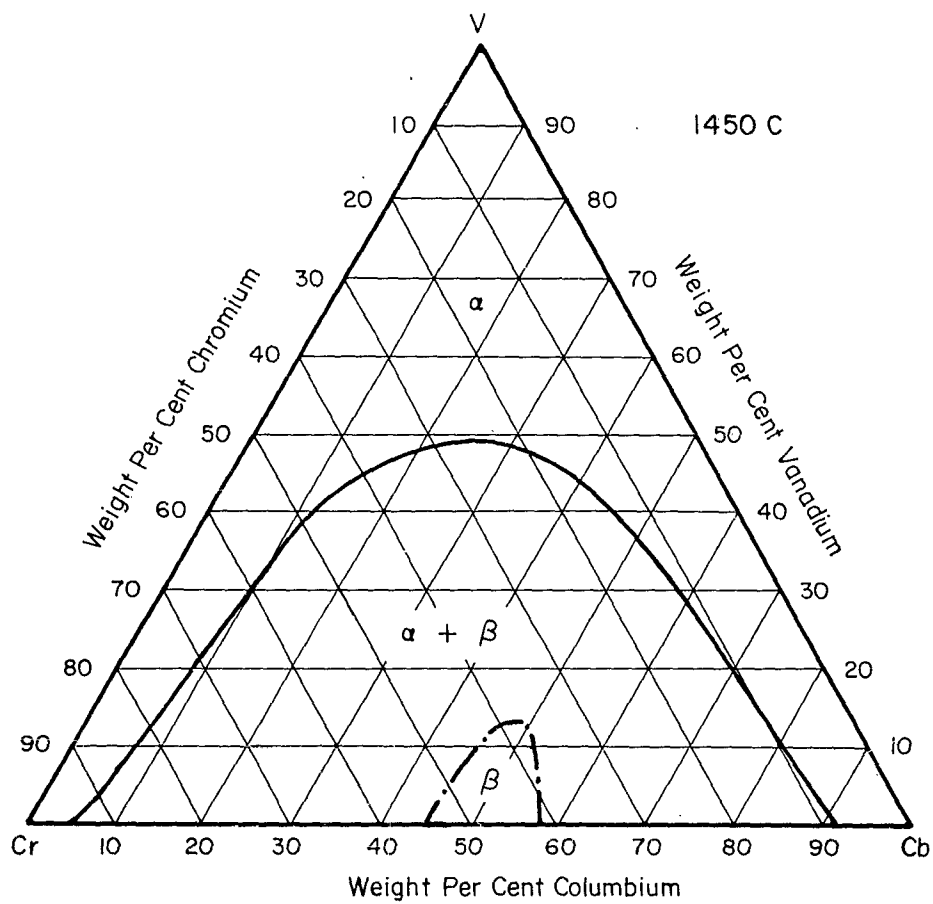


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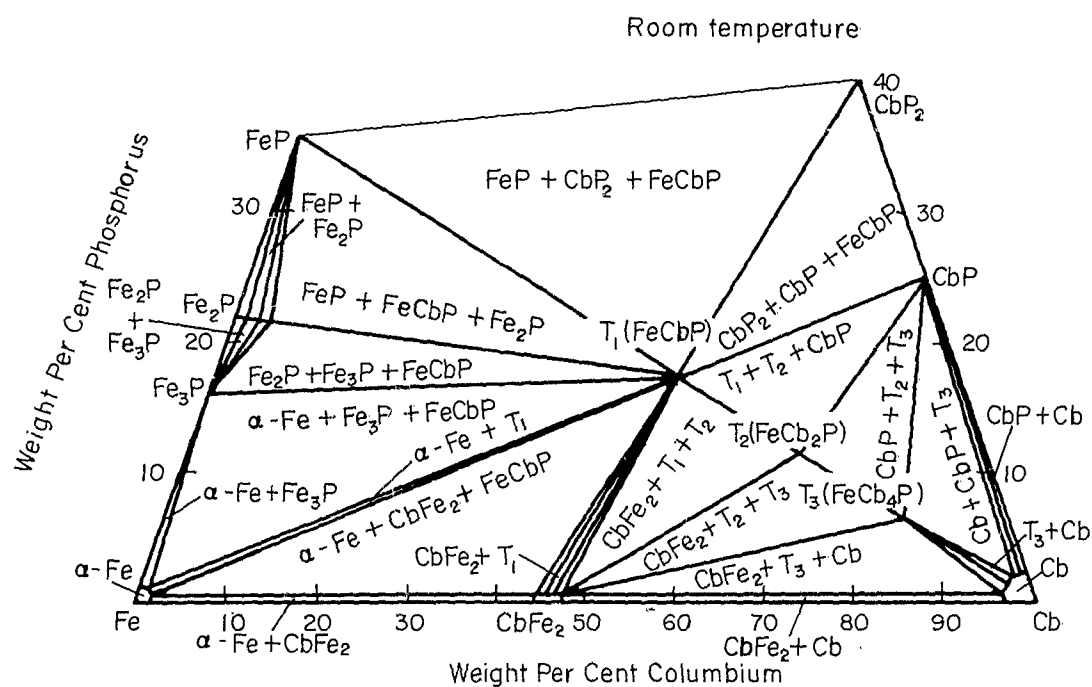
COLUMBIUM-CHROMIUM-SILICON SYSTEM (MELTING TEMPERATURES, C)⁽²⁷⁴⁾



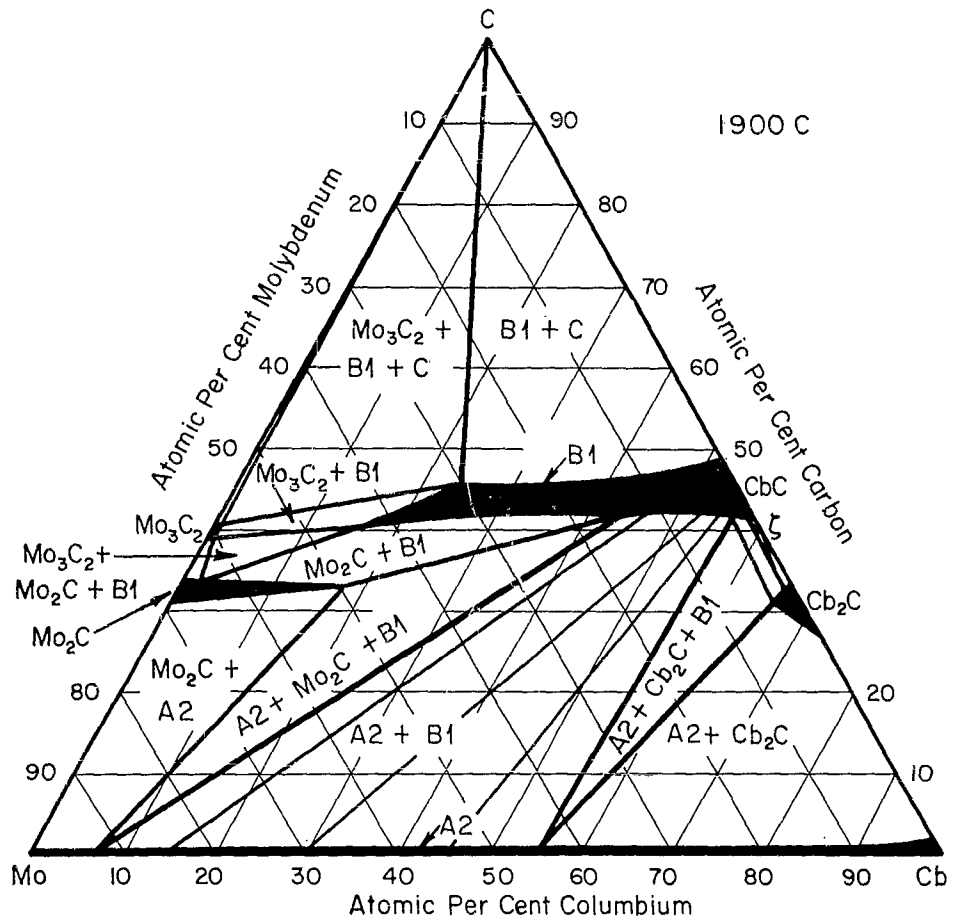
COLUMBIUM-CHROMIUM-VANADIUM SYSTEM (1450 C)⁽²⁷⁵⁾



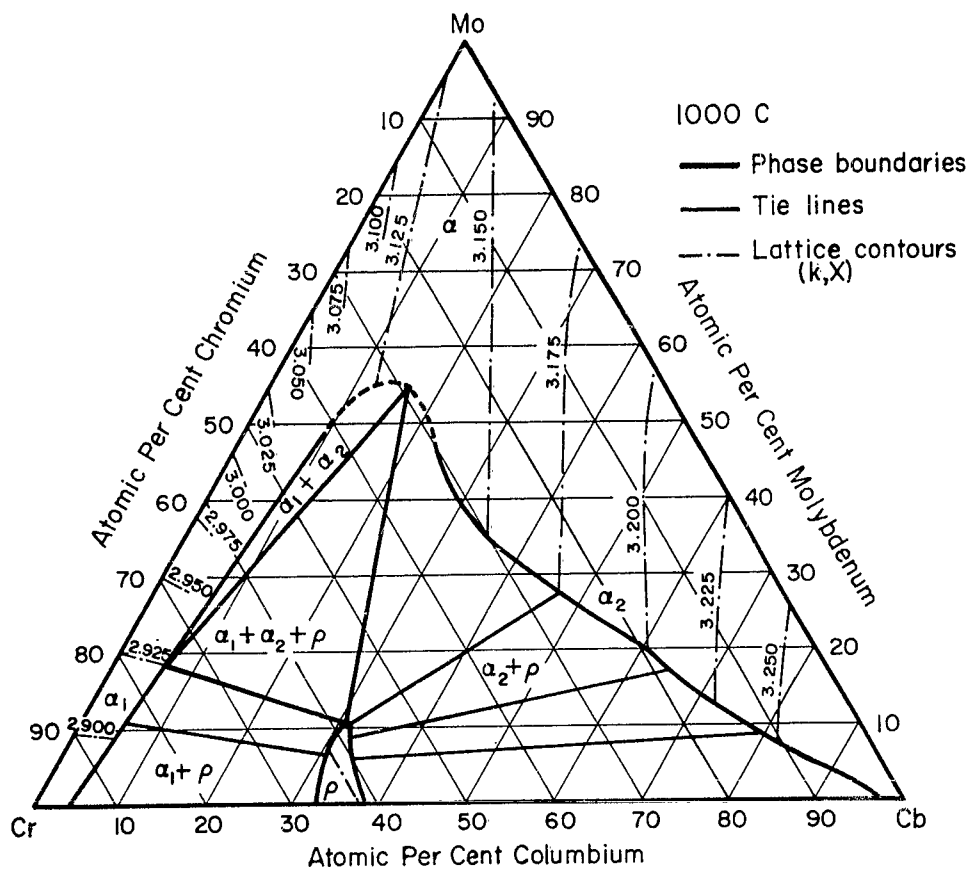
COLUMBIUM-IRON-PHOSPHORUS SYSTEM (ROOM TEMPERATURE)(276)



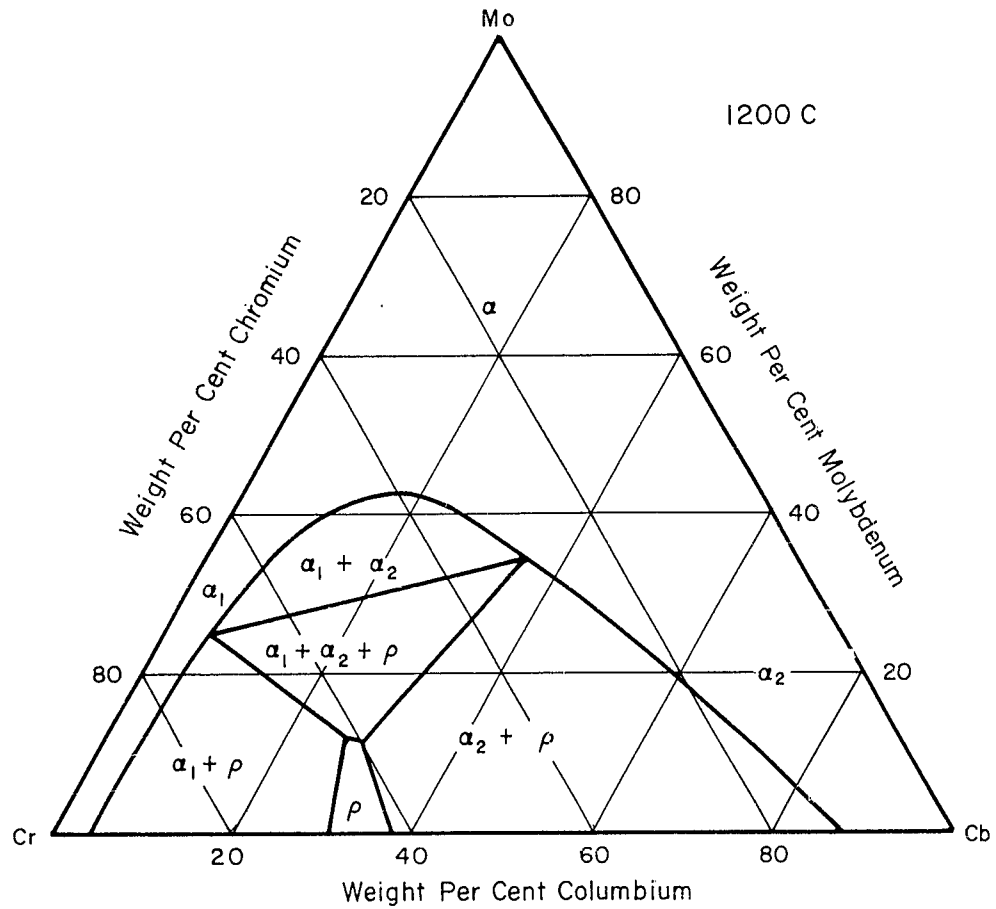
COLUMBIUM-MOLYBDENUM-CARBON SYSTEM (1900 C)(277)



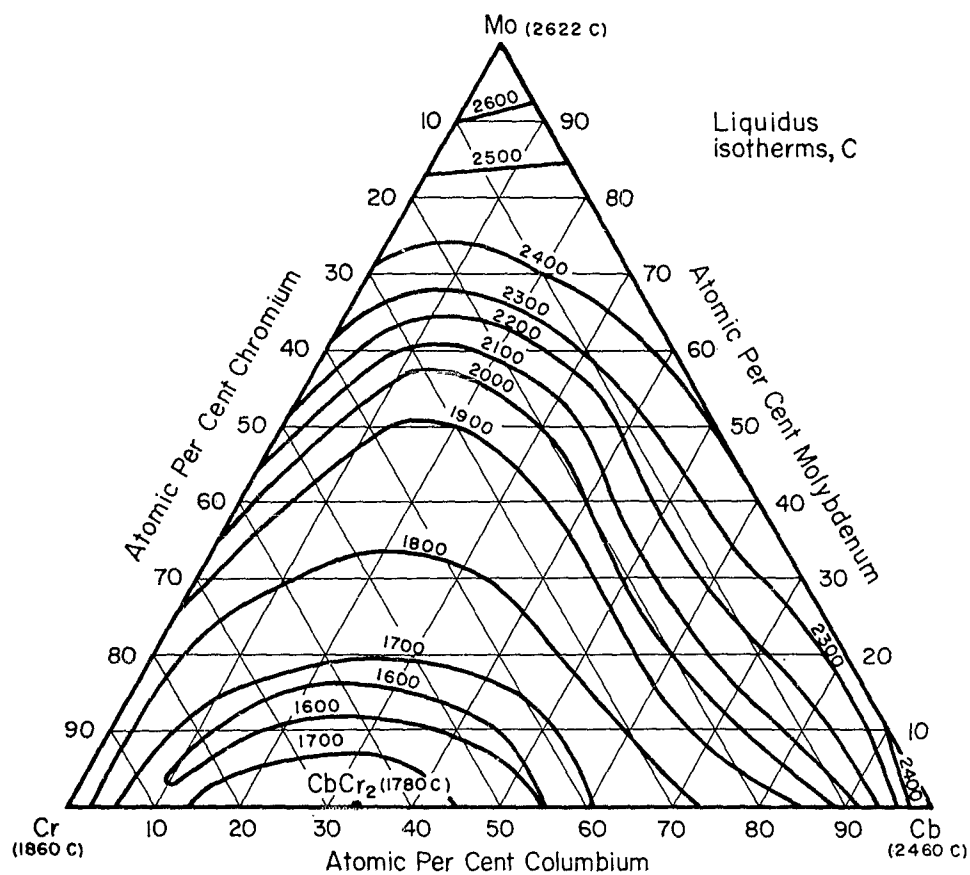
COLUMBIUM-MOLYBDENUM-CHROMIUM SYSTEM (1000 C)⁽²¹⁸⁾



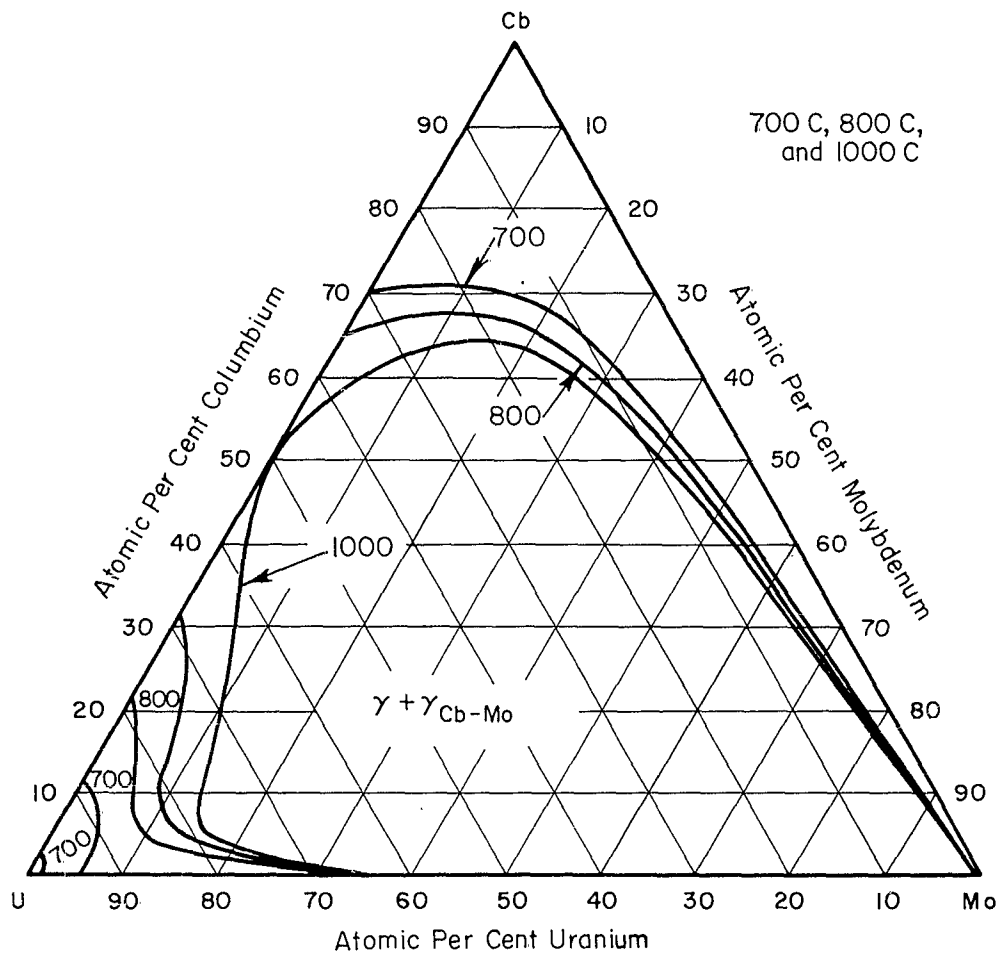
COLUMBIUM-MOLYBDENUM-CHROMIUM SYSTEM (1200 C)(279)



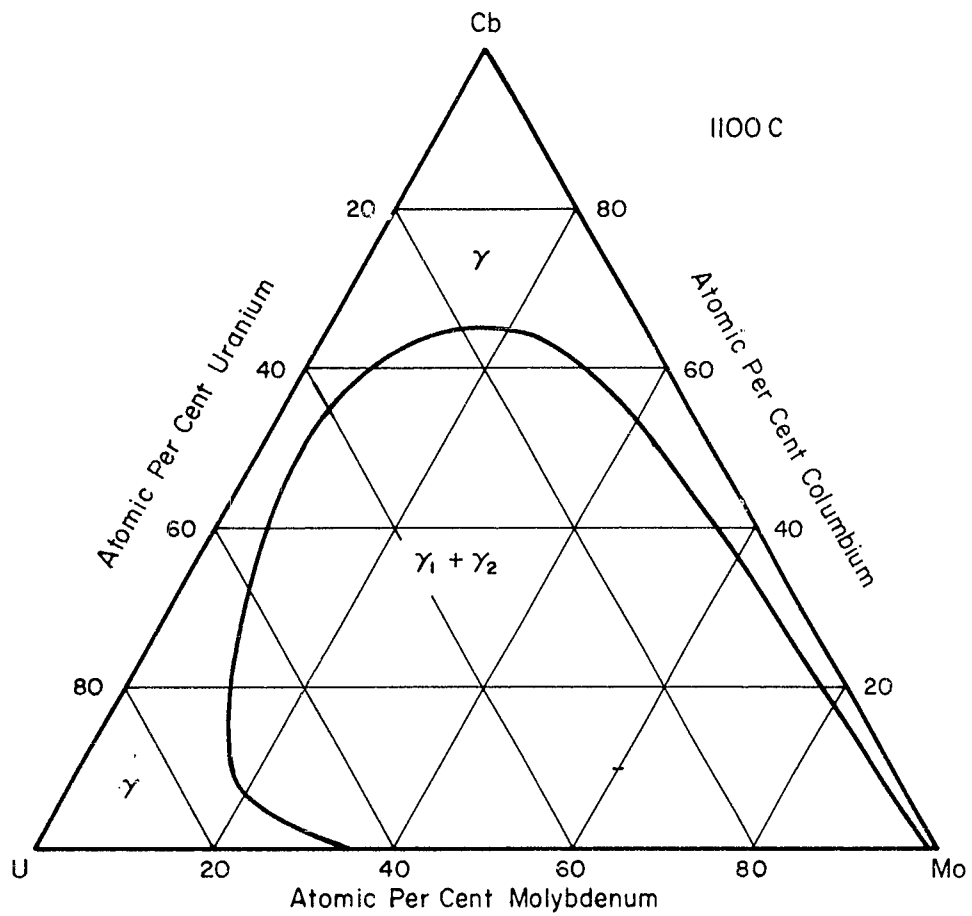
COLUMBIUM-MOLYBDENUM-CHROMIUM SYSTEM (LIQUIDUS ISOTHERMS)⁽²⁷⁸⁾



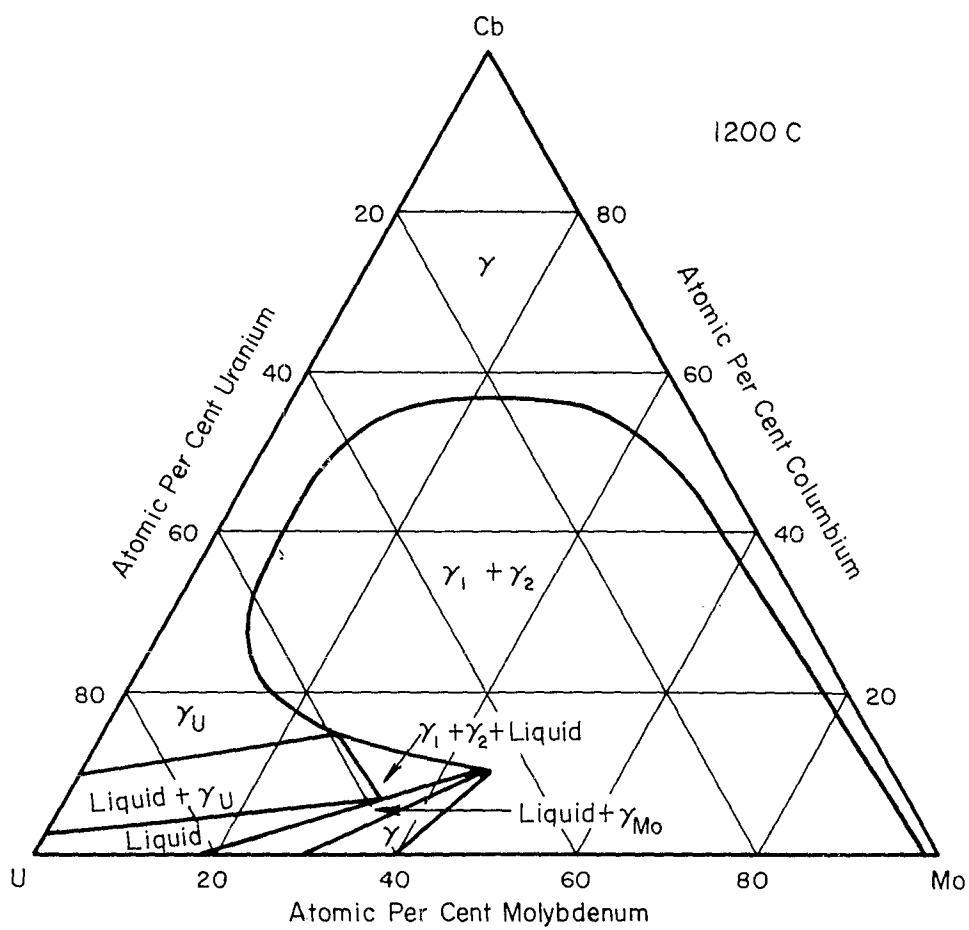
COLUMBIUM-MOLYBDENUM-URANIUM SYSTEM (700 C, 800 C, 900 C)(280)



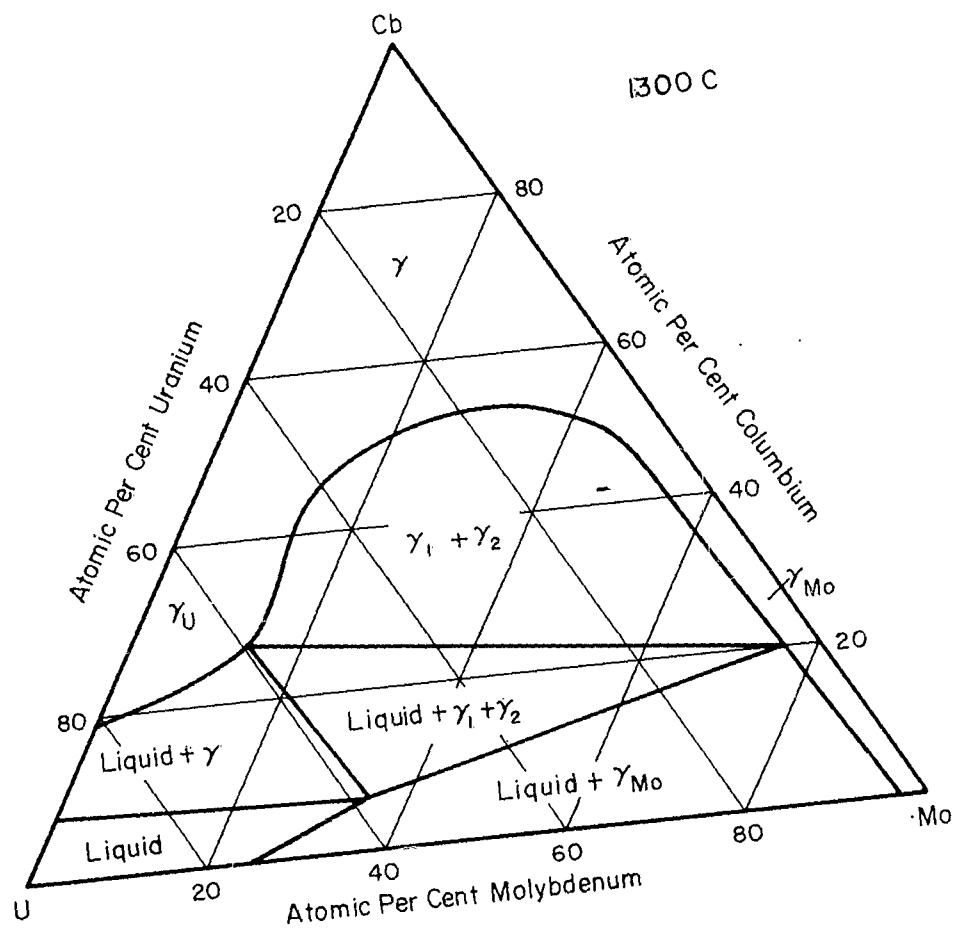
COLUMBIUM-MOLYBDENUM-URANIUM SYSTEM (1100 C)⁽³¹⁸⁾



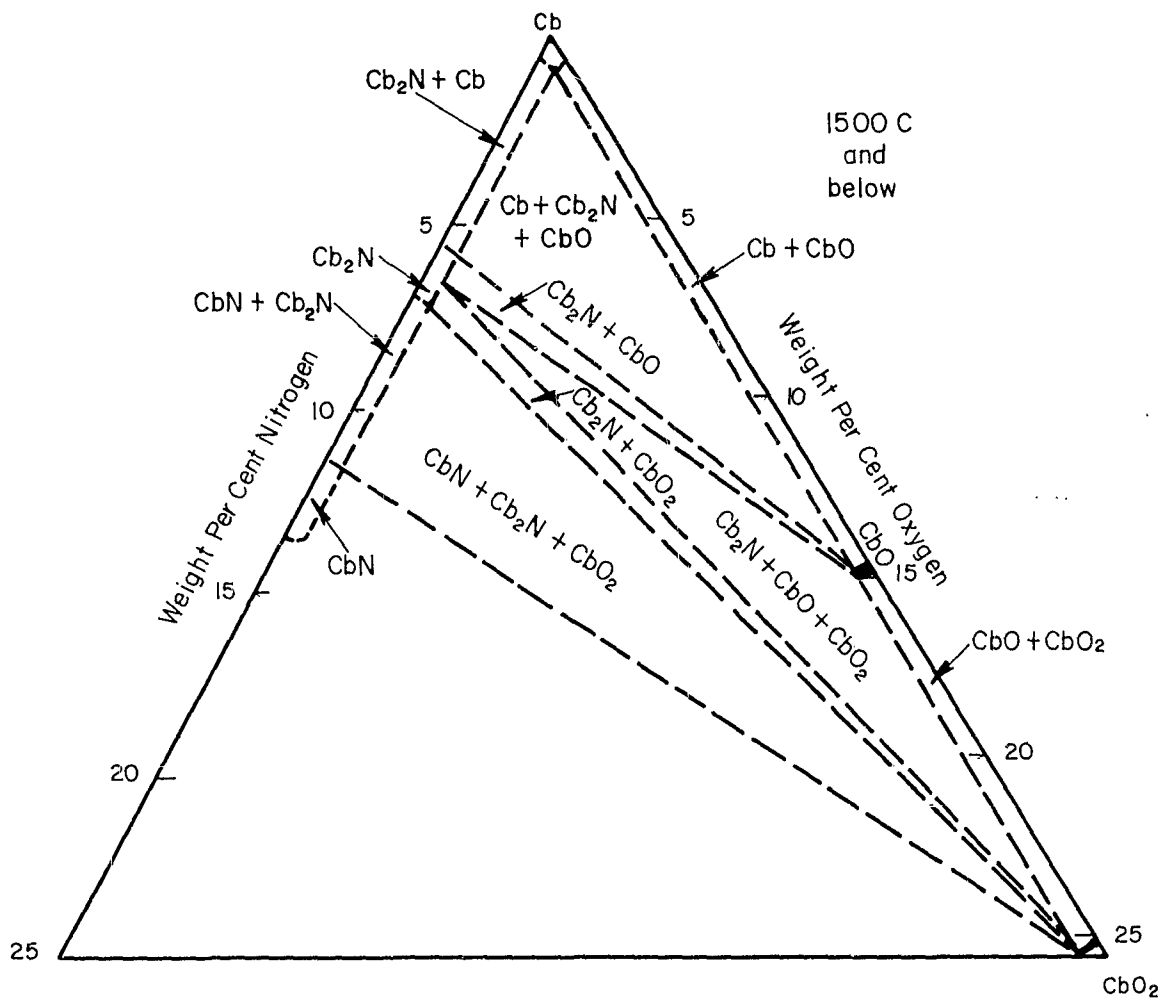
COLUMBIUM-MOLYBDENUM-URANIUM SYSTEM (1200 C)⁽³¹⁸⁾



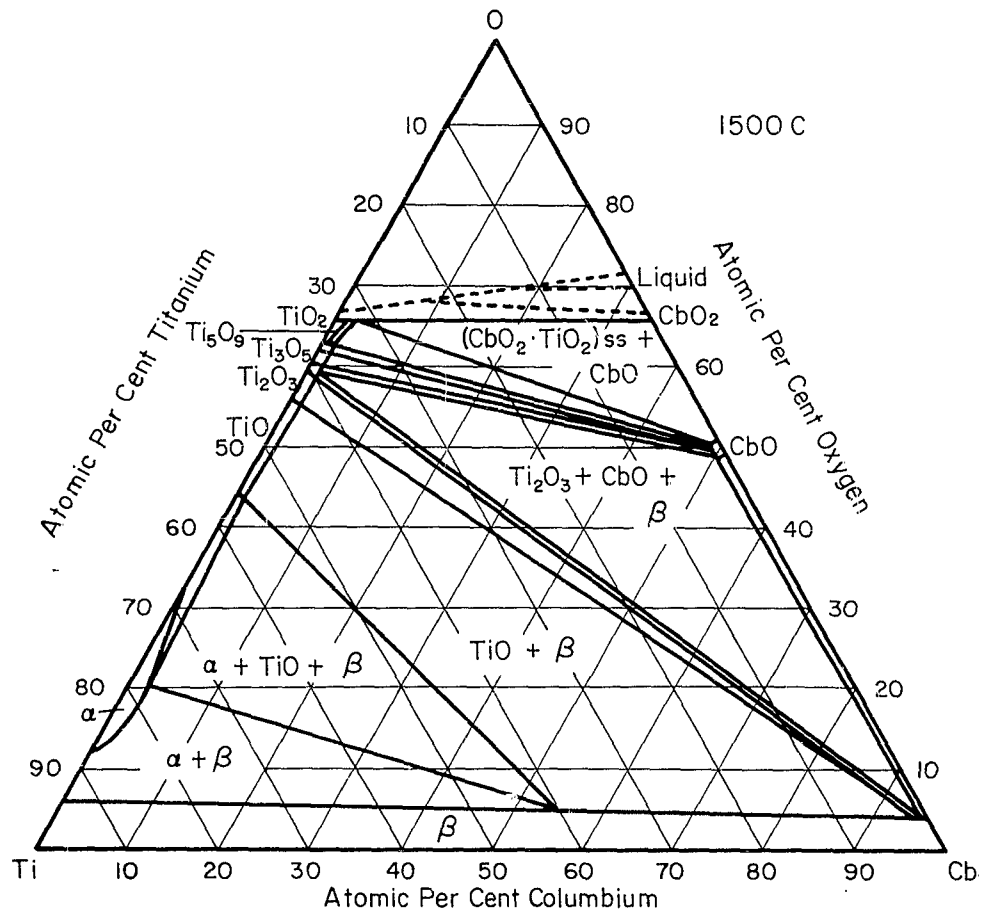
COLUMBIUM-MOLYBDENUM-URANIUM SYSTEM (1300 C)(318)



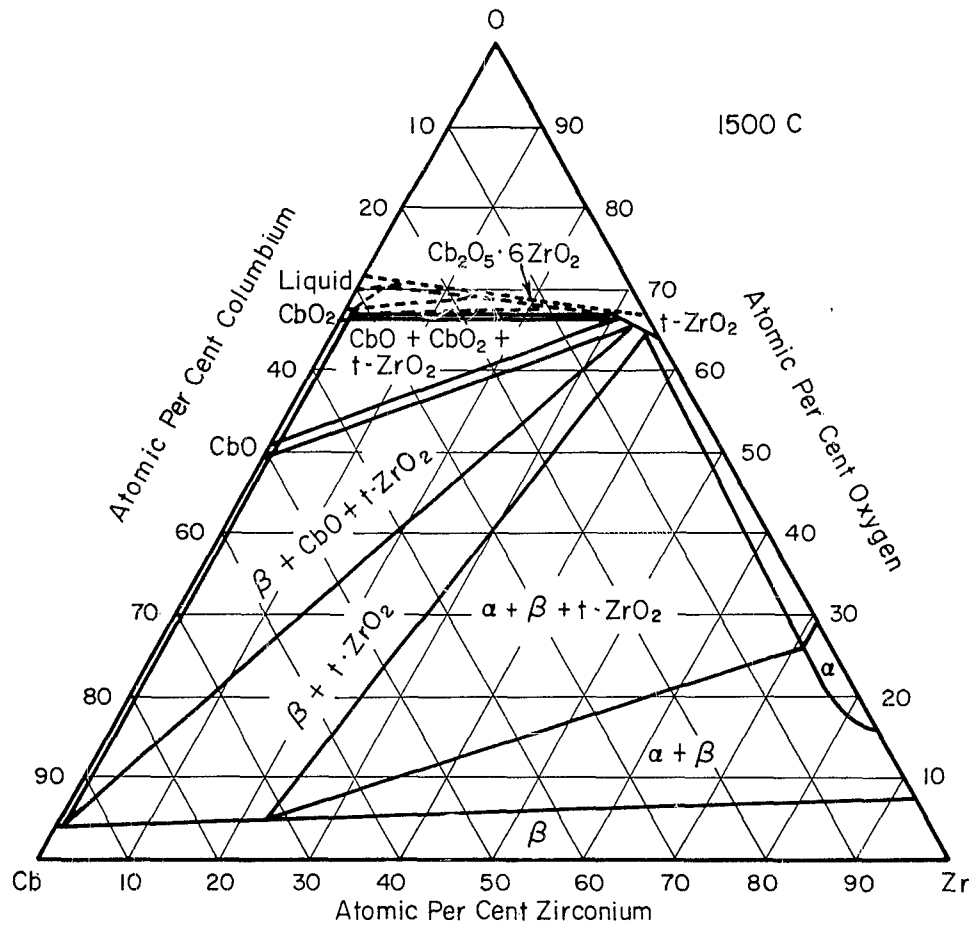
COLUMBIUM-NITROGEN-OXYGEN SYSTEM (1500 C)⁽²⁸¹⁾



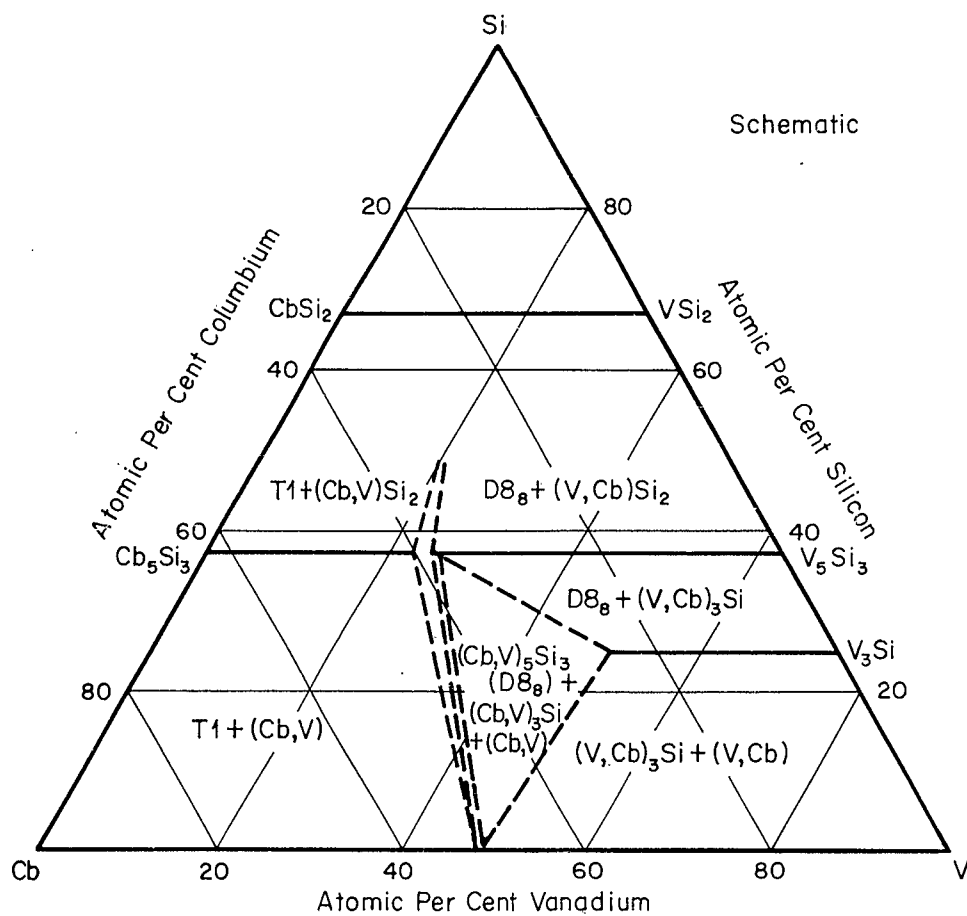
COLUMBIUM-OXYGEN-TITANIUM SYSTEM (1500 C)⁽³⁰¹⁾



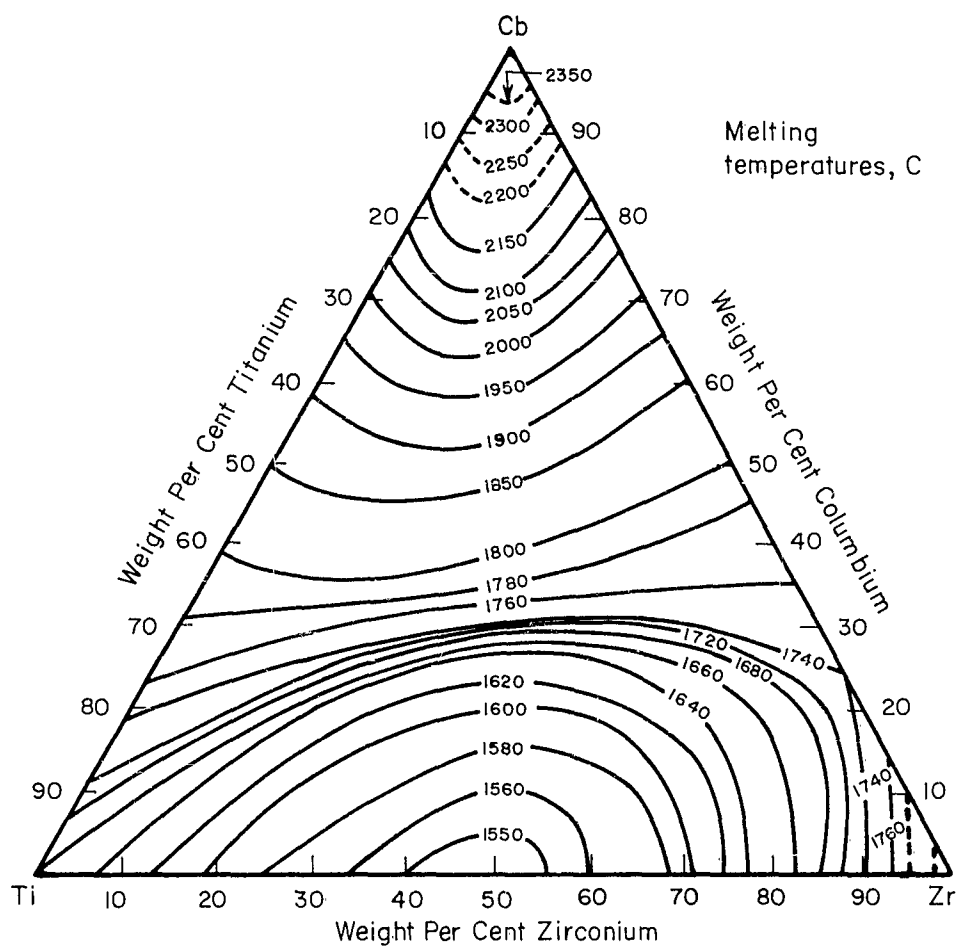
COLUMBIUM-OXYGEN-ZIRCONIUM SYSTEM (1500 C)⁽³⁰¹⁾



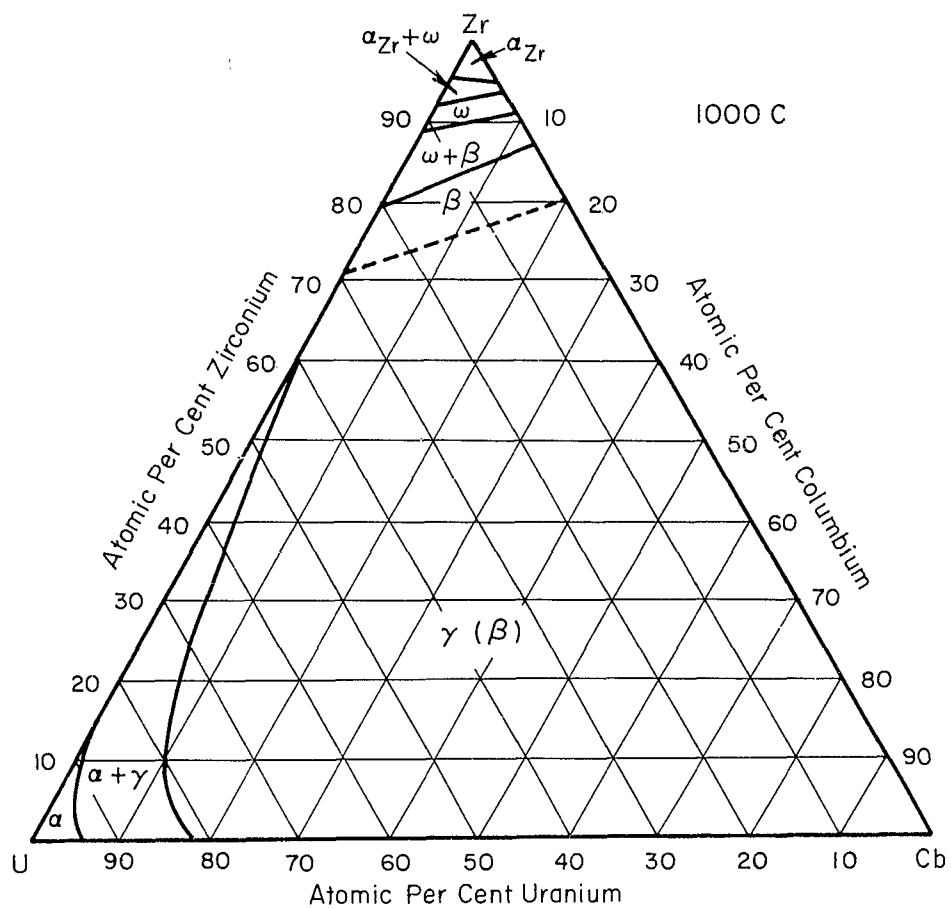
COLUMBIUM-SILICON-VANADIUM SYSTEM (SCHEMATIC)⁽³¹⁵⁾



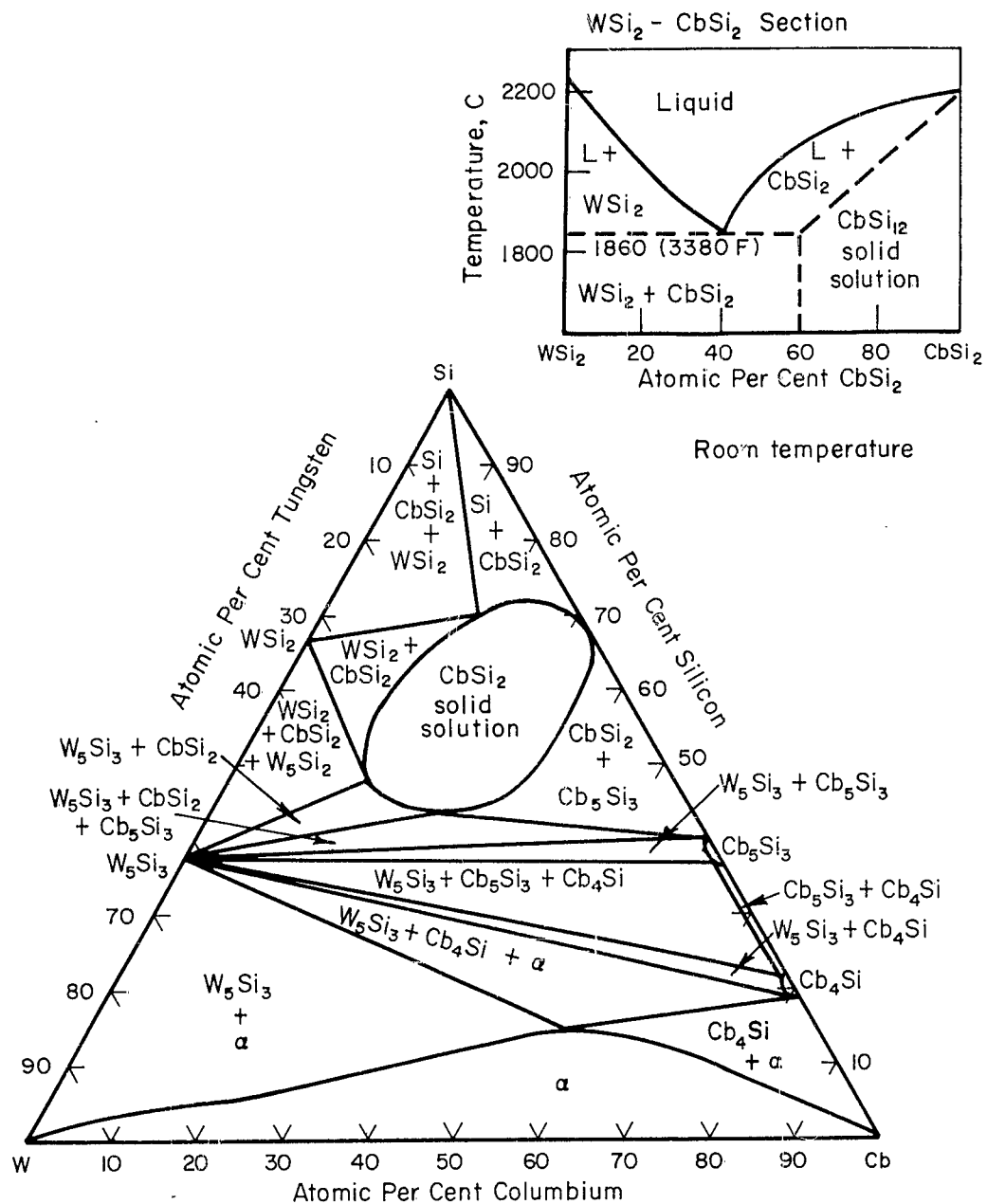
COLUMBIUM-TITANIUM-ZIRCONIUM SYSTEM (MELTING TEMPERATURES, C)⁽²⁸²⁾



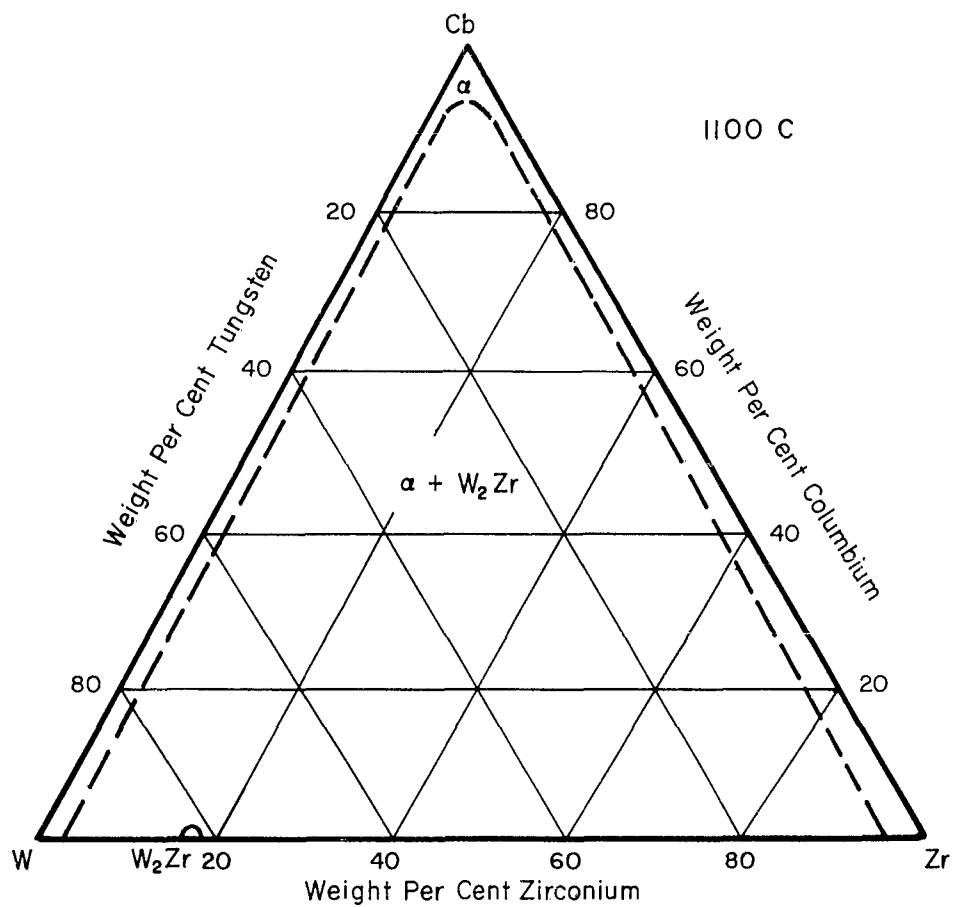
COLUMBIUM-URANIUM-ZIRCONIUM SYSTEM (1000 C)(280)



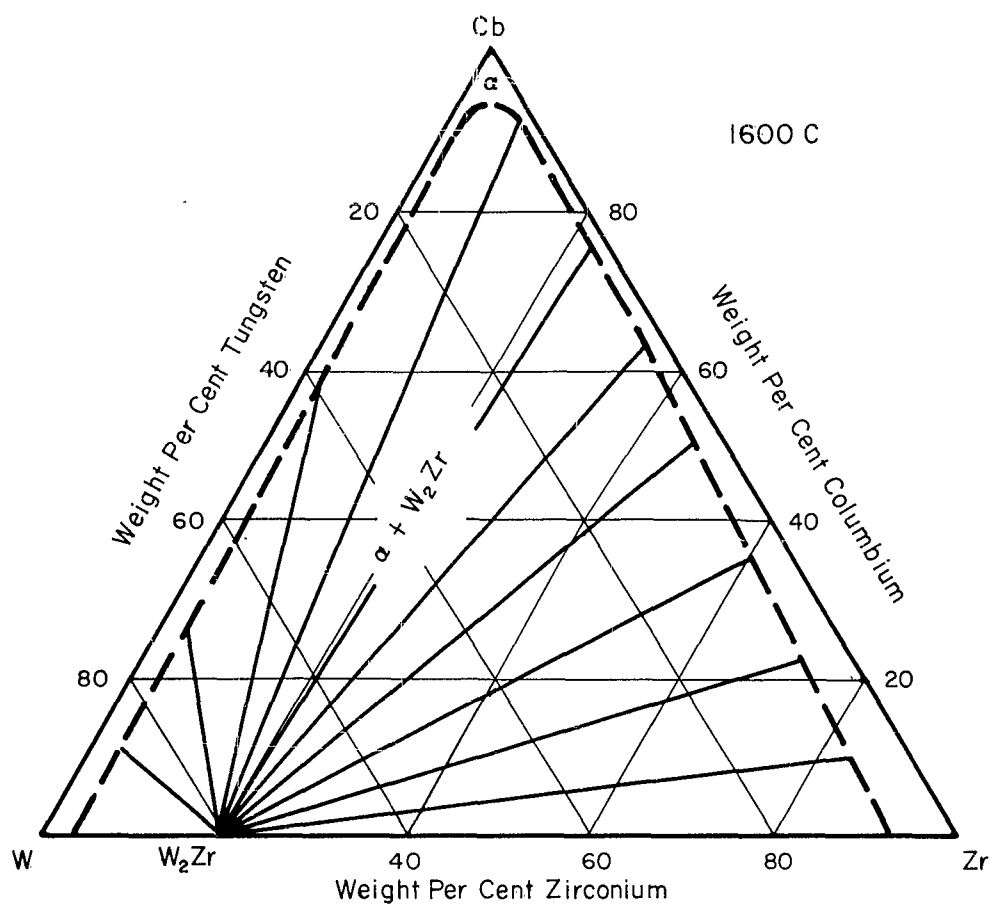
COLUMBIUM-TUNGSTEN-SILICON SYSTEM (ROOM TEMPERATURE)⁽³²¹⁾



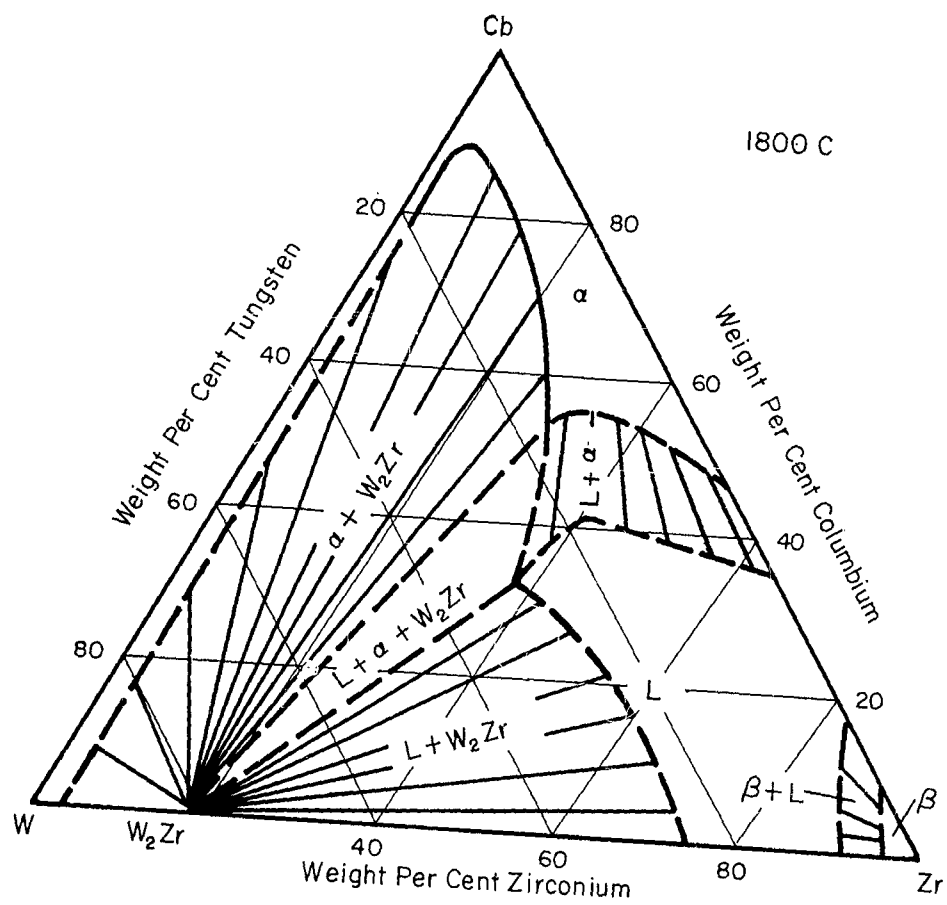
COLUMBIUM-TUNGSTEN-ZIRCONIUM SYSTEM (1100 C)⁽³¹⁹⁾



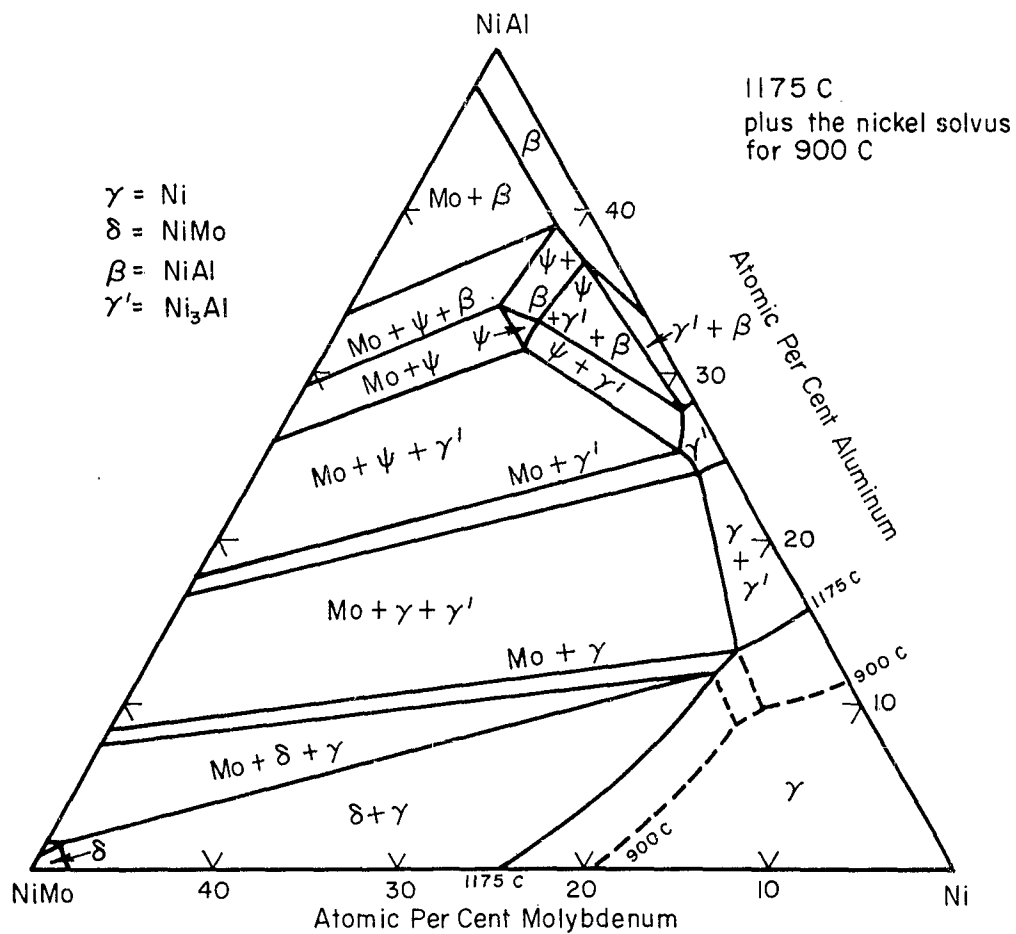
COLUMBIUM-TUNGSTEN-ZIRCONIUM SYSTEM (1600 C)⁽³¹⁹⁾



COLUMBIUM-TUNGSTEN-ZIRCONIUM SYSTEM (1800 C)(319)



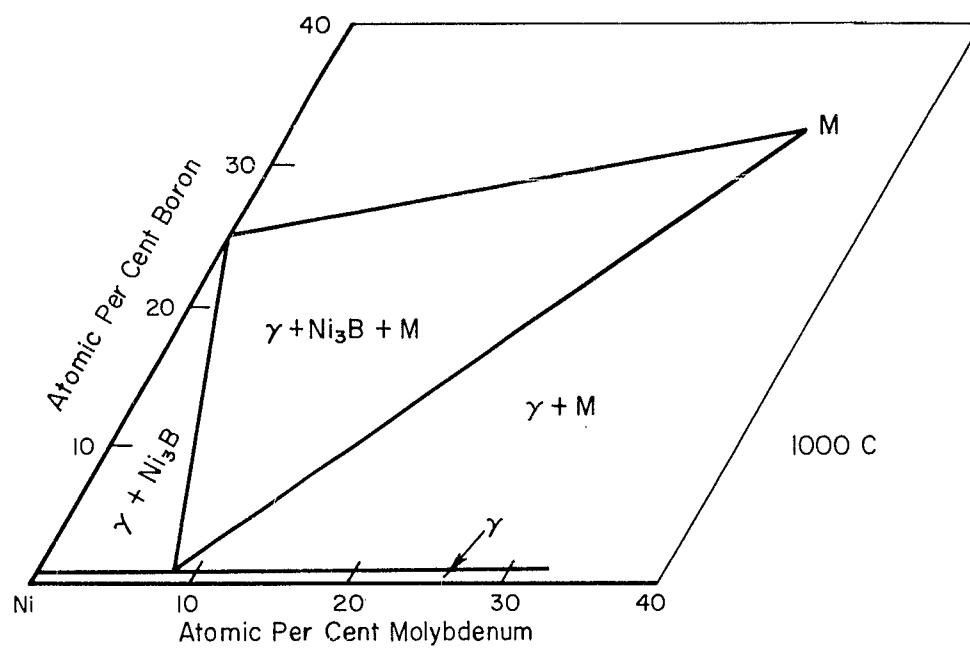
MOLYBDENUM-ALUMINUM-NICKEL SYSTEM (1175 C)(283)



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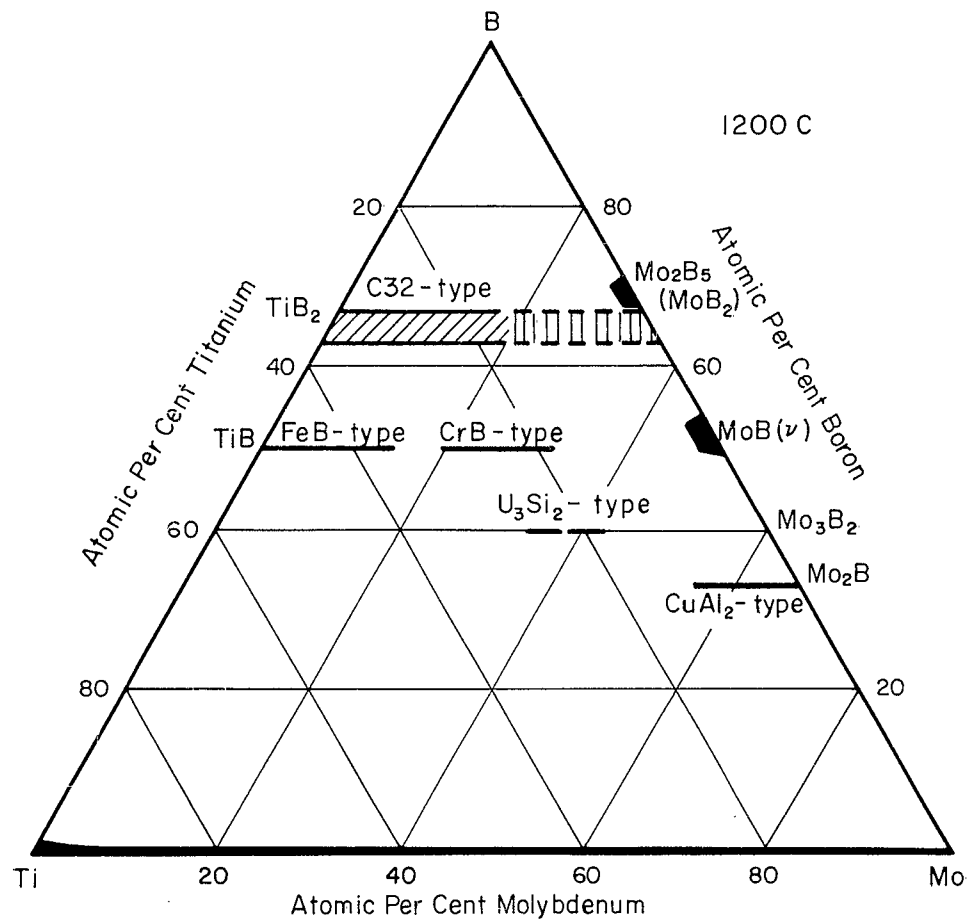


MOLYBDENUM-BORON-NICKEL SYSTEM (1000 C)⁽²⁸⁴⁾

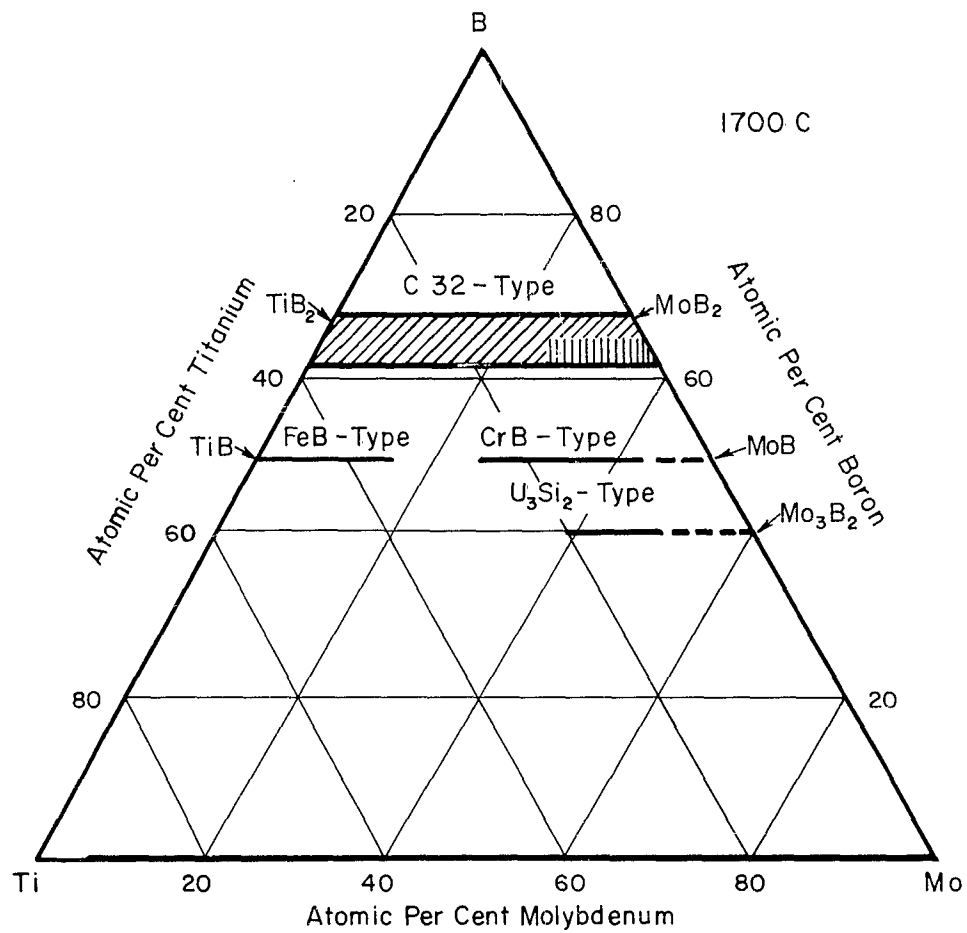


M = NiMoB

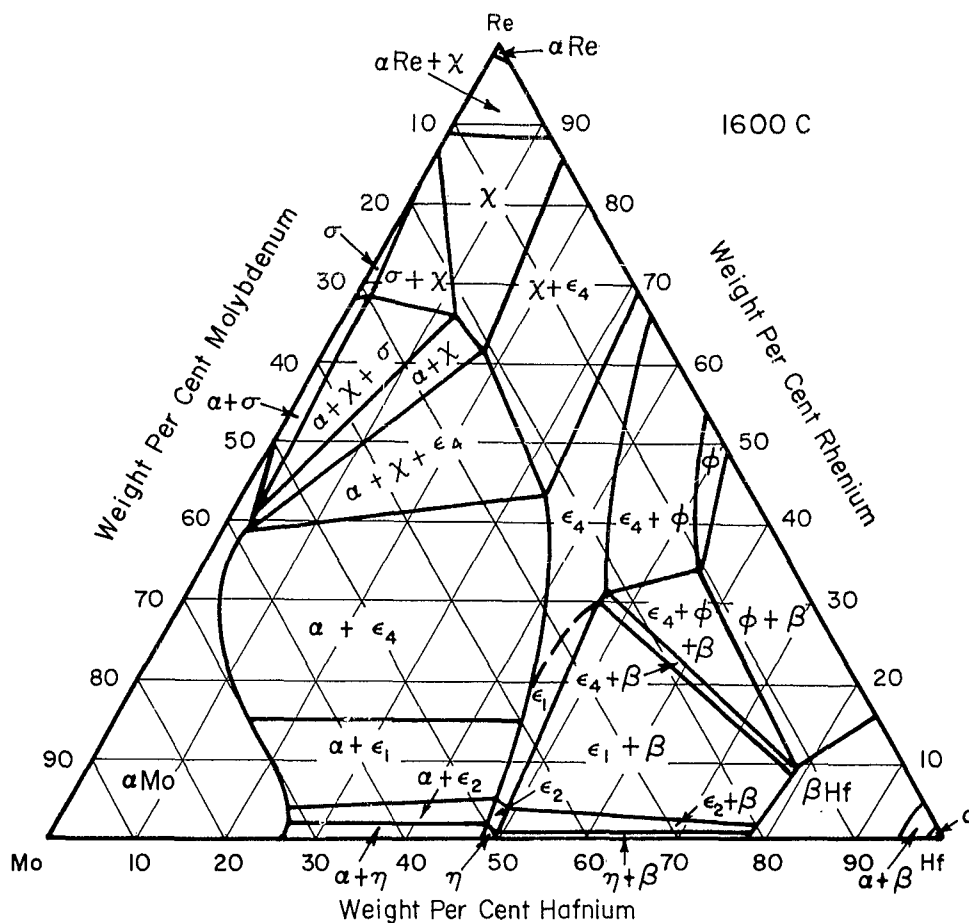
MOLYBDENUM-BORON-TITANIUM SYSTEM (1200 C)(285)



MOLYBDENUM-BORON-TITANIUM SYSTEM (1700 C)(285)

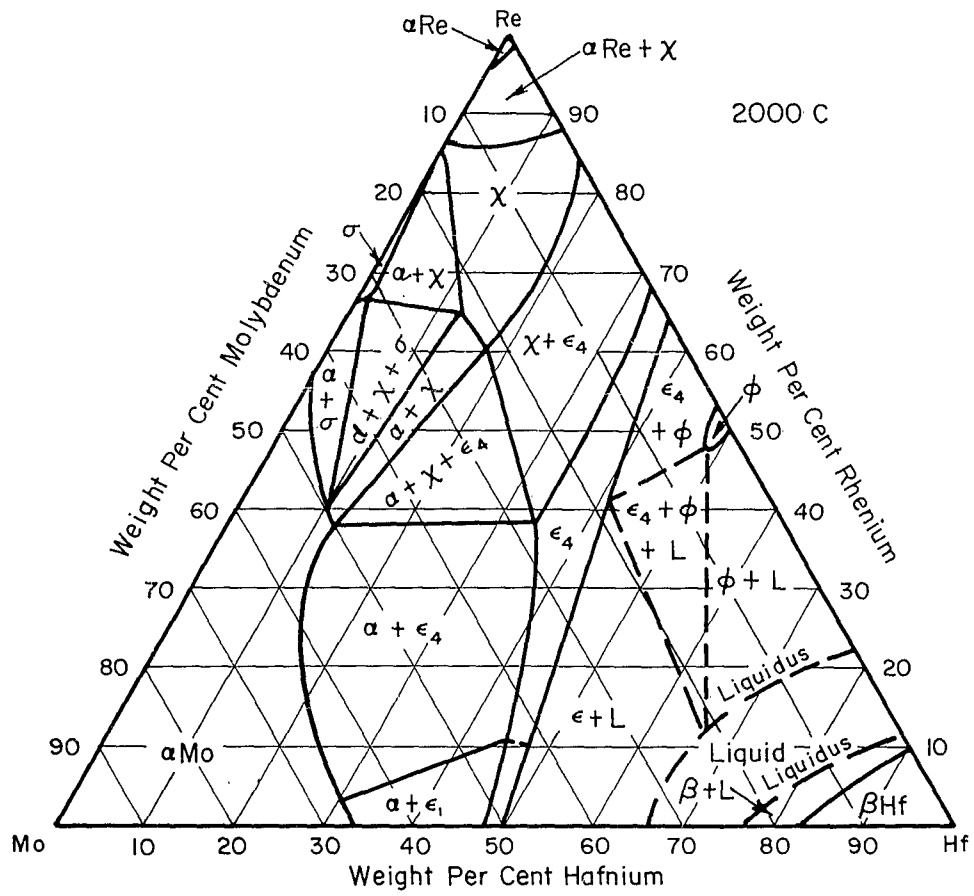


MOLYBDENUM-HAFNIUM-RHENIUM SYSTEM (1600 C)(250)

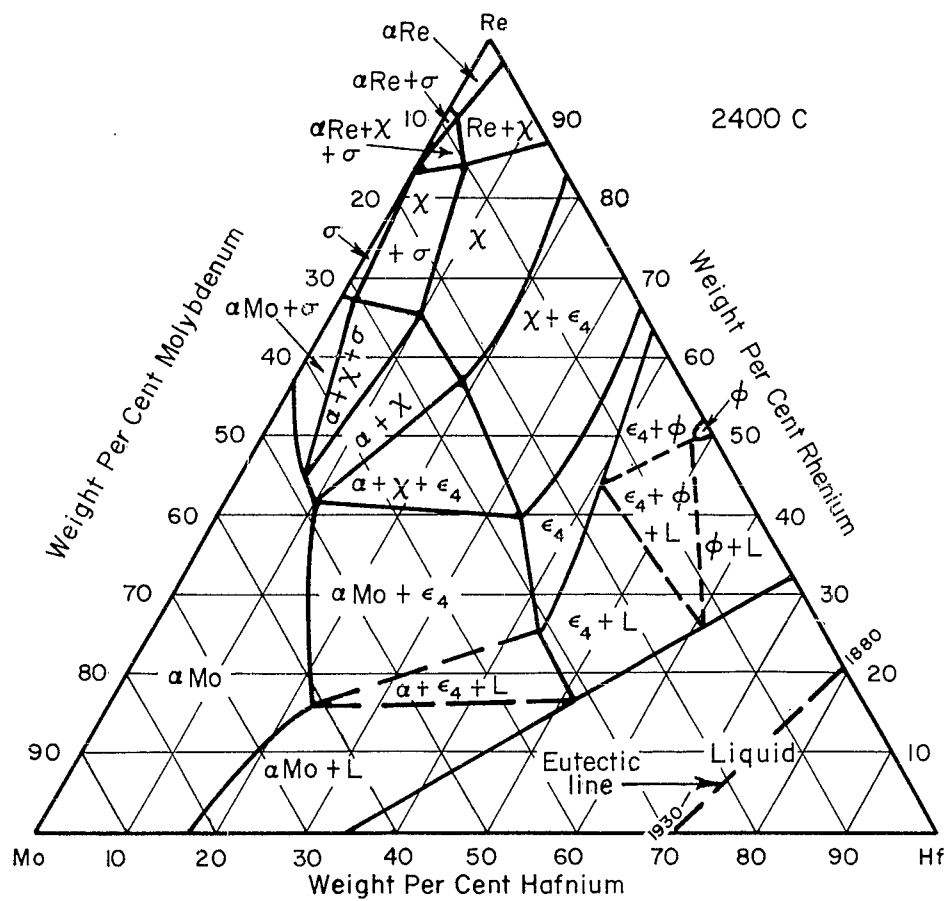


The phase ϕ has a tetragonal structure with $a = 8.90 \text{ \AA}$ and $c = 13.97 \text{ \AA}$; this phase forms peritectically at 2745 C and about 47.5 per cent hafnium in the binary rhenium-hafnium system. An intermediate Laves phase ϵ_4 (based on Re_2Hf) has the C14-MgZn_{12} type of structure with $a = 5.239 \text{ \AA}$, $c = 8.584 \text{ \AA}$, and $c/a = 1.638$.(250)

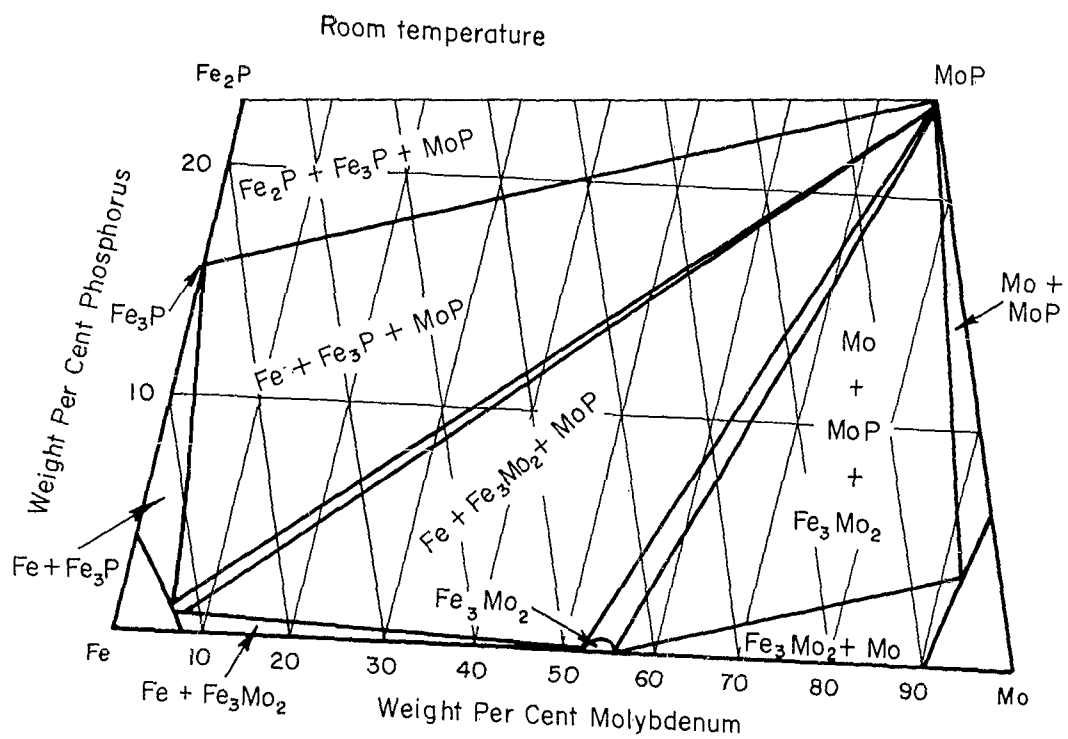
MOLYBDENUM-HAFNIUM-RHENIUM SYSTEM (2000 C)(250)



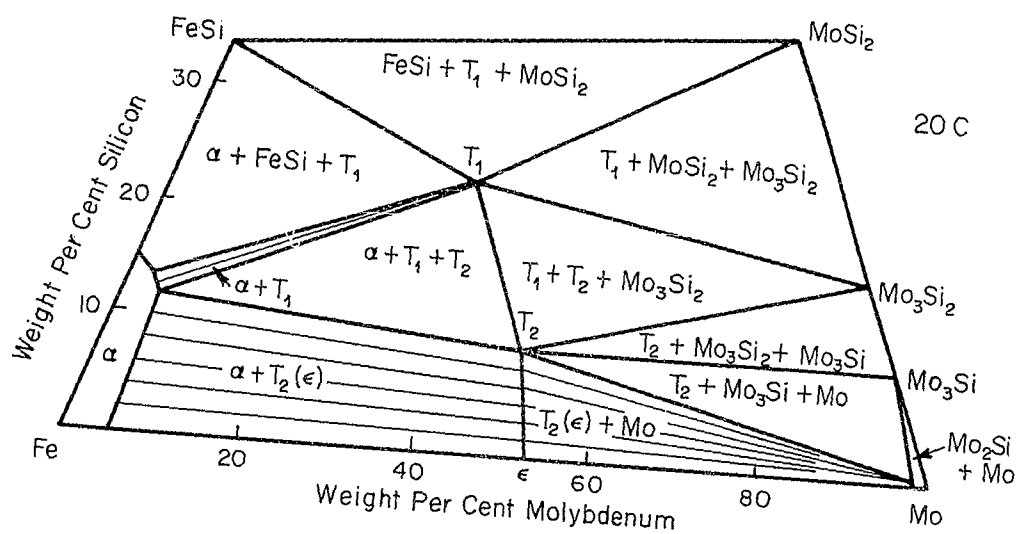
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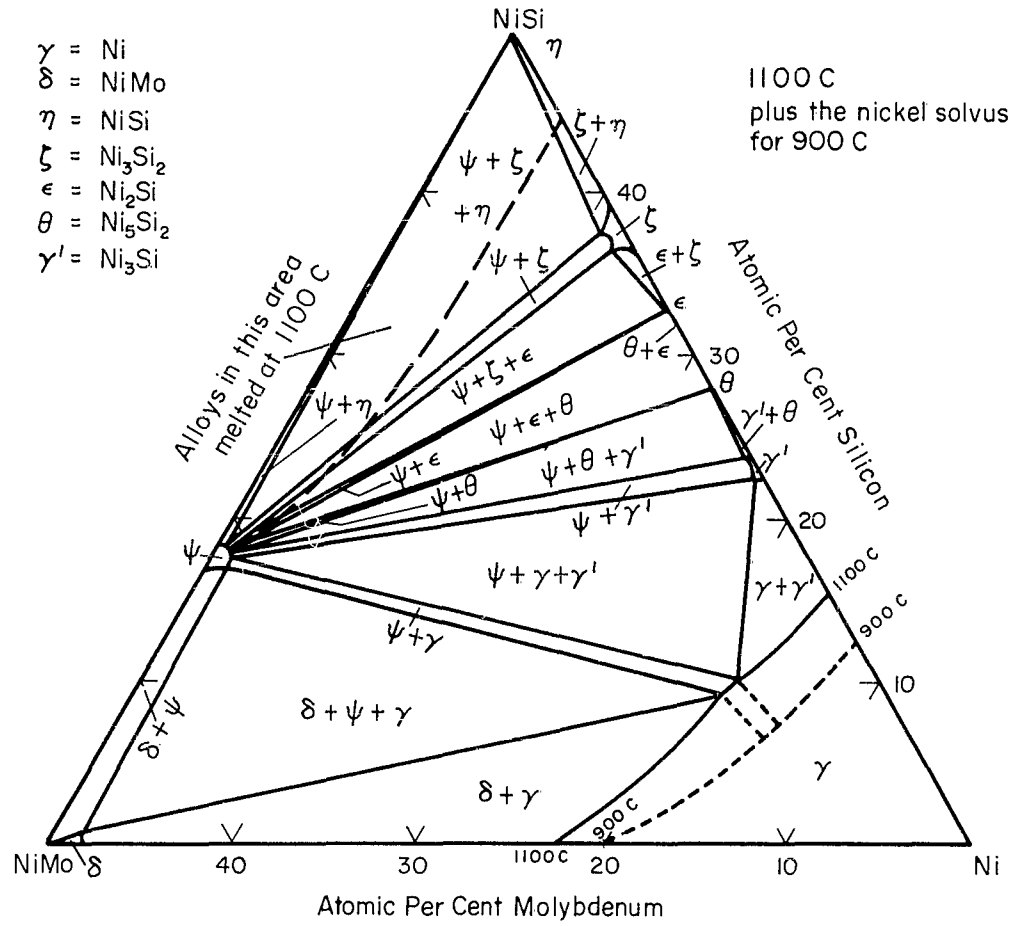
MOLYBDENUM-IRON-PHOSPHORUS SYSTEM (ROOM TEMPERATURE)⁽³¹³⁾



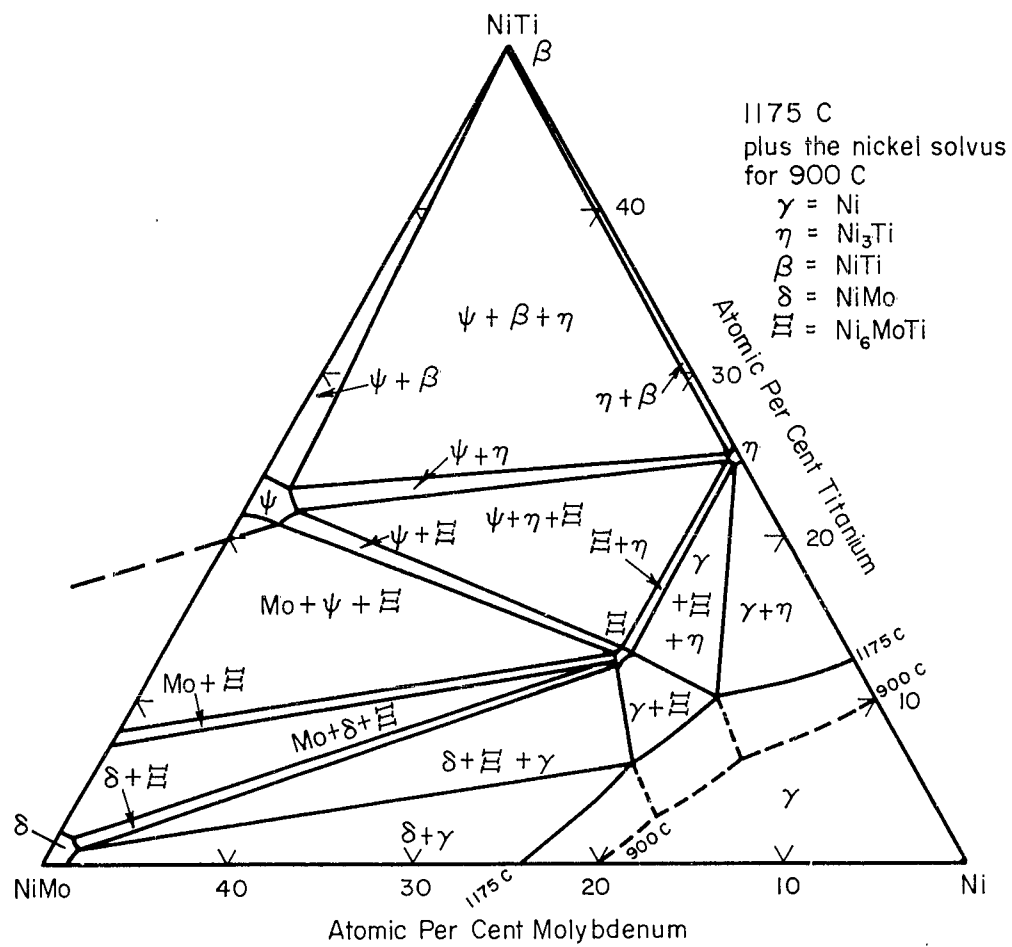
MOLYBDENUM-IRON-SILICON SYSTEM (20 C)(286)



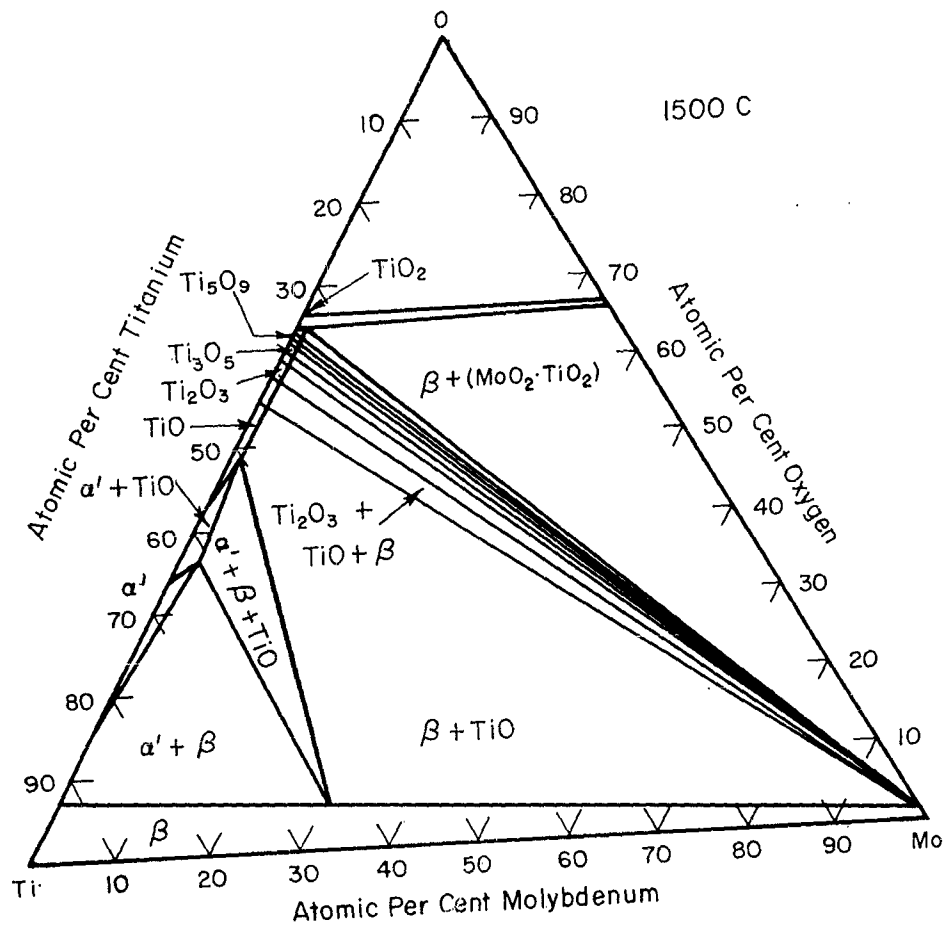
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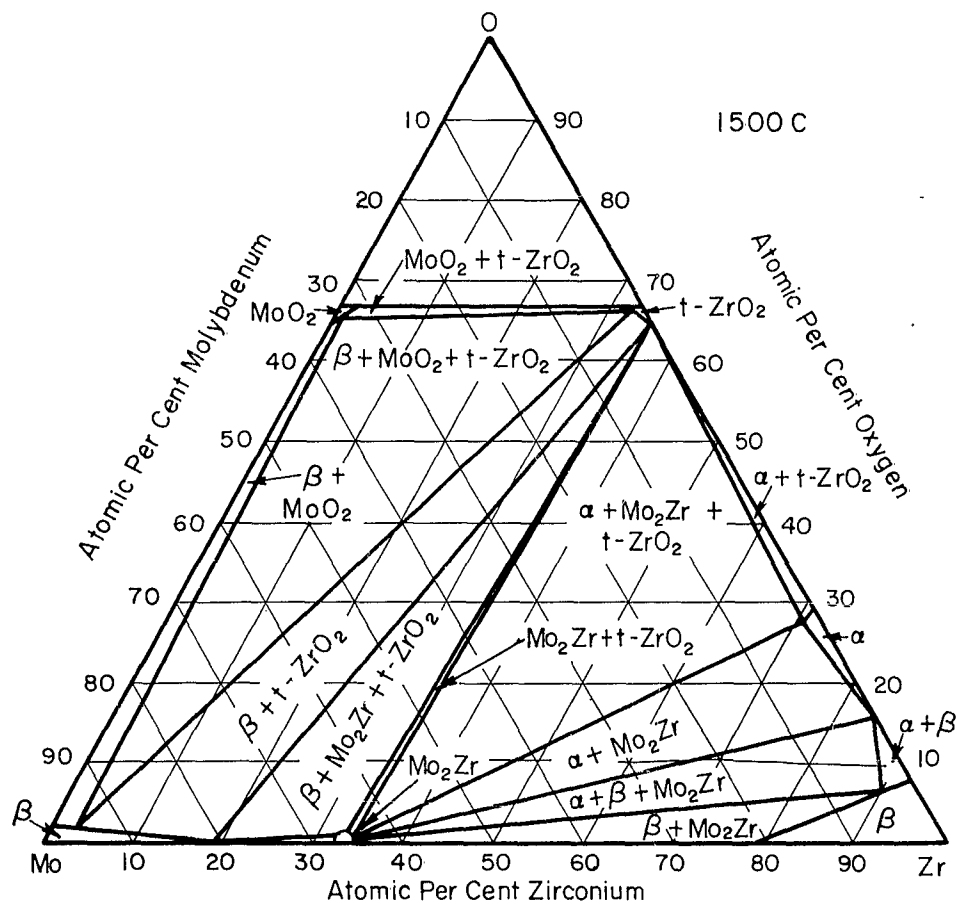
MOLYBDENUM-NICKEL-TITANIUM SYSTEM (1175 C)⁽³¹²⁾



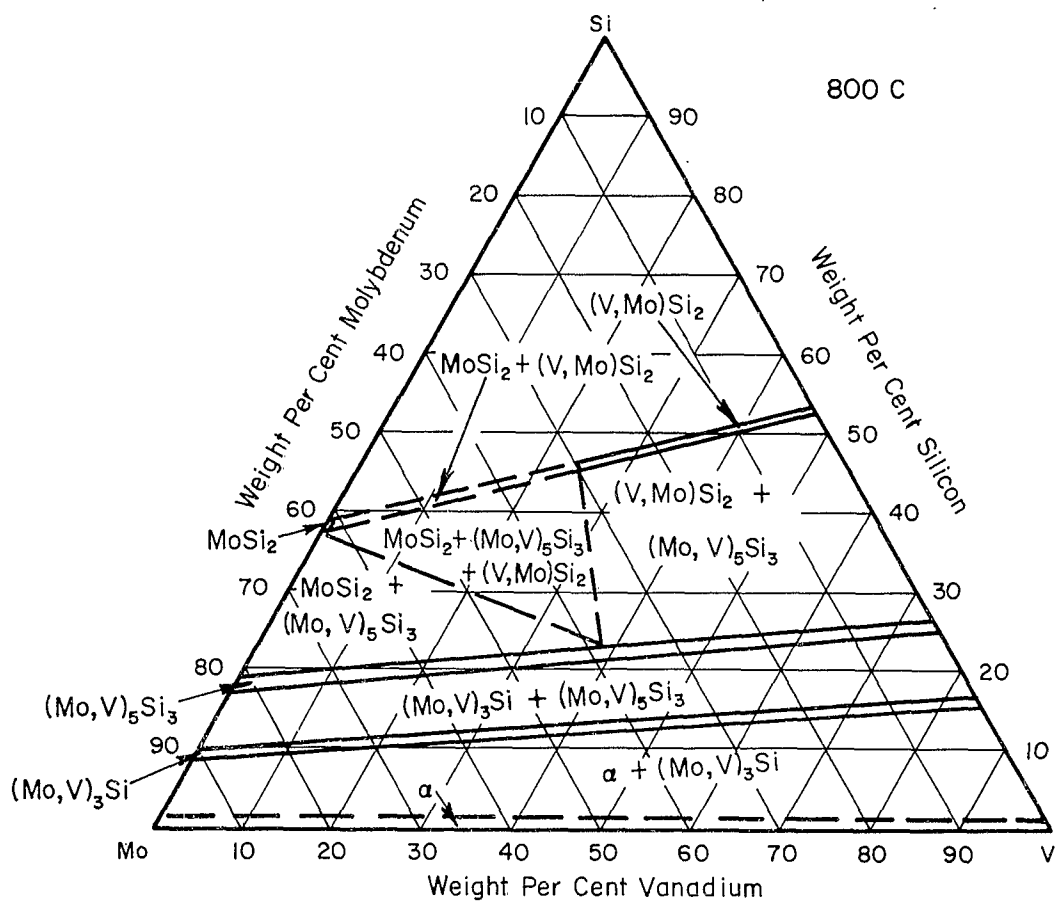
MOLYBDENUM-OXYGEN-TITANIUM SYSTEM (1500 C)⁽³¹⁴⁾



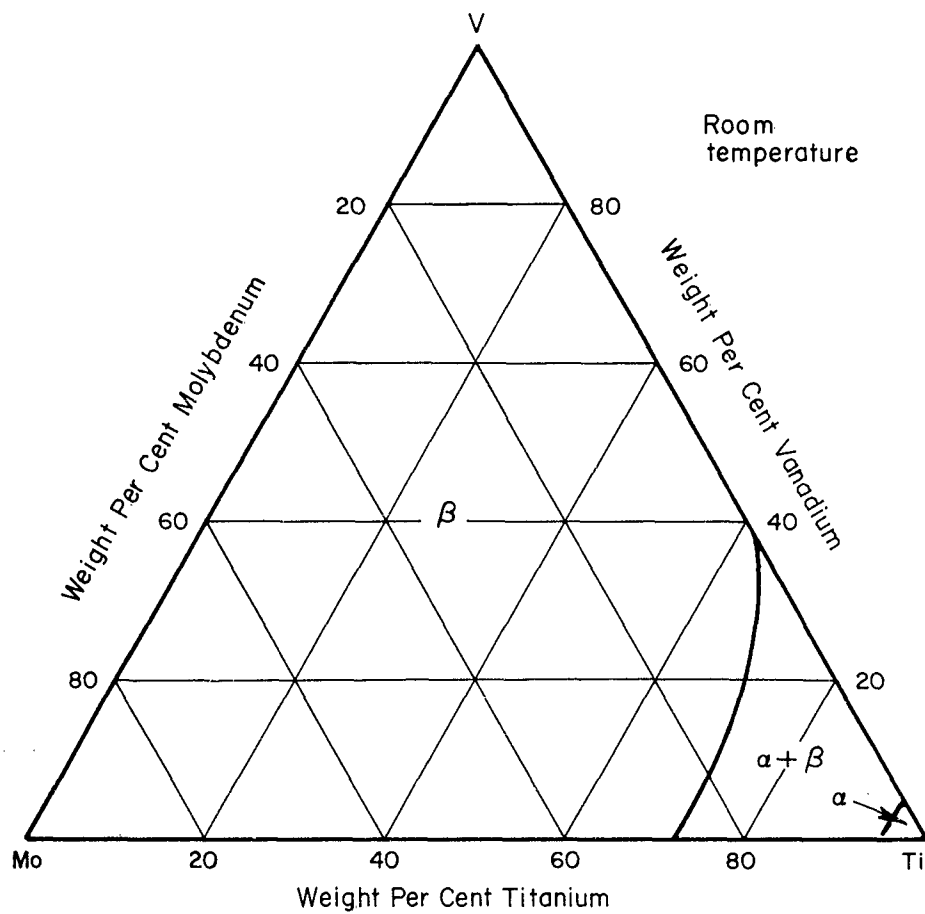
MOLYBDENUM-OXYGEN-ZIRCONIUM SYSTEM (1500 C)⁽³¹⁴⁾



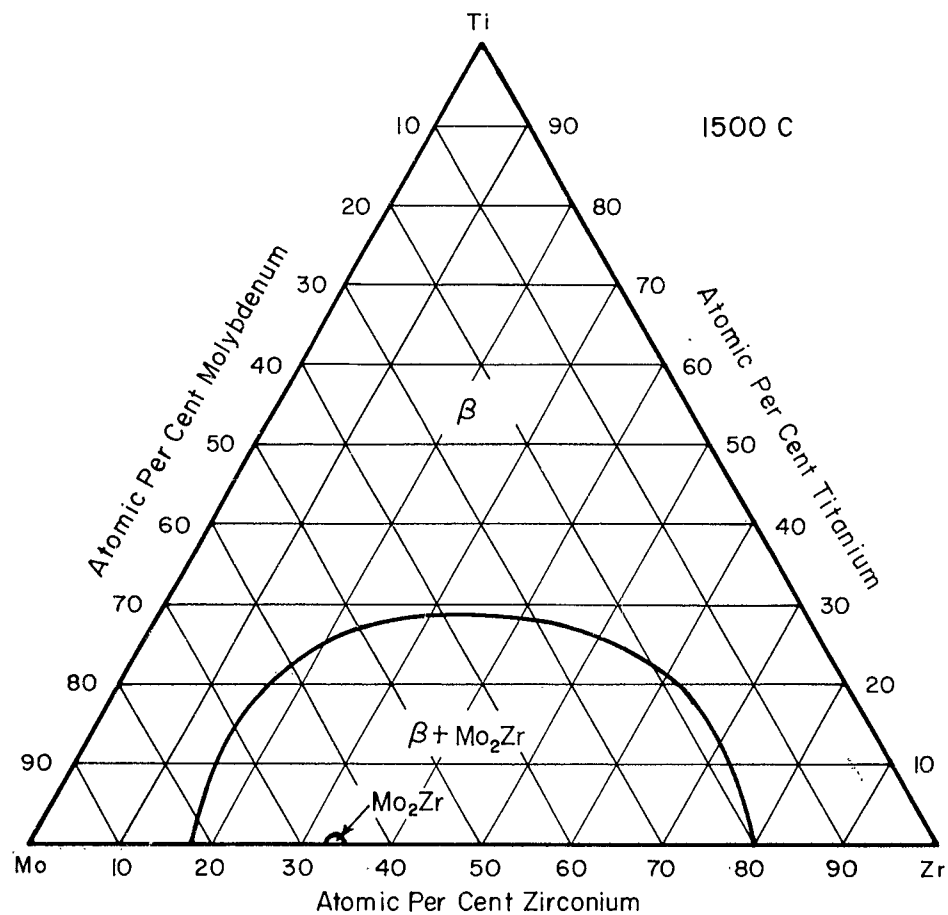
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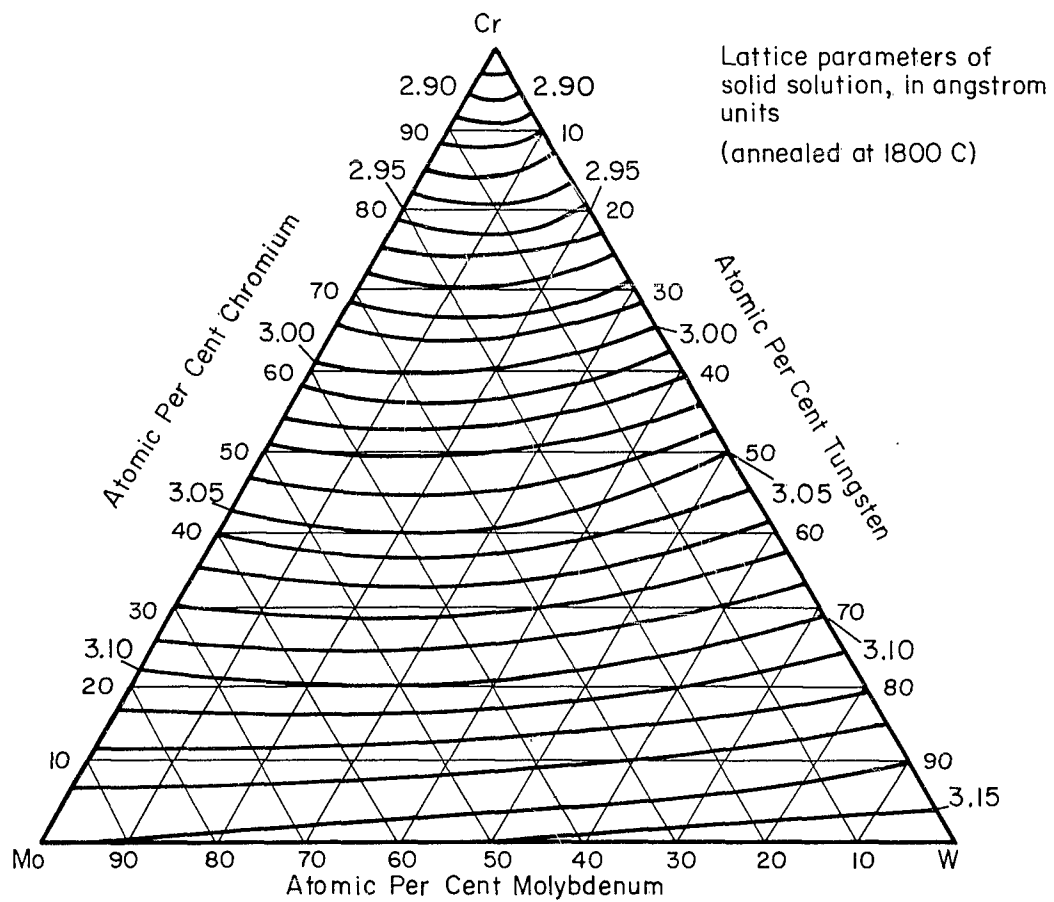
MOLYBDENUM-TITANIUM-VANADIUM SYSTEM (ROOM TEMPERATURE)⁽²⁸⁷⁾



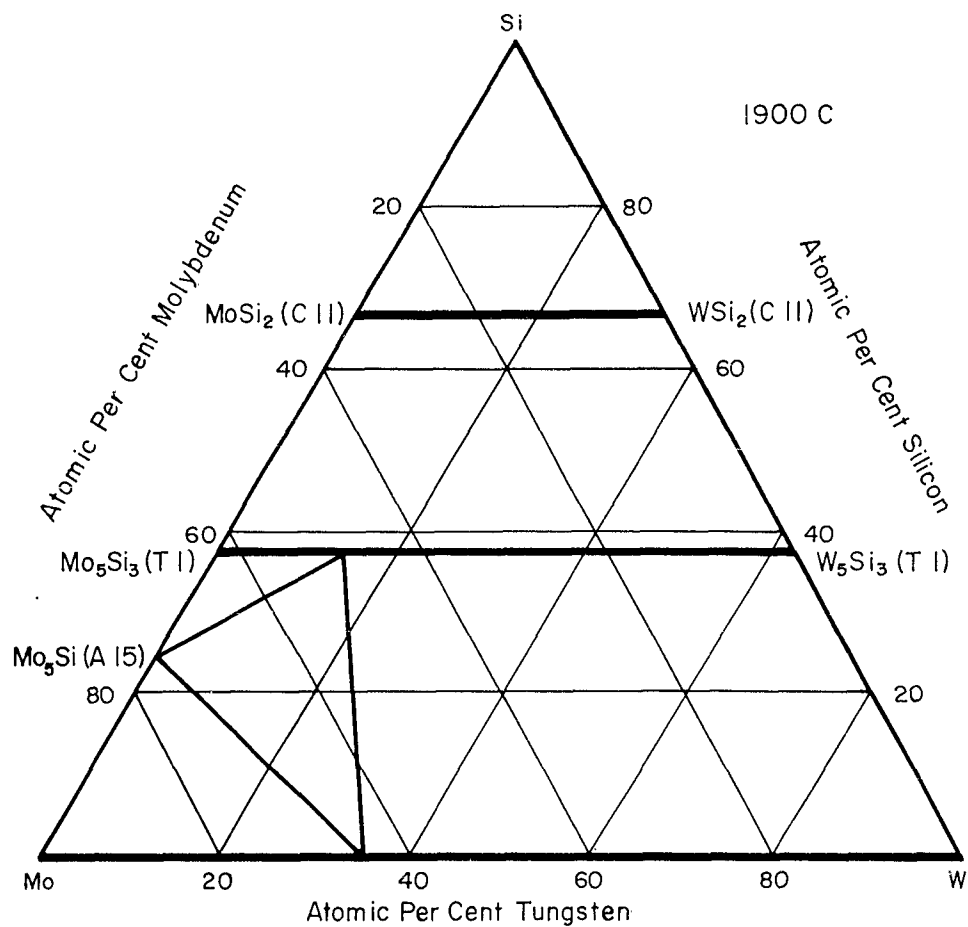
MOLYBDENUM-TITANIUM-ZIRCONIUM SYSTEM (1500 C)(314)



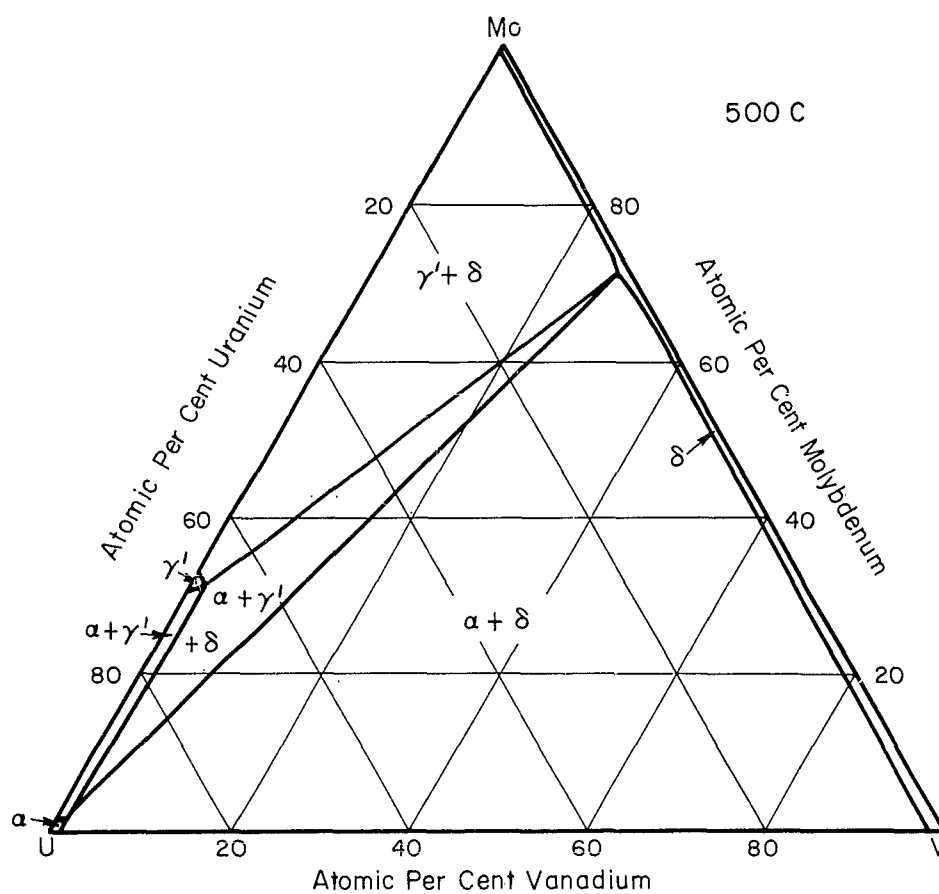
MOLYBDENUM-TUNGSTEN-CHROMIUM SYSTEM (1800 C)(288)



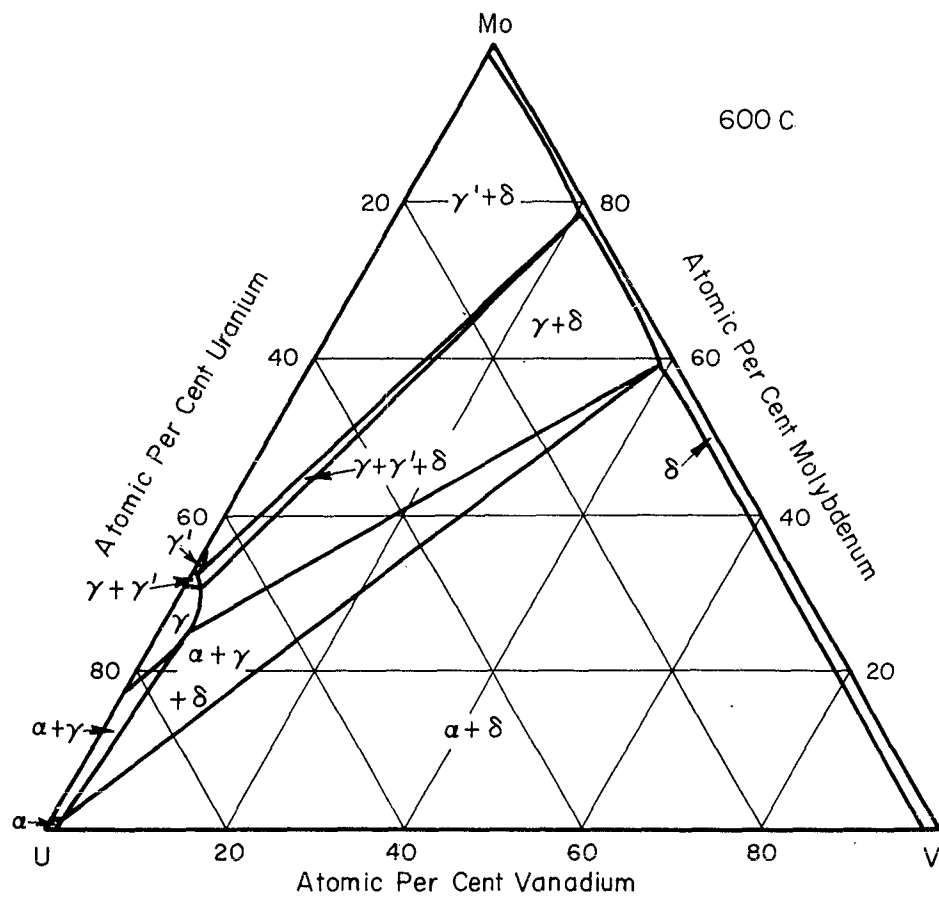
MOLYBDENUM-TUNGSTEN-SILICON SYSTEM (1900 C)⁽²⁹⁴⁾



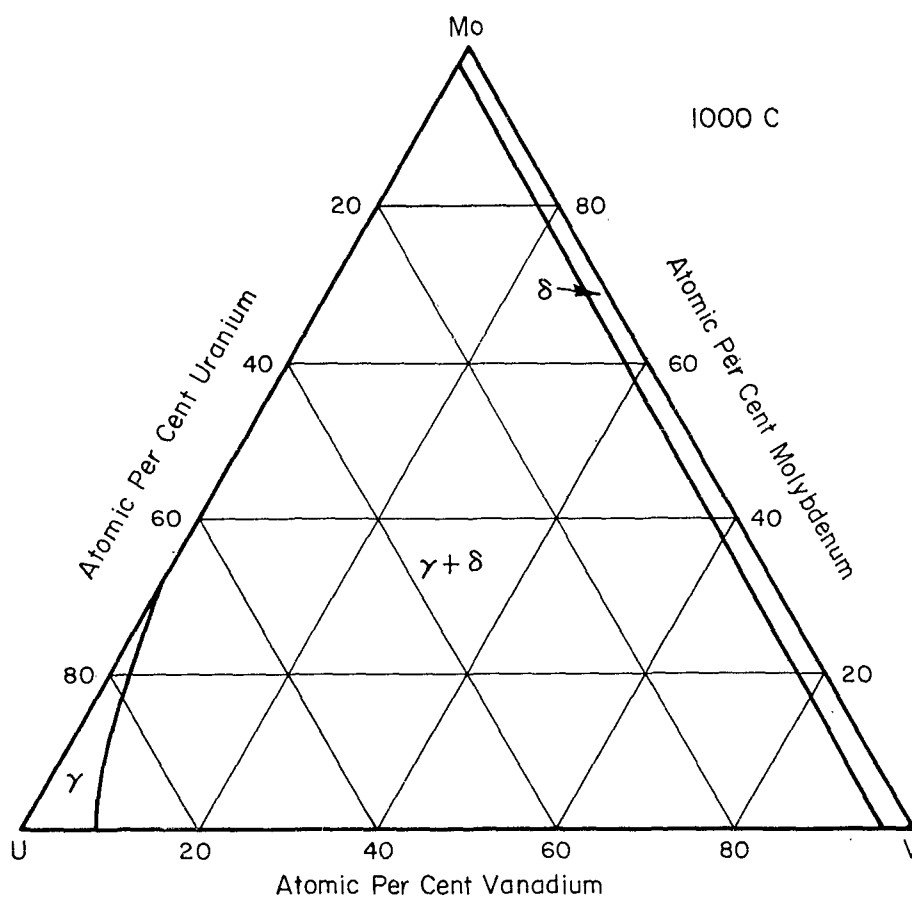
MOLYBDENUM-URANIUM-VANADIUM SYSTEM (500 C)⁽³⁰⁸⁾



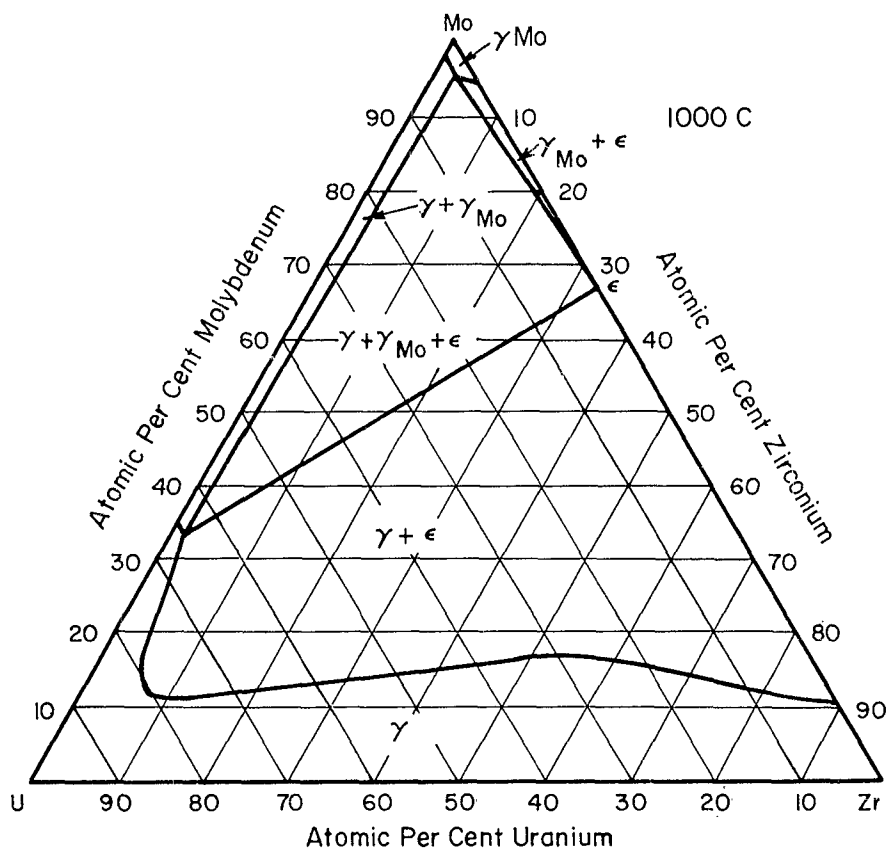
MOLYBDENUM-URANIUM-VANADIUM SYSTEM (600 C)(308)



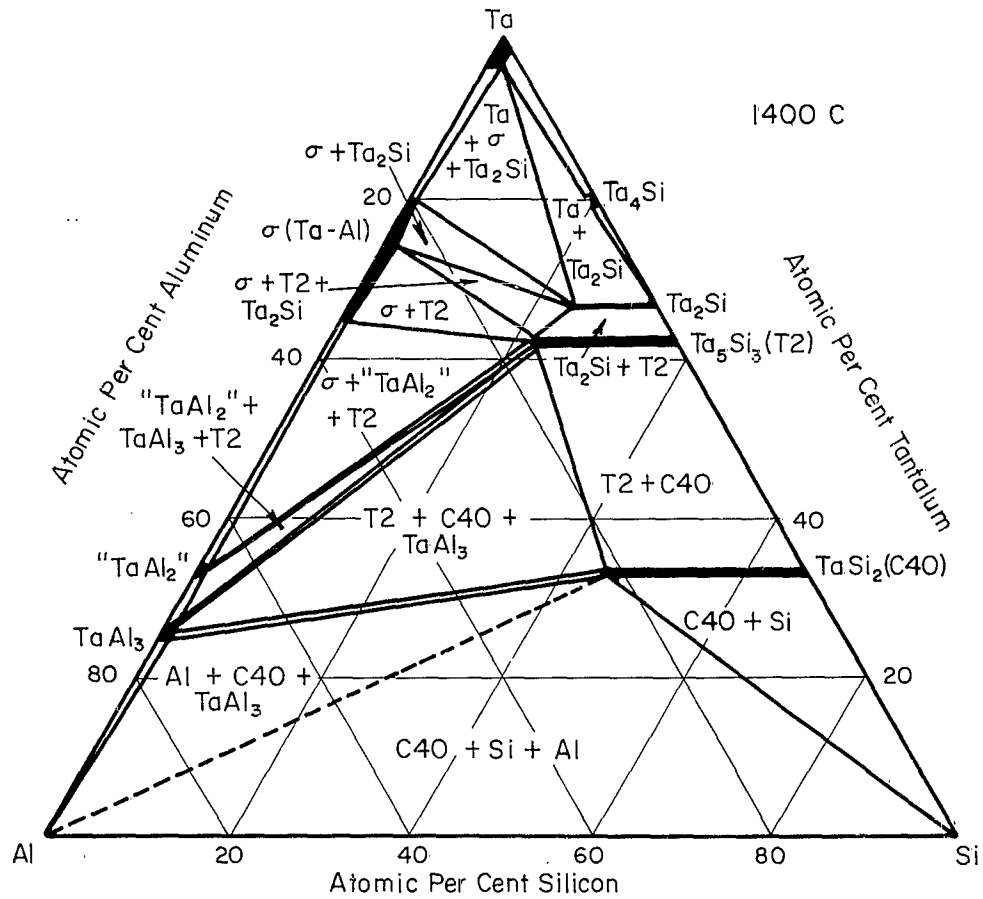
MOLYBDENUM-URANIUM-VANADIUM SYSTEM (1000 C)⁽³⁰⁸⁾



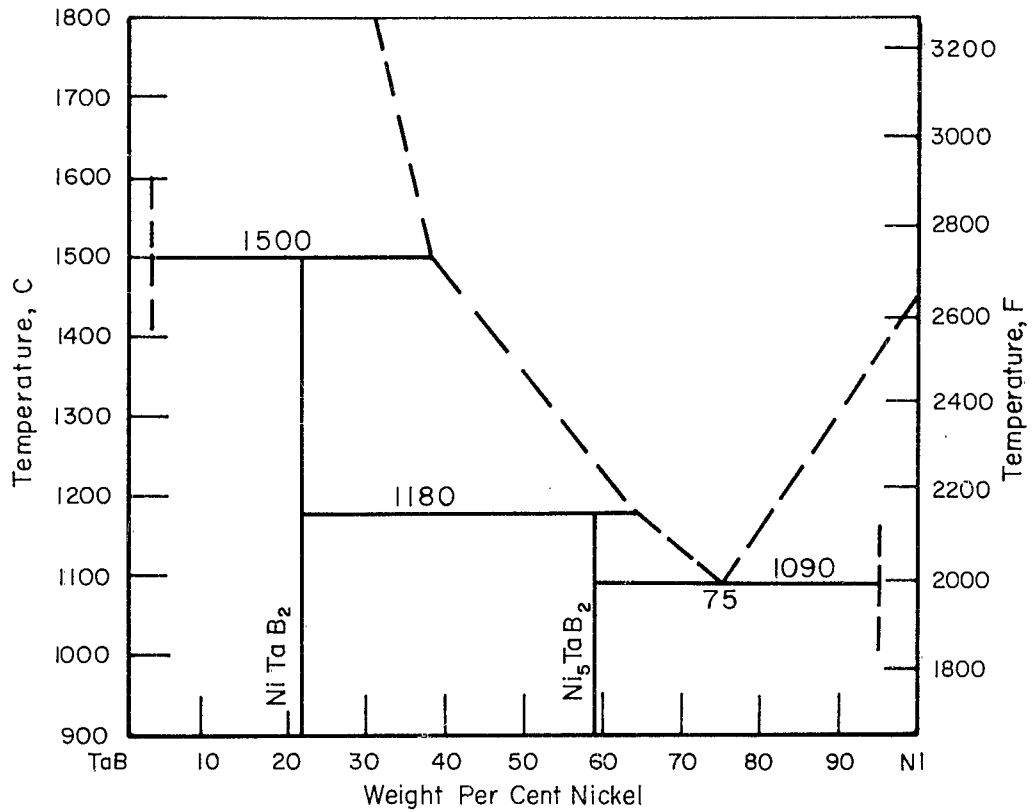
MOLYBDENUM-URANIUM-ZIRCONIUM SYSTEM (1000 C)⁽²⁸⁰⁾



TANTALUM-ALUMINUM-SILICON SYSTEM (1400 C)(266)

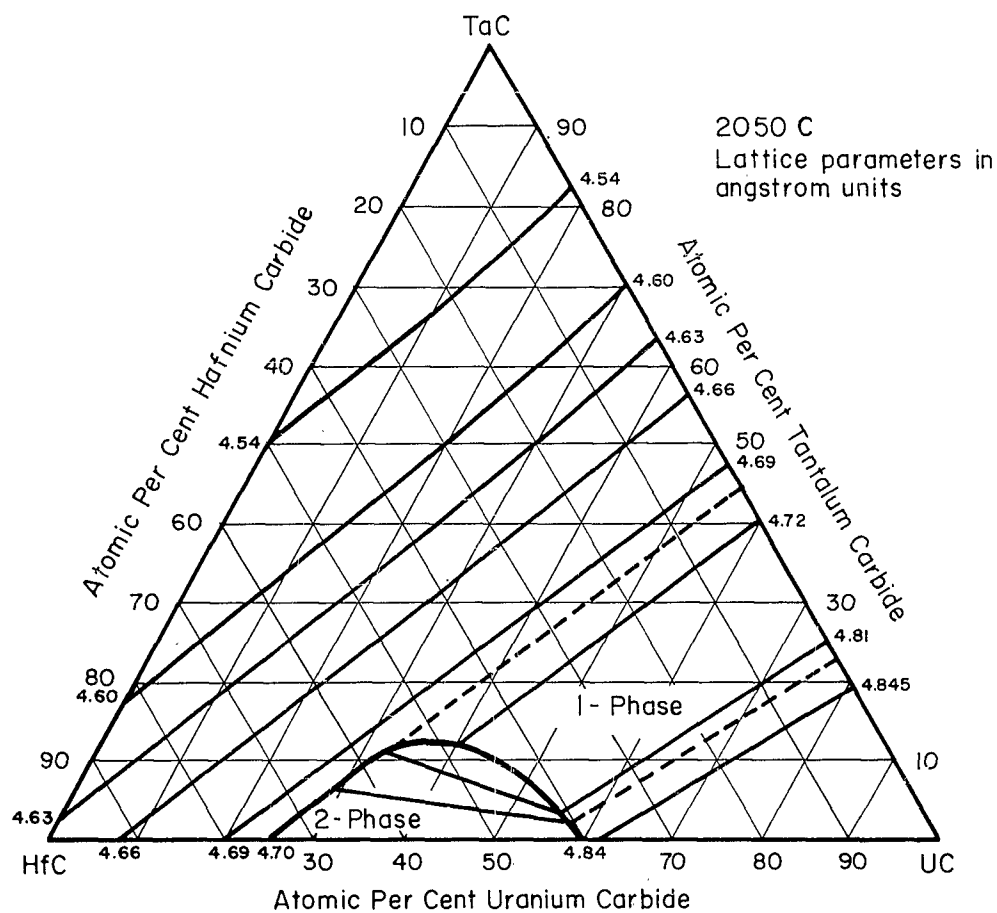


TANTALUM-BORON-NICKEL SYSTEM (TaB_2 -Ni)

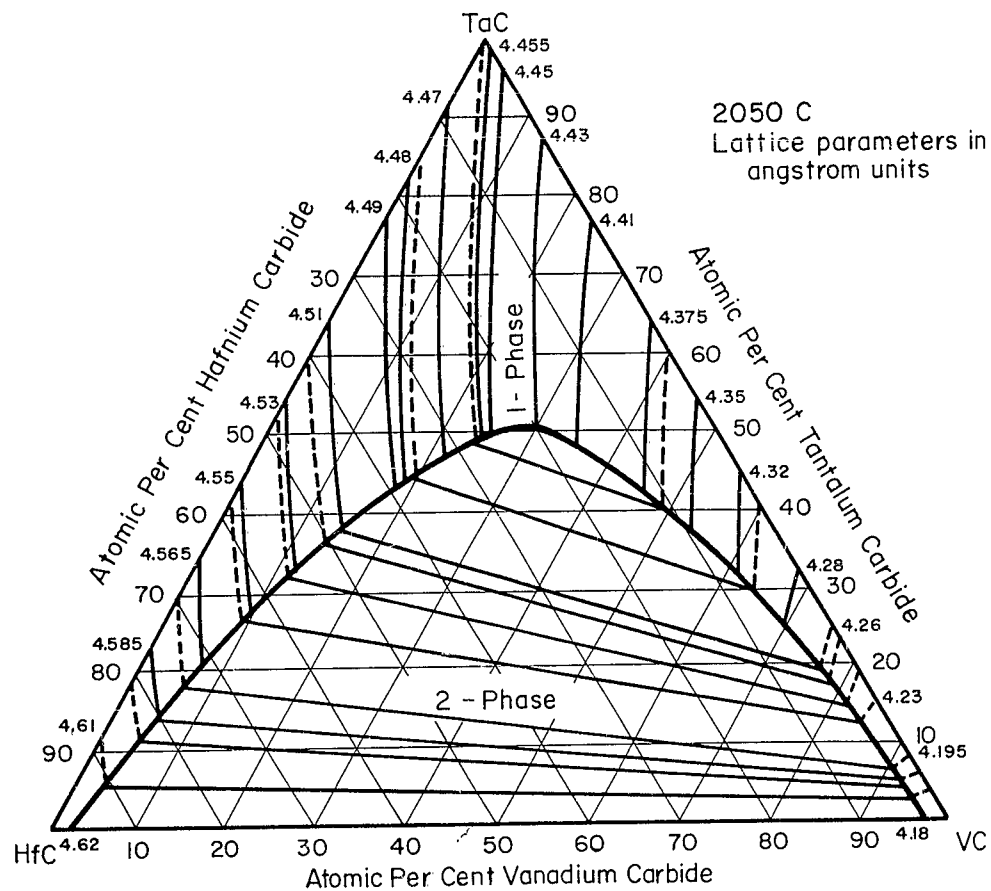


Ni_5TaB_2 has a face-centered cubic structure with $a = 10.56 \text{ \AA}$.⁽²⁸⁹⁾ The structure of NiTaB_2 was not identified.

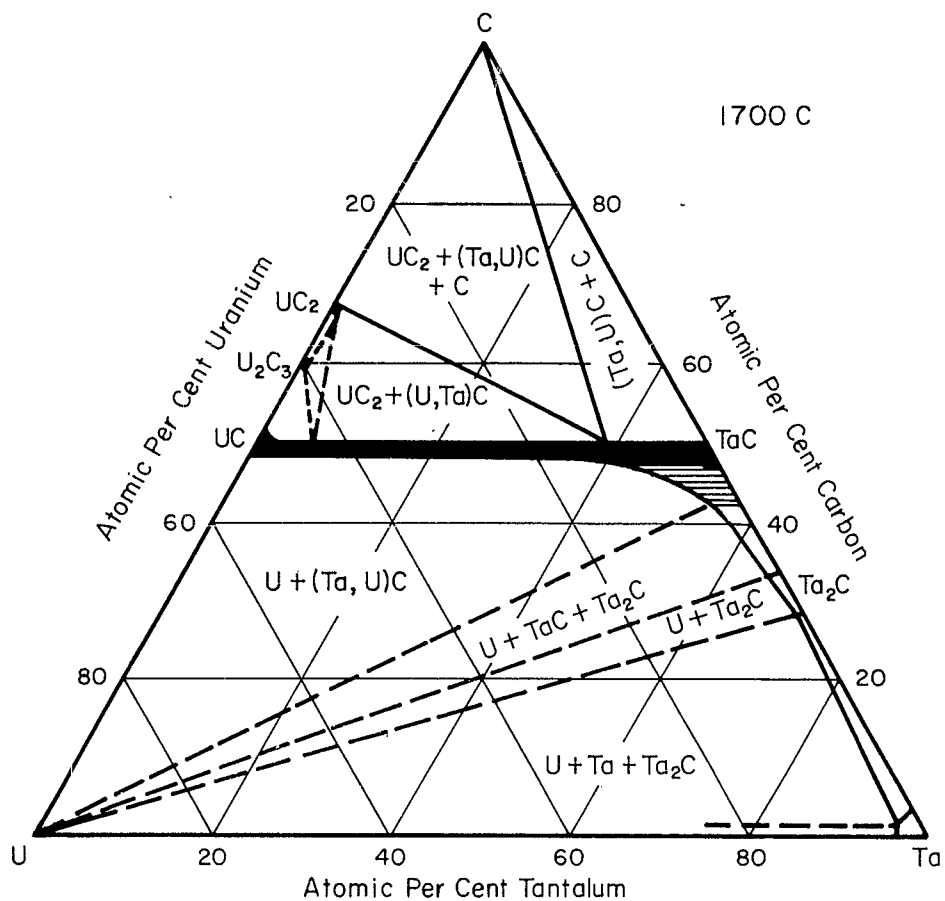
TANTALUM CARBIDE-HAFNIUM CARBIDE-URANIUM CARBIDE SYSTEM (2050 C)⁽²⁷¹⁾



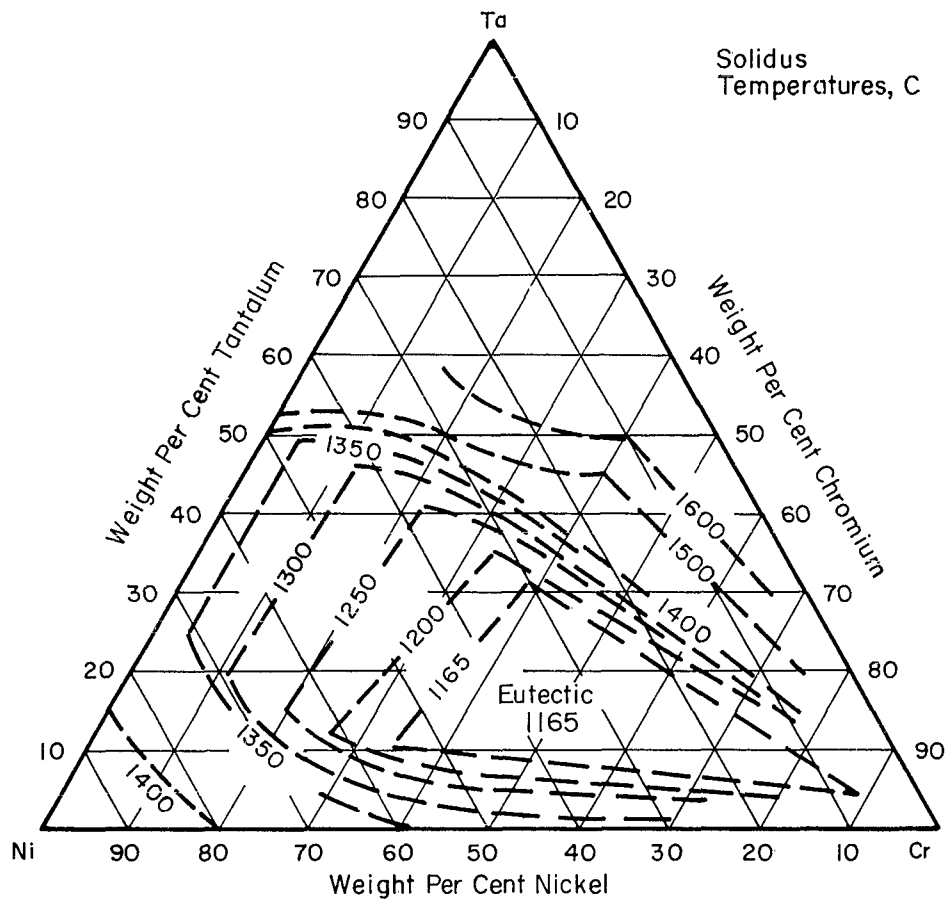
TANTALUM CARBIDE-HAFNIUM CARBIDE-VANADIUM CARBIDE SYSTEM (2050 C)⁽²⁷¹⁾



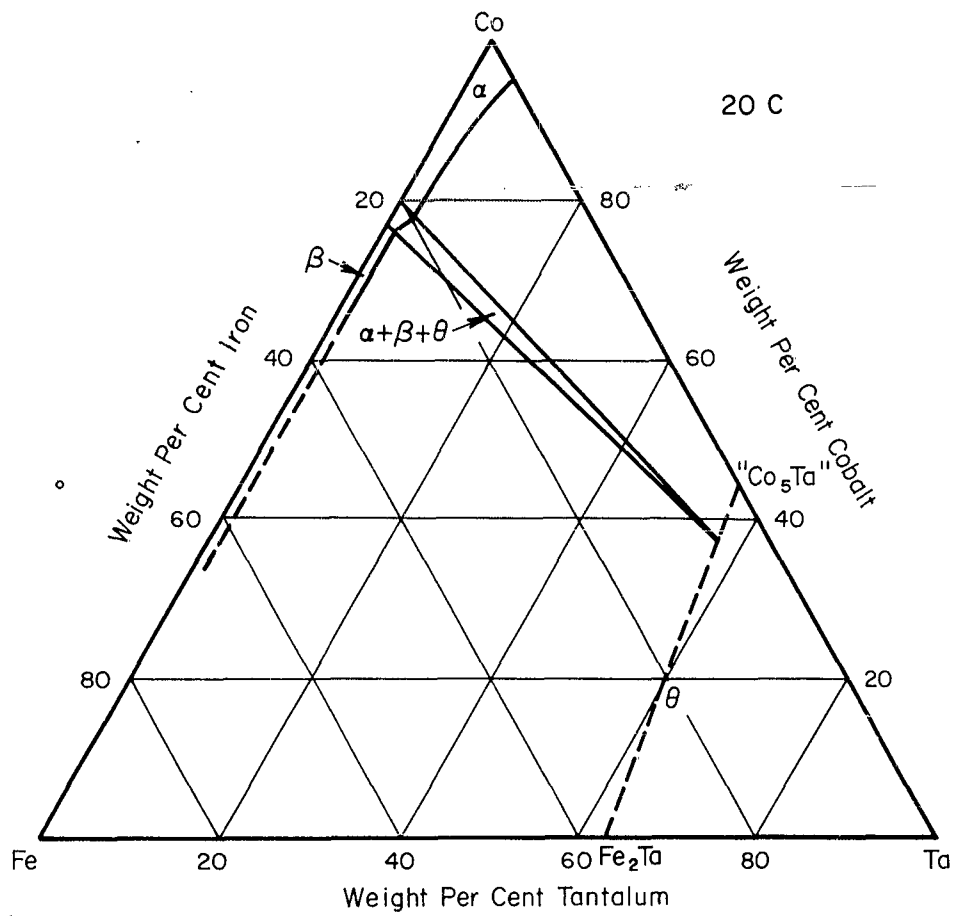
TANTALUM-CARBON-URANIUM SYSTEM (1700 C)⁽²⁷²⁾



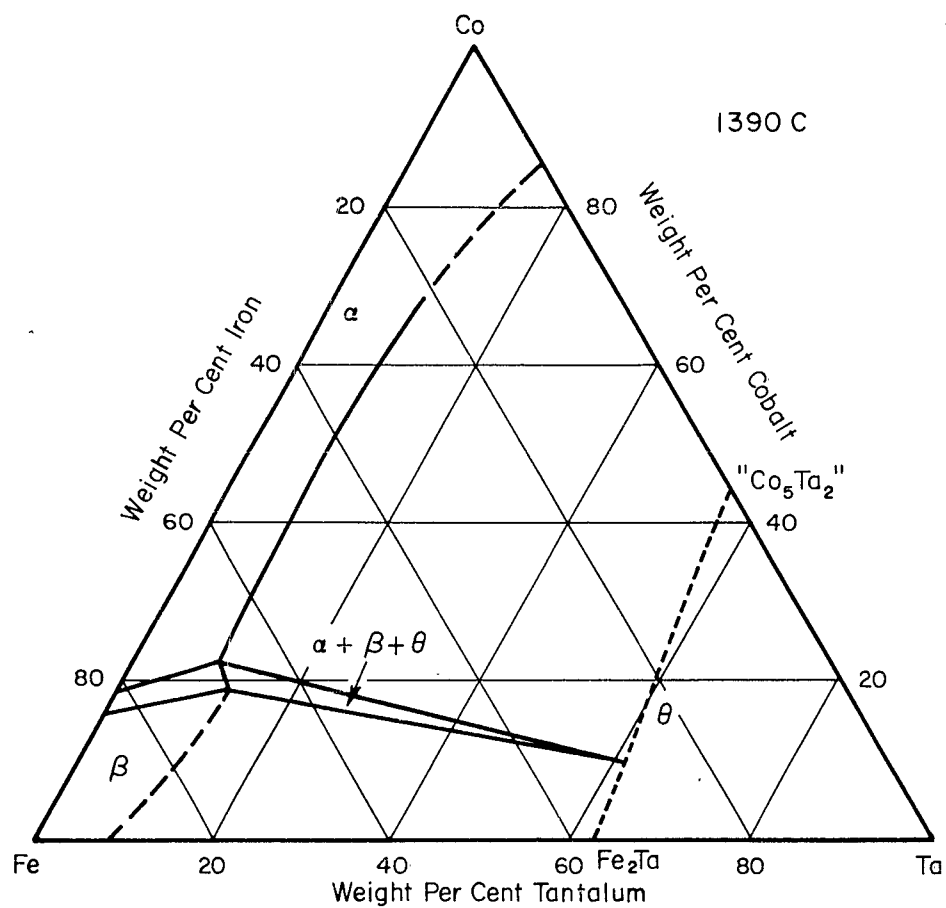
TANTALUM-CHROMIUM-NICKEL SYSTEM (SOLIDUS TEMPERATURE)⁽³²⁰⁾



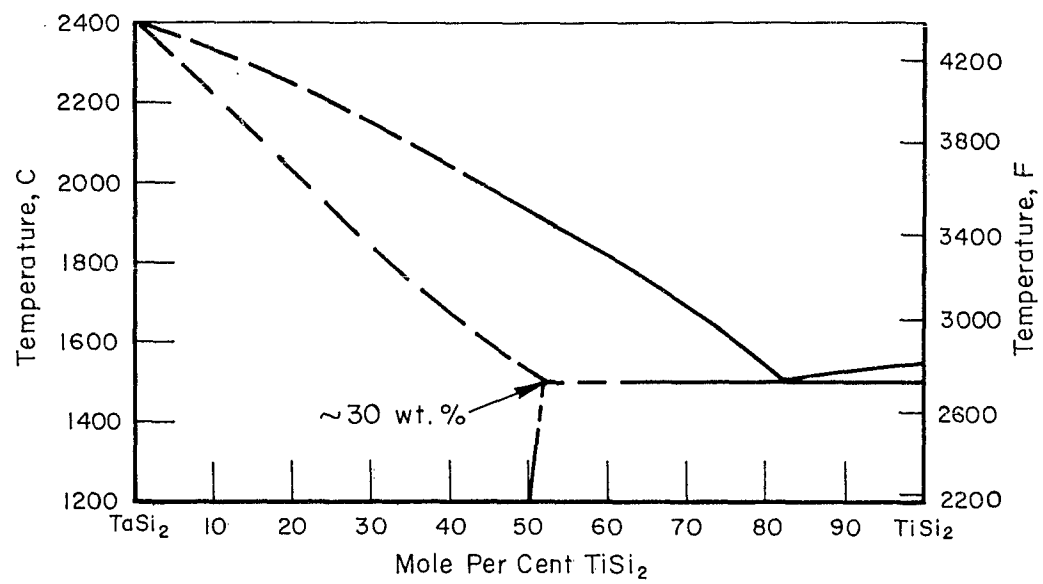
TANTALUM-COBALT-IRON SYSTEM (20 C)(290)



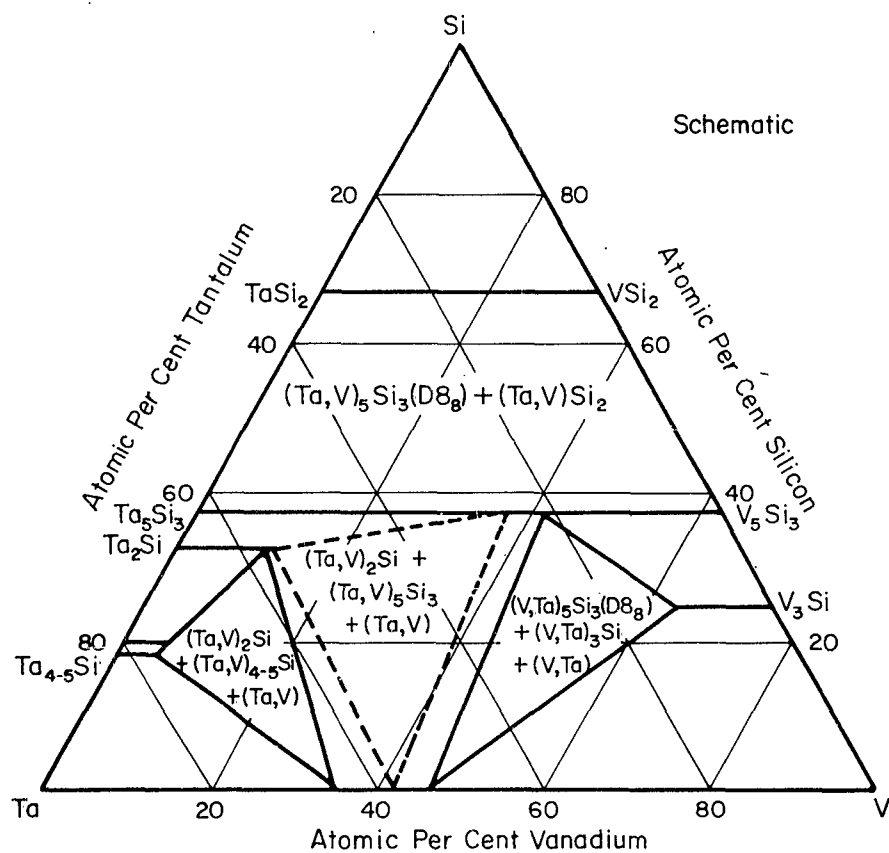
TANTALUM-COBALT-IRON SYSTEM (1390 C)(290)



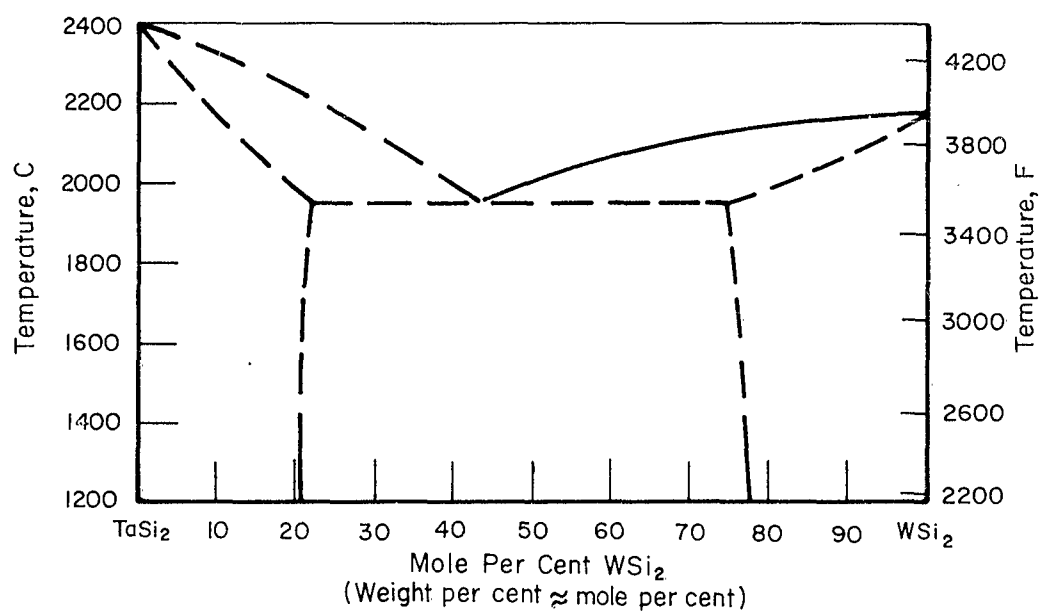
TANTALUM-SILICON-TITANIUM SYSTEM (TaSi_2 - TiSi_2)⁽³¹⁶⁾



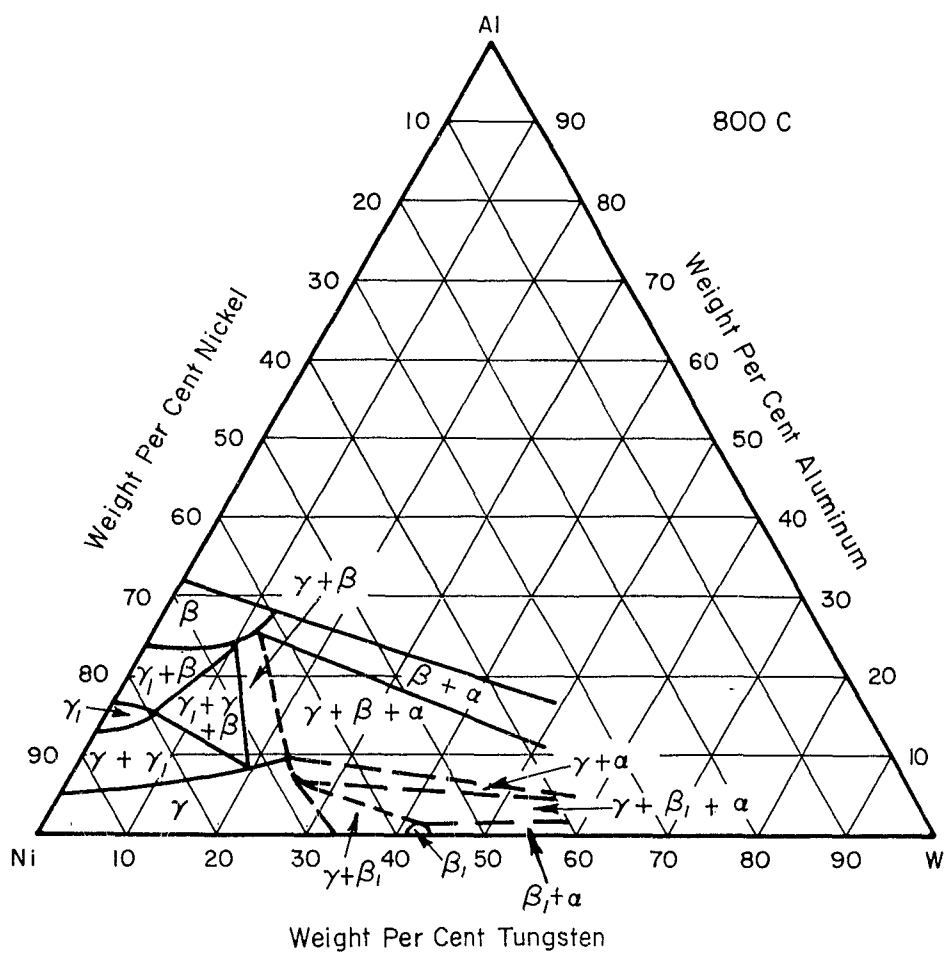
TANTALUM-SILICON-VANADIUM SYSTEM (SCHEMATIC)⁽³¹⁵⁾



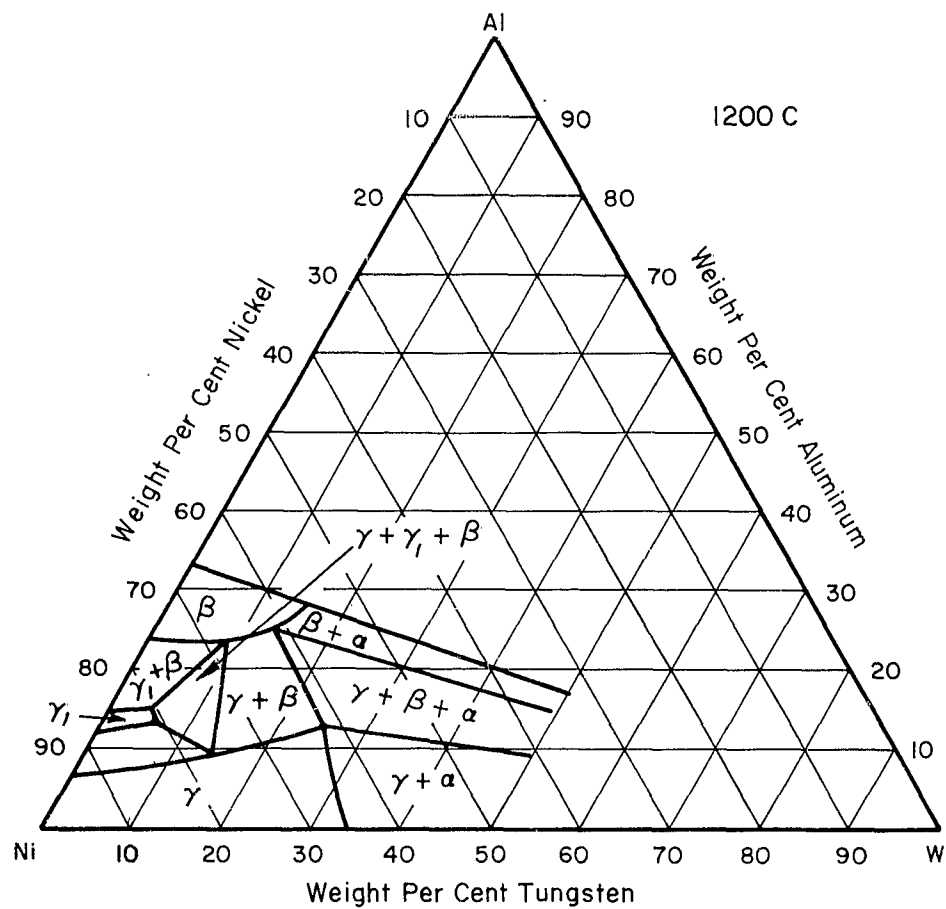
TANTALUM-TUNGSTEN-SILICON SYSTEM (TaSi_2 - WSi_2)(316)

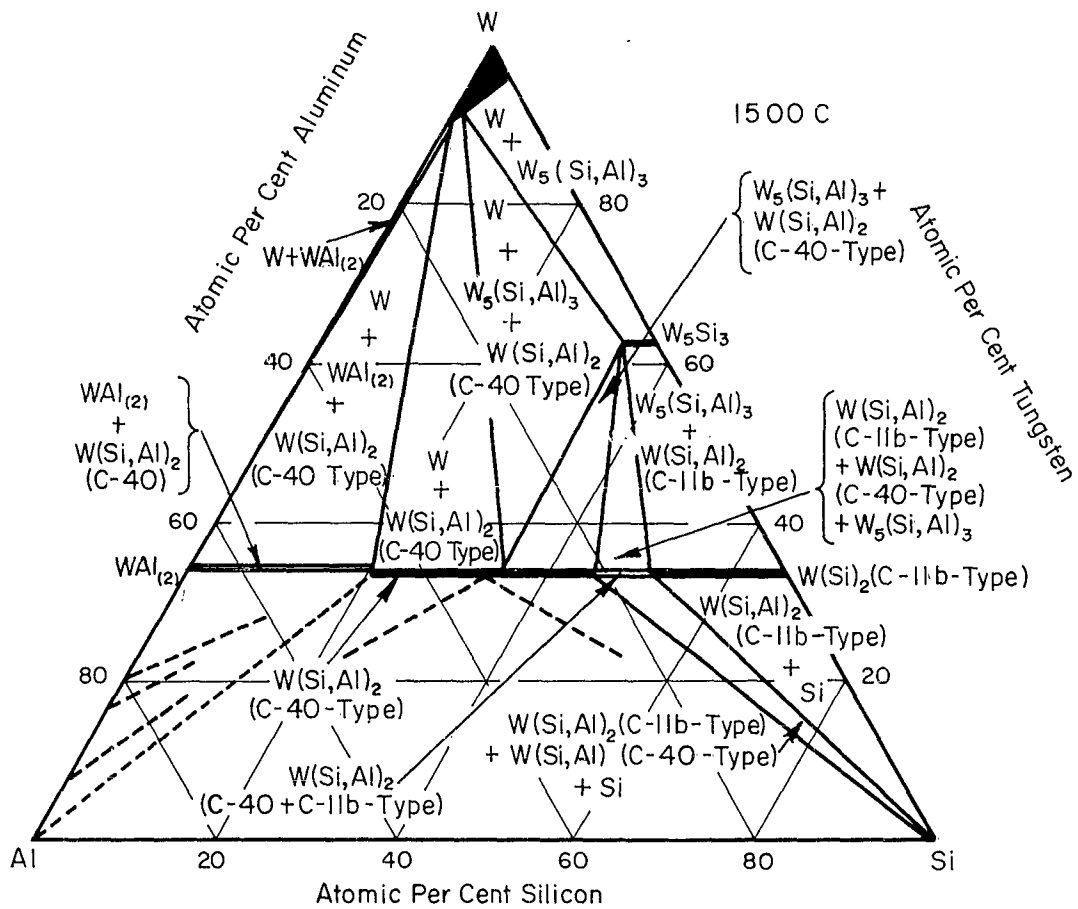


TUNGSTEN-ALUMINUM-NICKEL SYSTEM (800 C)(302)

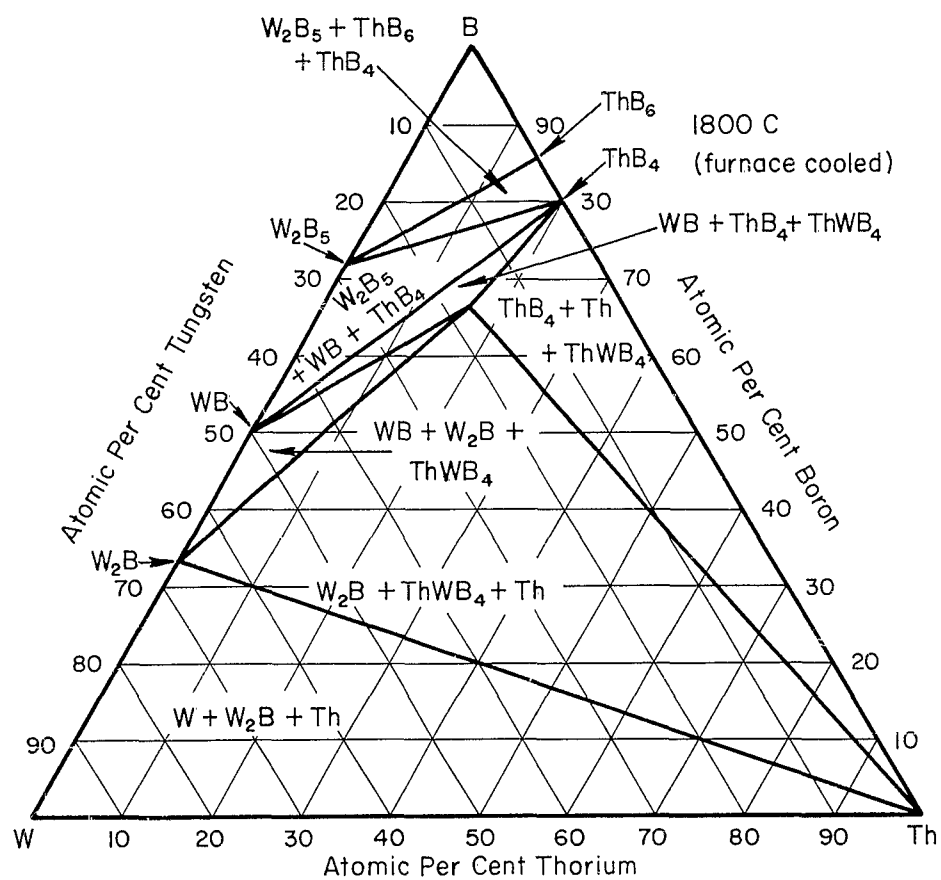


TUNGSTEN-ALUMINUM-NICKEL SYSTEM (1200 C)⁽³⁰²⁾



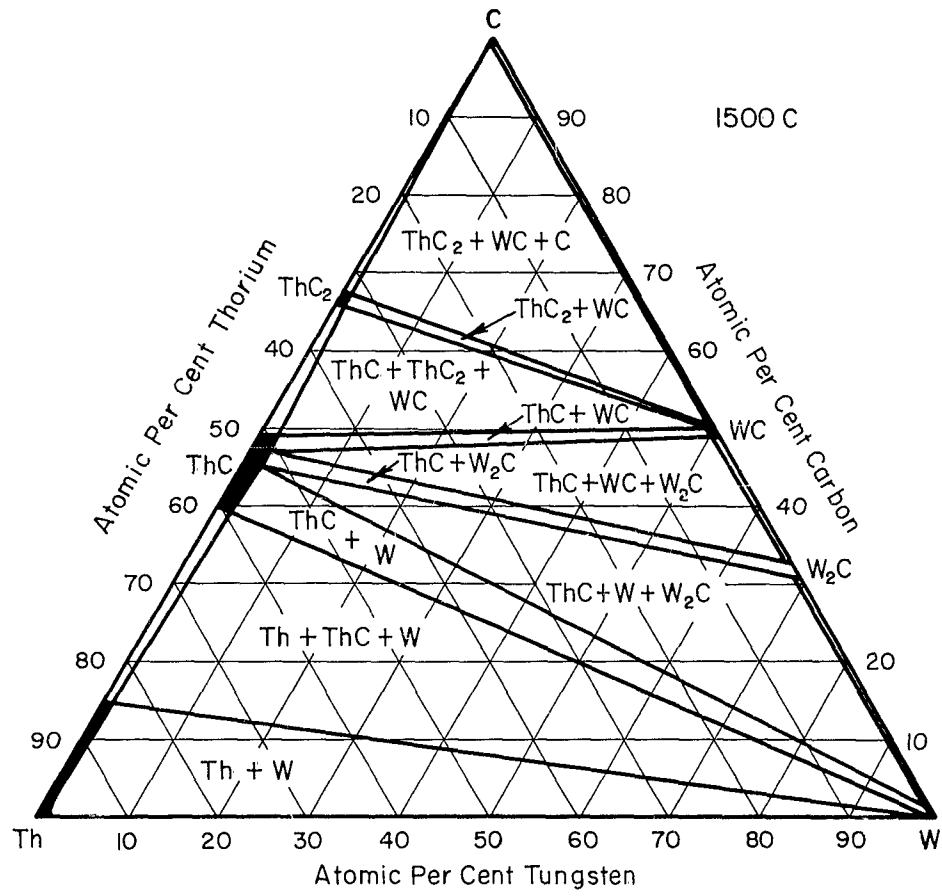
TUNGSTEN-ALUMINUM-SILICON SYSTEM (1500 C)⁽²⁶⁶⁾

TUNGSTEN-BORON-THORIUM SYSTEM (1800 C)⁽²⁹¹⁾

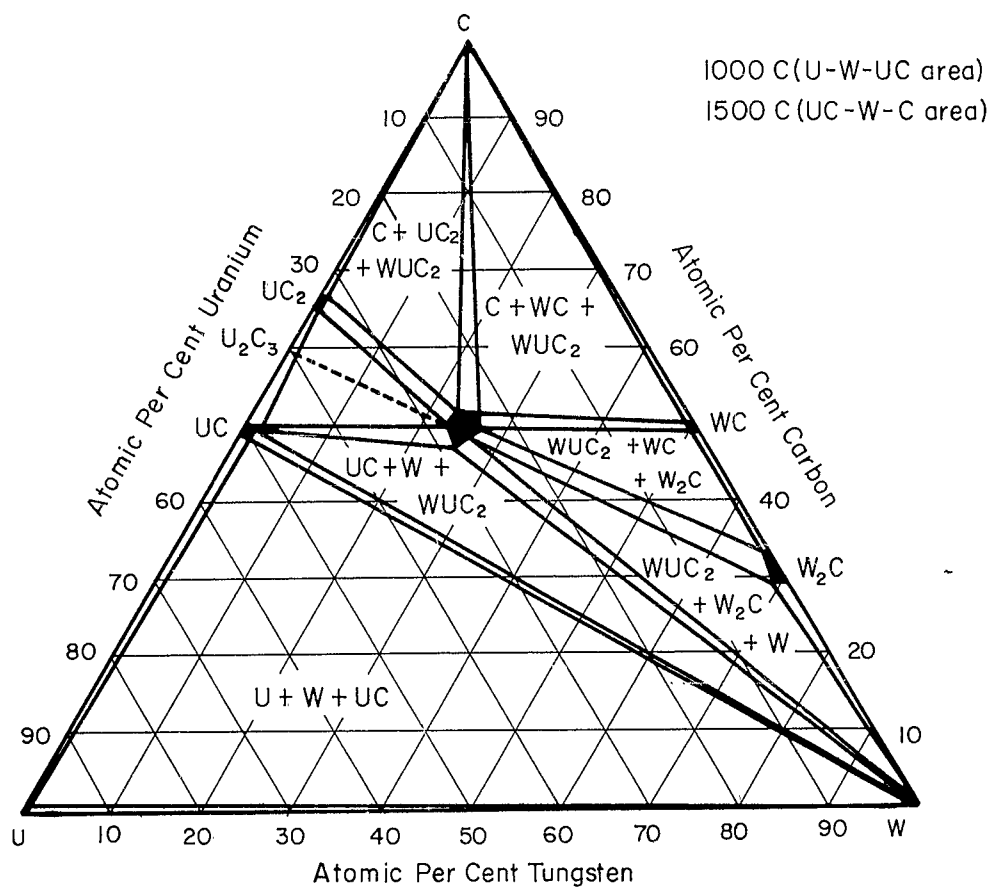


A ternary compound was found with a probable composition ThWB_4 . The compound is monoclinic with $a = 12.25 \text{ \AA}$, $b = 3.75 \text{ \AA}$, $c = 6.14 \text{ \AA}$, and $\beta = 104.1^\circ$.⁽²⁹¹⁾

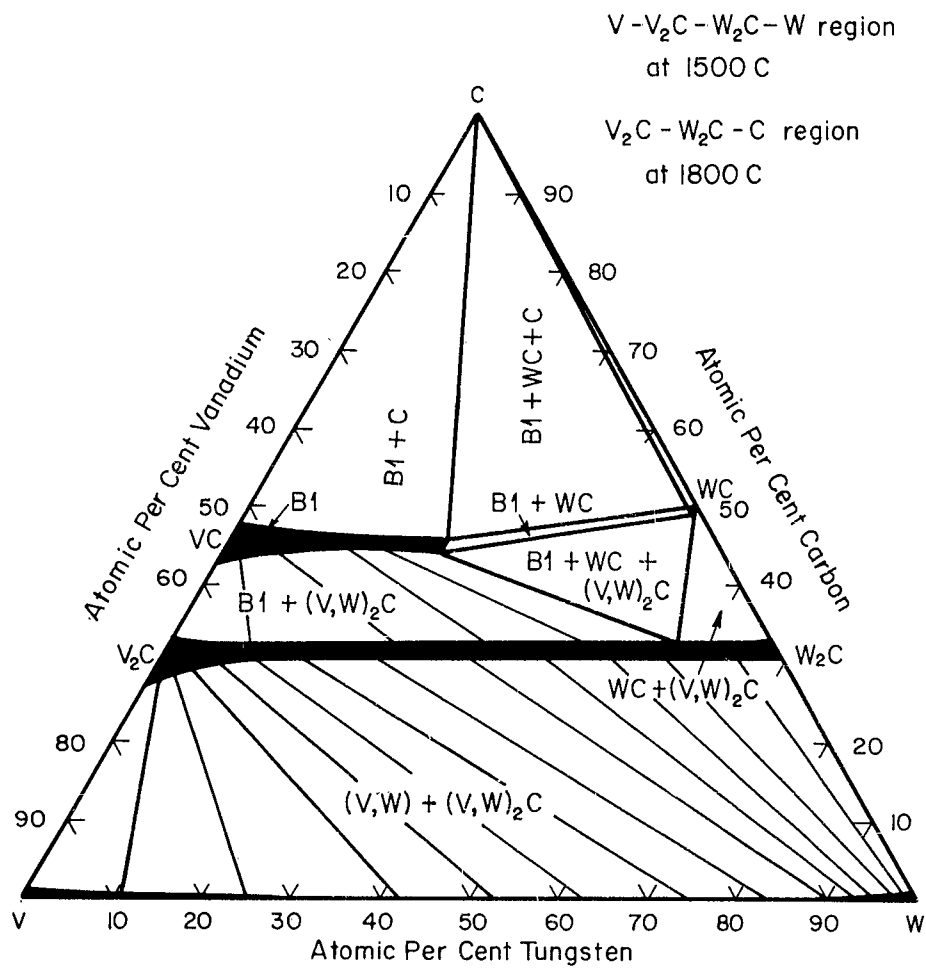
TUNGSTEN-CARBON-THORIUM SYSTEM (1500 C)⁽²⁹²⁾



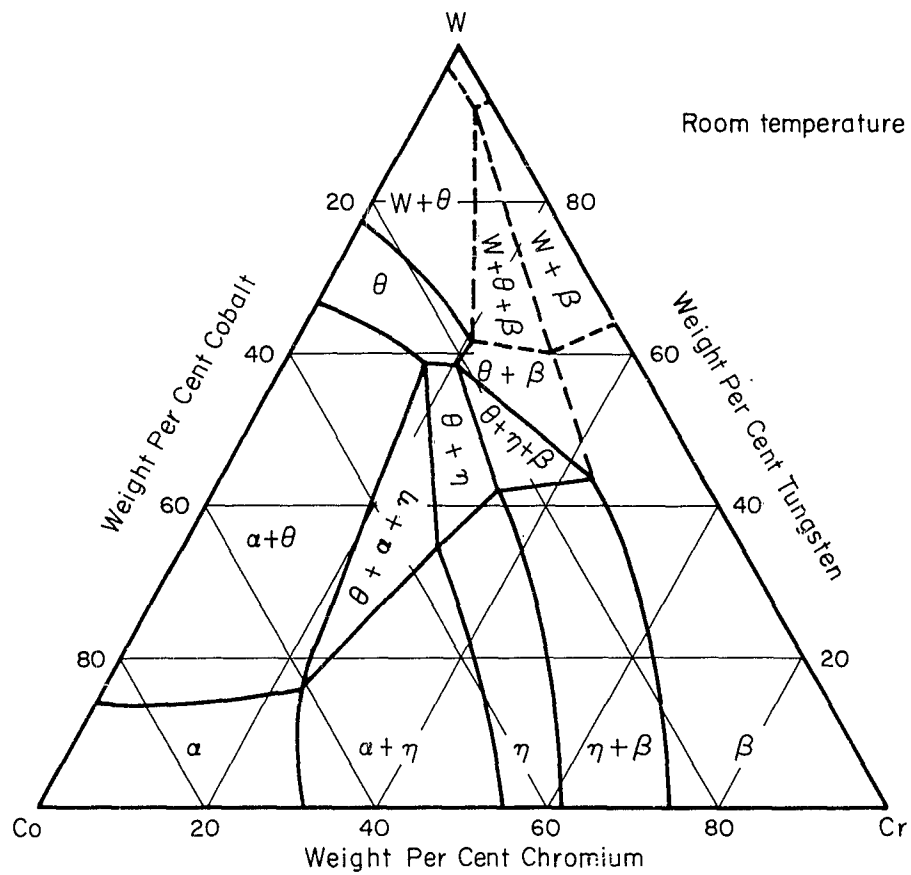
TUNGSTEN-CARBON-URANIUM SYSTEM (1000 AND 1500 C)(292)



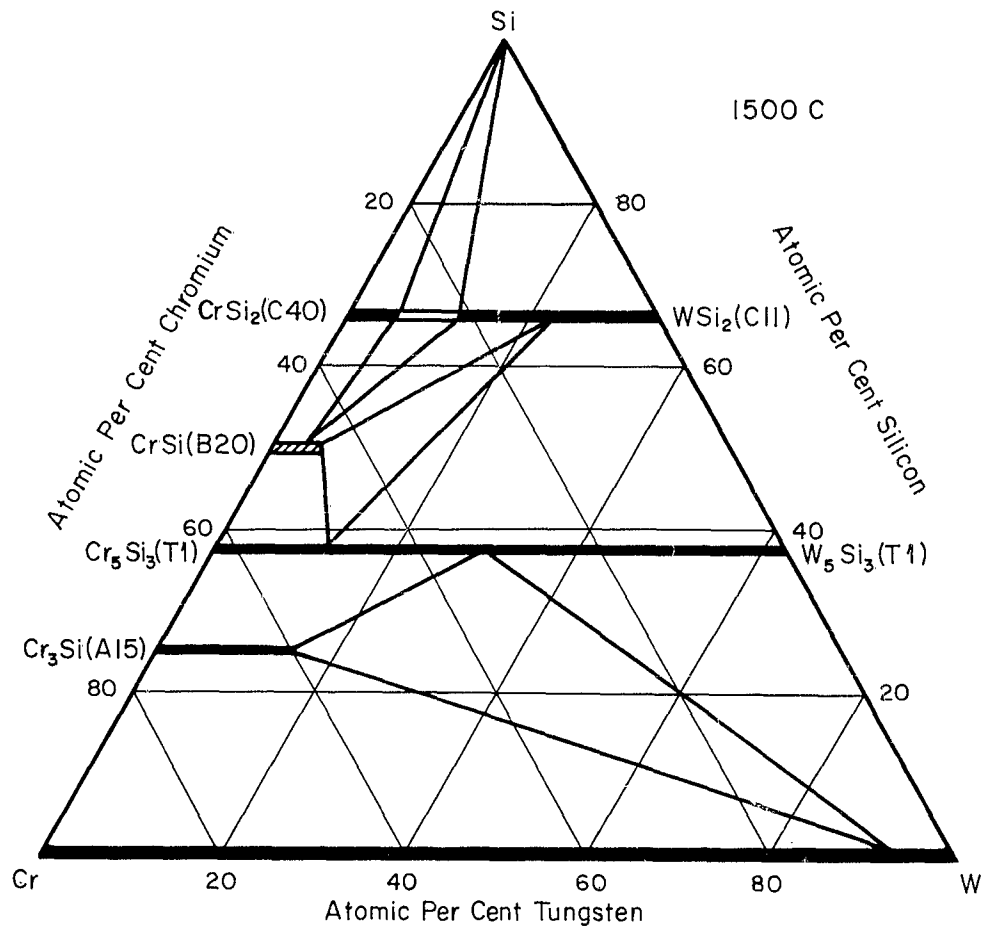
TUNGSTEN-CARBON-VANADIUM SYSTEM (1500 AND 1800 C)⁽³⁰⁹⁾



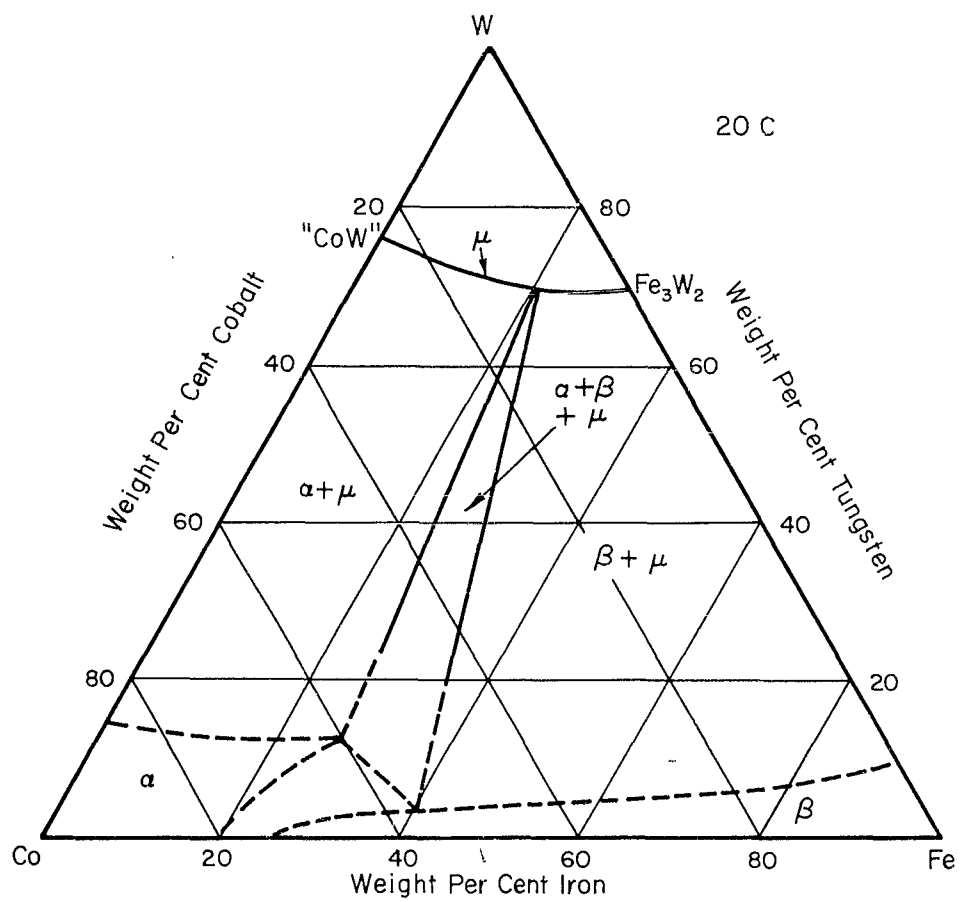
TUNGSTEN-CHROMIUM-COBALT SYSTEM (ROOM TEMPERATURE)(293)



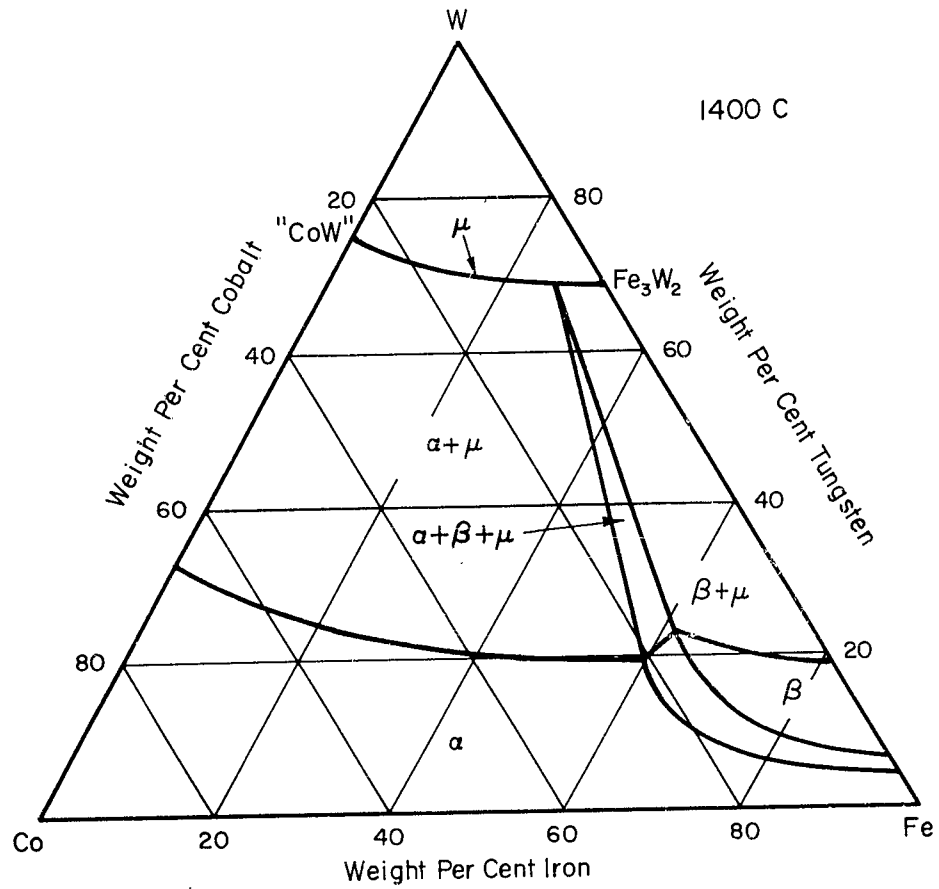
TUNGSTEN-CHROMIUM-SILICON SYSTEM (1500 C)⁽²⁹⁴⁾



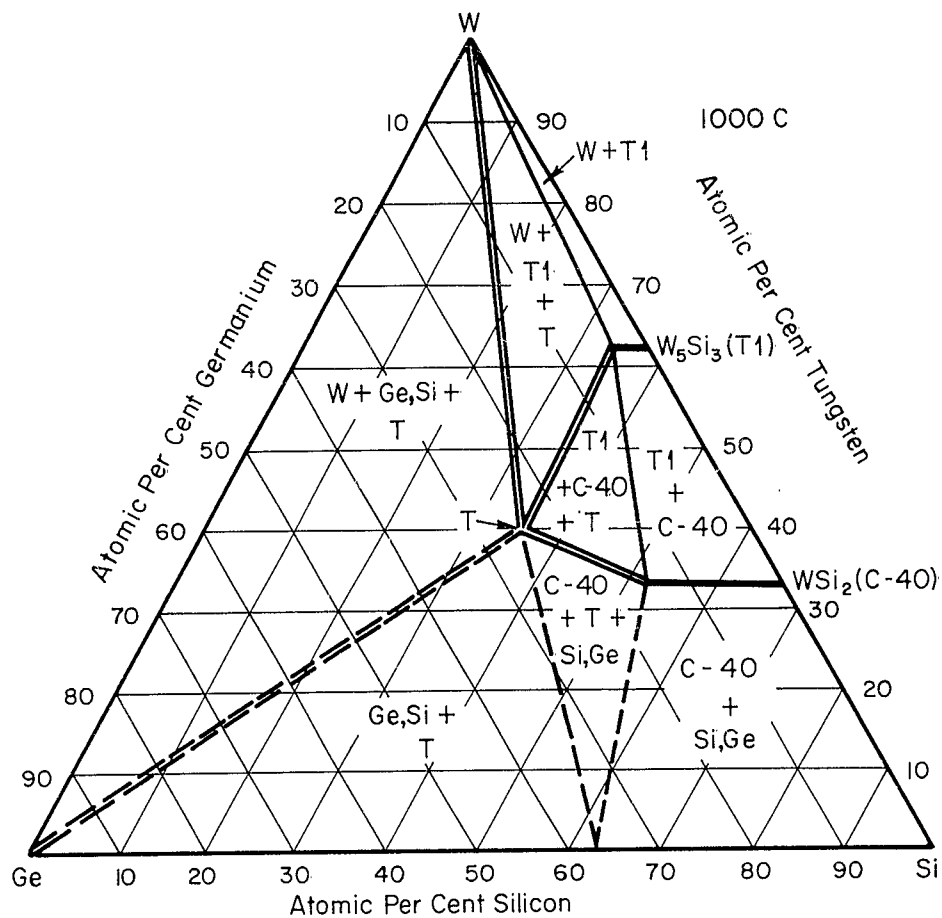
TUNGSTEN-COBALT-IRON SYSTEM (20 C)(295)



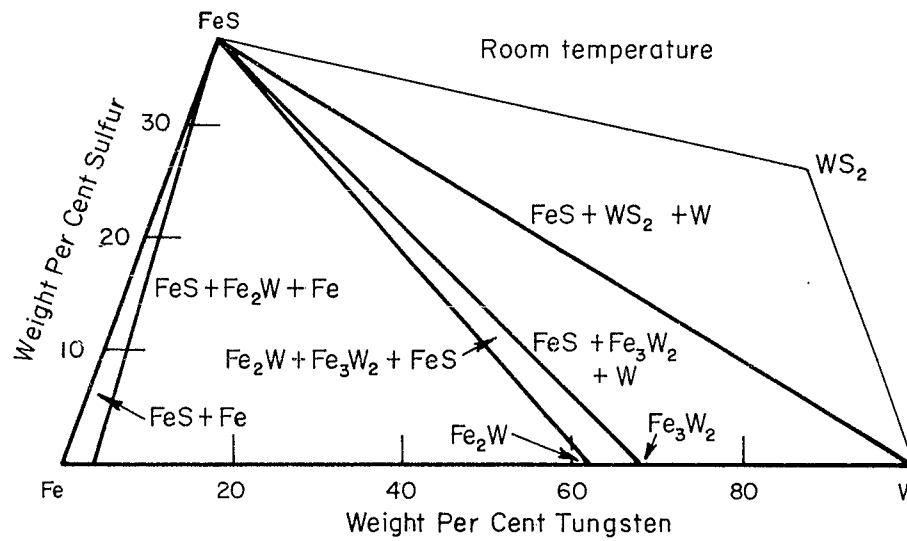
TUNGSTEN-COBALT-IRON SYSTEM (1400 C)⁽²⁹⁵⁾



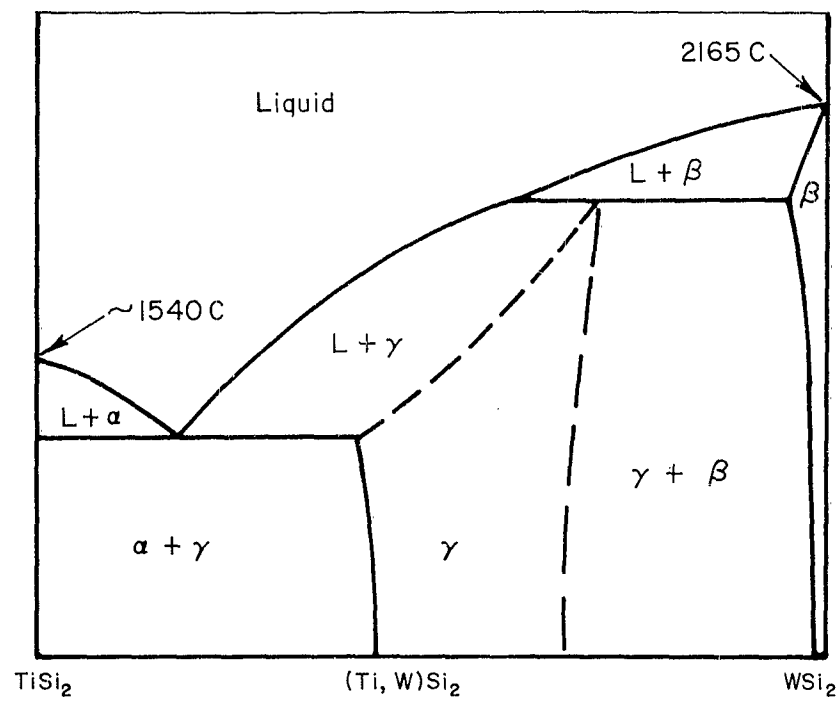
TUNGSTEN-GERMANIUM-SILICON SYSTEM (1000 C)⁽²⁹⁷⁾



TUNGSTEN-IRON-SULFUR SYSTEM (ROOM TEMPERATURE)⁽²⁹⁶⁾



TUNGSTEN-SILICON-TITANIUM SYSTEM (TiSi_2 - WSi_2 SCHEMATIC)(316)



Schematic Diagram

BIBLIOGRAPHY

- (234) Baron, V. V., and Savitskiy, Ye. M., "Structure and Properties of Nb-Al Alloys", Zhur. Neorg. Khimii, 6 (1), 182-5 (1961).
- (235) Nedumov, N. A., and Rabezova, V. I., "Alloys of the Nb-Al System", Izvest. Akad. Nauk S.S.S.R., Otdel. Tekh. Nauk, Met. i Topliva, No. 4, 68-70 (1961).
- (236) Gupta, K. P., "Sigma Phases with Aluminum", Trans. AIME, 221, 1047-1049 (1961).
- (237) Pleasance, R. J., "The Solubilities of Niobium, Cesium, and Strontium in Liquid Bismuth", J. Inst. Metals, 88, 45-46 (1959-60).
- (238) Storms, E. K., and Krikorian, N. H., "The Niobium-Niobium Carbide System", J. Physical Chem., 64, 1471-1477 (October, 1960).
- (239) Kimura, H., and Sasaki, Y., "Phase Diagram of the Niobium-Carbon System", Trans. Japan Inst. Metals, 2, 98-104 (1961).
- (240) Seebold, R. E., and Birks, L. S., "Elevated Temperature Diffusion in the Systems Nb-Pt, Nb-Se, Nb-Zn, Nb-Co, Ni-Ta, and Fe-Mo, J. Nuclear Materials, 3, 260-66 (1961).
- (241) Povov, I. A., and Shiryayeva, N. V., "State Diagram of the System Nb-Cu, Zhur. Neorg. Khimii, 6 (10), 2334-2340 (1961).
- (242) Schubert, K., Anantharaman, T. R., et al., "Einige strukturelle Ergebnisse an metallischen Phasen (6)", Naturwiss., 47, 512 (1960).
- (243) Elliott, R. P., and Komjathy, S., "Columbium-Nitrogen System", paper presented at the Columbium Metallurgy Symposium, Lake George, New York, June 10, 1960.
- (244) Savitskiy, Ye. M., Baron, V. V., and Khotinskaya, A. N., "Composition Diagram of the System Nb-Pd", Zhur. Neorg. Khimii, 6 (11), p 2603-2605 (1961).
- (245) Kimura, H., and Akira, I., "The Platinum-Niobium System", Nippon Kinzoku Gakkari, 25, 88-91 (January, 1961).
- (246) Novoselova, A. V., Grigoryan, L. A., and Simanov, Yu. P., "The Niobium-Tellurium System", Dokl. Akad. Nauk S.S.S.R., 135, 864-67 (1960).
- (247) Lundin, C. E., and Klodt, D. T., presented at ASM-AEC Joint Symposium on the Rare Earths and Related Metals, Chicago, Illinois, November, 1959.
- (248) Taylor, A., preliminary work under AF Contract AF 33(616)-8315.
- (249) Lundin, C. E., and Klodt, D. T., "The Alloy Systems of the Group V-A Metals with Yttrium", J. Inst. Metals, 90, 341-347 (May, 1962).

- (250) Taylor, A., Doyle, N., and Kagle, B., "Refractory Metal Constitution Diagrams - Part II", Westinghouse Research Laboratories, Air Force Contract AF 33(616)-7157, March, 1962.
- (251) Anderson, E., and Hume-Rothery, W., "The Equilibrium Diagram of the System Molybdenum-Ruthenium", J. Less-Common Metals, 2 (6), 443-450 (1960).
- (252) Raub, E., Beeskow, H., and Menzel, D., "Tantalum-Gold Alloys", Z. Metallk., 52, 189-93 (1961).
- (253) Nevitt, M. V., and Downey, J. W., "Sigma Phases Containing Osmium and Iridium", J. Metals, 9, 1072 (1957).
- (254) Giessen, B., "Refractory Metal Constitution Diagrams - Part II", Massachusetts Institute of Technology, Air Force Contract AF 33(616)-7157, March, 1962.
- (255) Dwight, A. E., and Beck, P. A., "Close-Packed Ordered Structures in Binary AB₃ Alloys of Transition Elements", Trans. AIME, 215 (6), 976-979 (1959).
- (256) Gebhardt, E., Seghezzi, H. D., and Fromm, E., "Investigation of the Equilibrium in the System Tantalum-Nitrogen", Z. Metallk., 52, 464-76 (1961).
- (257) Browning, B. D., "Investigation of the Pt-Rich Portion of the Platinum-Tantalum Phase Diagram", Air Force Institute of Technology, August, 1961.
- (258) Pease, L. F., Brophy, J. H., and Wulff, J., "Refractory Metal Constitution Diagrams - Part II", Massachusetts Institute of Technology, Air Force Contract AF 33(616)-7157.
- (259) Goldschmidt, H. S., and Brand, J. A., "Investigation Into the Tungsten-Rich Regions of the Binary Systems Tungsten-Carbon, Tungsten-Boron and Tungsten-Beryllium", Air Force Contract No. AF 61(052)-306, 1959-1961.
- (260) Dolloff, R. T., and Sara, R. V., "Research Study to Determine the Phase Equilibrium Relations of Selected Carbides at High Temperatures", Research Laboratory National Carbon Company, Progress Report No. 7, May 1, 1961, to July 31, 1961.
- (261) Raub, E., and Walter, P., Festschrift aus Anlass des 100-jährigen Jubiläums der Firma W. C. Heraeus G.m.b.H., Hanau (1951), pp 124-146.
- (262) Rapperport, E. J., and Smith, M. F., "Refractory Metal Constitution Diagrams - Part II", Nuclear Metals, Inc., Air Force Contract AF 33(616)-7157, March, 1962.
- (263) Smithells, C. J., Tungsten, Chemical Publishing Company, New York (1953), p 266.
- (264) St. Pierre, G. R., Ebihara, W. T., Pool, M. J., and Speiser, R., "The Tungsten-Oxygen System", Trans. AIME, 224, 259-264 (April, 1962).

- (265) Tylkina, M. A., Polyakova, V. P., and Savitskiy, Ye. M., "State Diagram of the System Palladium-Tungsten", Zhur. Neorg. Khimii, 6 (6), 1471-1473 (1961).
- (266) Nowotny, H., Brukl, C., and Benesovsky, F., "Investigation of the Tantalum-Aluminum-Silicon and Tungsten-Aluminum-Silicon Systems", Mh. Chemie, 92 (1), 116-127 (1961).
- (267) Edshammar, L., and Holmberg, B., "The σ -Phase Ta_2Al ", Acta Chem. Scand., 14, 1219 (1960).
- (268) Brukl, C., Nowotny, H., and Benesovsky, F., "Untersuchungen in den Dreistoff systemen: V-Al-Si, Nb-Al-Si, Cr-Al-Si, Mo-Al-Si bzw. Cr(Mo)-Al-Si", Mh. Chemie, 92 (5), 967-980 (1961).
- (269) Popov, I. A., and Rabezova, V. I., "Equilibrium Phase Diagram of the Niobium-Titanium-Aluminum System", Zhur. Neorg. Khim., 7, 436-39 (1962).
- (270) Nowotny, H., Benesovsky, F., Rudy, E., and Wittmann, A., "Aufbau und Zunderverhalten von Niob-Bor-Silicium-Legierungen", Monatsh. Chemie, 91 (5), 975-990 (1960).
- (271) Rudy, E., "Über Hafniumkarbid enthaltende Karbedsysteme", Mh. Chem., 91 (1), 180-181 (1960).
- (272) Benesovsky, F., and Rudy, E., "Zur Kenntnis der Systeme Uran-Zirkonium-(Hafnium, Niob, Tantal) Kohlenstoff", Plansee Pulvermetal., 9, 65-76 (1961).
- (273) Svechnikov, V. N., and Pan, V. M., "An Investigation of the Chromium-Nickel-Niobium Ternary System", Issledovaniya Po Zharoprochnym Splavam, 6, 240-250 (1960).
- (274) Goldschmidt, H. J., and Brand, J. A., "The Constitution of the Chromium-Niobium-Silicon System", J. Less-Common Metals, 3, 34-43 (1961).
- (275) Svechnikov, V. N., Kocherzhinskiy, Yu. A., Maystrenko, Ye. Ye., Pan, V. M., and Shurin, A. K., "A Study of the System Cr-Nb-V", Voprosy fiziki metallov i metallovedeniya, Kiev (1959), pp 120-132.
- (276) Vogel, R., and Bleichroth, W., "Das Dreistoffsystem Eisen-Phosphor-Niob", Arch. Eisenhutten., 33 (3), 208 (1962).
- (277) Rudy, E., Benesovsky, F., and Sedlatschek, K., "Untersuchungen im System Niob-Molybdän-Kohlenstoff", Mh. Chemie, 92 (4), 841-855 (1961).
- (278) Goldschmidt, H. J., and Brand, J. A., "The Constitution of the Chromium-Niobium-Molybdenum System", J. Less Common Metals, 3, 44-61 (1961).
- (279) Zakharova, M. I., Prokoshkin, D. A., "Examination of the Niobium-Molybdenum-Chromium System", Izvest. Akad. Nauk S.S.S.R., Otdel. Tekh. Nauk, Met. I. Topliva, 4, 59-67 (1961).

- (280) Ivanov, V. E., and Badajeva, T. A., "Phase Diagrams of Certain Ternary Systems of Uranium and Thorium", Proceedings of the Second United Nations International Conference on the Peaceful Uses of Atomic Energy", 6, 139-155 (1958).
- (281) Elliott, R. P., and Komjathy, S., "Niobium Phase Diagrams", Armour Research Foundation, USAEC Contract No. AT(11-1)-515, May 15, 1960.
- (282) Mikheev, V. S., and Belousov, O. K., "Fusion Diagram of the System Titanium-Zirconium-Niobium", Zhur. Neorg. Khim., 6, 1905-08 (1961).
- (283) Guard, R. W., and Smith, E. A., "Constitution of Nickel-Base Ternary Alloys, Ni-Mo-Al and Ni-Mo-Si Systems", J. Inst. Metals, 88, 283-287 (February, 1960).
- (284) Kolomytsev, P. T., et al., "Phase-Compositions of the System Nickel-Molybdenum-Boron", Atomnaya Energiya, 10, 276-277 (March, 1961).
- (285) Wittmann, A., Nowotny, H., and Boller, H., "Ein Beitrag zum Dreistoff Titan-Molybdän-Bor", Mh. Chem., 91 (4), 608-619 (1960).
- (286) Vogel, R., and Gerhardt, R., "Das System Eisen-Molybdän-Silizium", Arch. Eisenhutten., 32 (1), 53 (1961).
- (287) Kornilov, I. I., Polyakova, R. S., "Examination into Properties of Alloys of Titanium-Vanadium-Molybdenum System", Izvest. Akad. Nauk S.S.S.R., Otdel Tekh. Nauk, Met. I. Toplivo, 4, 76-82 (1961).
- (288) Grum-Grzhimailu, N. V., and Prokof'ev, D. I., "X-ray Investigation at High Temperatures of Solid Solutions of the System Chromium-Tungsten-Molybdenum", Zhur. Neorg. Khim., 6, 1154-64 (1961).
- (289) Lavendel, H. W., "Alloys of Tantalum Diboride With Iron, Cobalt, and Nickel", Plansee Pulvermet., 9 (1-2), 80-95 (1961).
- (290) Köster, W., and Becker, C., "The System Iron-Cobalt-Tantalum", Arch. Eisenhüttenwesen, 8, 557-560 (1935).
- (291) Pitman, D. T., and Das, D. K., "A Study of the Thorium-Tungsten-Boron System", J. Electrochem Soc., 107 (9), 763-66 (1960).
- (292) Rudy, Erwin, Rudy, Elisabeth, and Benesovsky, F., "Untersuchungen in den Systemen Thorium-Wolfram-Kohlenstoff und Uran-Wolfram-Kohlenstoff, Monatsh. Chemie, 93 (2), 532-534 (1962).
- (293) Köster, W., "The Ternary System Cobalt-Chromium-Tungsten", Z. Metallkunde, 25, 22-27 (1933).
- (294) Kieffer, R., Schob, O., Nowotny, H., and Benesovsky, F., "Untersuchungen in den Dreistoffen: Cr-W-Si and Mo-W-Si", Monatsh. Chemie, 93 (2), 519-521 (1962).

- (295) Köster, W., and Tonn, W., "The System Iron-Cobalt-Tungsten", Arch. Eisenhüttenwesen, 5, 431-440 (1932).
- (296) Vogel, R., and Weizenkorn, H., "Über das Dreistoffsystem Eisen-Schwefel-Wolfram", Arch. Eisenhüttenwesen, 32 (6), 419 (1961).
- (297) Nowotny, H., Benesovsky, F., and Brukl, C., "Untersuchungen im System: Wolfram-Silicium-Germanium", Mh. Chemie, 92 (2), 365-370 (1961).
- (298) Savitskii, E. M., and Kopetskii, Ch. V., "Phase Diagram of the Manganese-Tantalum System", Zhur. Neorg. Khimii, 5 (11), 2638 (1960).
- (299) Savitskii, E. M., and Kopetskii, Ch. V., "Physicochemical Behavior of Manganese with Niobium", Zhur. Neorg. Khimii, 5 (3), 755 (1960).
- (300) Gregor'ev, A. T., Sokolovskaya, E. M., Maksimova, M. V., Sokolova, I. G., and Nedumov, N. A., "Polymorphic Transformations of Chromium in Alloys With Tantalum", Zhur. Neorg. Khimii, 5 (11), 2640 (1960).
- (301) Wyder, W. C., and Hoch, M., "The System Niobium (Columbium)-Titanium-Zirconium-Oxygen at 1500 C", Naval Research Contract Nonr-2168(02), Project NR 039-032; also AIME Trans, 224, 373-378 (1962).
- (302) Budberg, P. B., "Study of Alloys of the Ternary System Nickel-Aluminum-Tungsten", Zhur. Neorg. Khimii, 3 (3), 694-698 (1958).
- (303) Savitsky, Ye. M., and Baron, V. V., "Investigation of the System Vanadium-Molybdenum-Silicon", Zhur. Neorg. Khimii, 7 (5), 1117-1125 (1962).
- (304) Wyman, L. L., Cuthill, J. R., Moore, G. A., Park, J. J., and Yakowitz, H., "Intermediate Phases in Superconducting Niobium-Tin Alloys", Journal Research Nat. Bur. Standards - A. Physics and Chemistry, 66A (4), 351-63 (July-August, 1962).
- (305) Dragsdorf, R. D., and Forgeng, W. D., "The Intermetallic Phases in the Cobalt-Tantalum System", Acta Cryst., 15 (6), 531-534 (1962).
- (306) McMasters, O. D., and Larsen, W. L., "Phase Equilibria in the Thorium-Tantalum System", J. Less-Common Metals, 3, 312-320 (1961).
- (307) Anderson, E., and Hume-Rothery, W., "The Equilibrium Diagram of the System Molybdenum-Rhodium", J. Less-Common Metals, 2, 19-28 (1960).
- (308) Petzow, G., and Gitlesen, G., "Aufbau und Eigenschaften von Uran-Vanadium-Molybdän Legierungen", Z. Metallkde., 53 (8), 519-520 (1962).
- (309) Rudy, Erwin, Benesovsky, F., and Rudy, Elizabeth, "The System Vanadium-Tungsten-Carbon", Monatsh. Chemie, 93, 693-707 (1962).
- (310) Glazov, V. M., and Mal'tsev, M. V., "Study of the Al-Ta Phase Diagram", Izvest. Akad. Nauk, S.S.S.R., Otdel. Tekh. Nauk., No. 4, 131-36 (1956).

- (311) Baron, V. V., Yefimov, Yu. V., and Savitskiy, Ye. M., "Structure and Properties of the Alloys of the Vanadium-Molybdenum System", *Izvest. Akad. Nauk, S.S.S.R., Otdel. Tekh. Nauk*, No. 4, 36-40 (1958).
- (312) Guard, R. W., and Smith, E. A., "Constitution of Nickel-Base Ternary Alloys. IV. Nickel-Molybdenum-Titanium System", *J. Inst. Metals*, 88, 369-374 (1960).
- (313) Vogel, R., and Horstmann, D., "The Iron-Iron Phosphide-Molybdenum Phosphide-Molybdenum Constitution Diagram", *Arch. Eisenhüttenwesen*, 24, 369-374 (1954).
- (314) Hoch, M., and Desjardins, M., "The System Molybdenum-Titanium-Zirconium-Oxygen at 1500 C", *Trans. AIME*, 224, 821-827 (August, 1962).
- (315) Nowotny, H., Lux, B., and Kudielka, H., "The Behavior of High-Melting Metal Silicides With Respect to Boron, Carbon, Nitrogen and Oxygen", *Monatsh. Chemie*, 87 (3), 447-470 (1956).
- (316) Kudielka, H., and Nowotny, H., "Disilicide Systems", *Monatsh. Chemie*, 87 (3), 471-482 (1956).
- (317) Mints, R. S., Byelyayeva, G. F., and Malkov, Yoo. S., "Constitution Diagram of the System $\text{Ni}_3\text{Al-Ni}_3\text{Nb}$ ", *Zhur. Neorg. Khim.*, 7 (10), 2382-2387 (1962).
- (318) Bannister, G. H., and Murray, J. R., "Some Observations on Uranium-Molybdenum-Niobium Alloys", *J. Less-Common Metals*, 2, 372-382 (1960).
- (319) Savitskiy, E. M., and Zakharov, A. M., "Phase Diagrams of the Ternary System Niobium-Tungsten-Zirconium", *J. Neorg. Khimii*, 7 (11), 2575-2580 (1962).
- (320) Kubaschewski, O., and Speidel, H., "Oxidation Resistance and Some Phase Relationships in the System Chromium-Tantalum-Nickel", *J. Inst. Metals*, 75, 417-430 (1948-49).
- (321) Dokukina, N. V., and Shamrai, F. I., "Phase Equilibrium in the System W-Nb-Si and Some Properties of Alloys", *Poroshkovaya Met., Akad. Nauk Ukr. SSR*, 2 (6), 32-41 (1962).
- (322) Clare, J. W. H., "Compounds Present in Aluminum-Rich Alloys of the Aluminum-Molybdenum System", *J. Inst. Metals*, 89, 232-234 (1960-61).
- (323) Auld, J. H., and Ryan, N. E., "The Solid Solubility of Tantalum in Chromium", *J. Less-Common Metals*, 3, 221-225 (1961).
- (324) Hartley, C. S., Baum, W. L., Fisher, D. W., and Rapperport, E. S., "Phase Relationships in Tantalum-Rich Tantalum-Ruthenium Alloys at 1500 C", *WADD-TN-60-288*, March, 1961.
- (325) Deardorf, D. K., and Kato, H., "The Constitution Diagram of Tungsten-Hafnium. Discussion", *Trans. AIME*, 227, 264-65 (1963).

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